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## Ordering state in orthopyroxene as determined by precession electron diffraction

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18  
19

20 **Abstract**

21  $\text{Fe}^{2+}$  and Mg distribution on octahedral M1 and M2 sites of the orthopyroxene  
22 structure is an indicator of the cooling rate and closure temperature of the mineral. It is  
23 generally obtained by single crystal X-ray diffraction, which is limited in spatial resolution. In  
24 this work, we determine the cationic distribution at a sub-micron scale in a transmission  
25 electron microscope using precession electron diffraction. Two orthopyroxene samples  
26 coming from the same metamorphic rock are studied, a naturally ordered one and a disordered  
27 one. The latter was obtained from the ordered sample by annealing at high temperature and  
28 rapid quenching. Both samples have been first studied in X-ray diffraction and then in  
29 precession electron diffraction. Intensities recorded in zone-axis precession electron  
30 diffraction experiments have been quantitatively analyzed and compared to simulations,  
31 taking into account dynamical interactions between diffracted beams. Our structure  
32 refinement results are in good agreement with those obtained by single-crystal X-ray  
33 diffraction. They enable to distinguish between the ordered sample and the disordered one in  
34 terms of the observed molar fractions of Fe at M1 and M2 sites. We discuss the sensitivity of  
35 the method as a function of experimental parameters. The larger dispersion of the results  
36 obtained on the ordered specimen is attributed to structural heterogeneities inherent to the  
37 sample.

38

39 **Keywords:** ordering, orthopyroxene, precession electron diffraction, site occupancy, structure  
40 refinement, transmission electron microscopy.

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## INTRODUCTION

45

46  $\text{Fe}^{2+}$  and Mg distribution on octahedral M1 and M2 sites of the orthopyroxene (OPX)  
47 structure is an indicator of the cooling rate and closure temperature of the mineral (Ganguly  
48 1982; Ganguly et al. 1994; Stimpfl et al., 1999; Stimpfl et al., 2005). These data are of great  
49 importance, as they permit the retrieval of the thermal history of the crystal (closure  
50 temperature and cooling rate). The cationic distribution is generally accessible thanks to the  
51 quantitative analysis of diffracted intensities as obtained by X-ray diffraction (XRD), leading  
52 to the determination of atomic positions and site occupancies with a good accuracy.  
53 Nevertheless, XRD is limited in spatial resolution. Contradictory results in cooling rate  
54 determination based on site occupancies as determined by XRD have been explained by the  
55 occurrence of microstructural features such as local variations of composition, exsolution  
56 lamellae and Guinier-Preston zones (Zema et al, 1999; Cámara et al., 2000; Heinemann et al.,  
57 2008). These features can only be revealed by transmission electron microscopy, whereas  
58 XRD analysis generally leads to averaged information, which may induce misinterpretation.

59 In this work, we present results on site occupancy determination obtained at a  
60 microscopic scale in a transmission electron microscope (TEM) using precession electron  
61 diffraction (PED). Since its development in 1994 (Vincent and Midgley 1994), PED has  
62 become an efficient and widely used method for solving structures of inorganic compounds  
63 (Boulahya et al. 2007 (perovskite related  $\text{LaBaCuCoO}_{5.2}$  and  $\text{Ba}_6\text{Mn}_5\text{O}_{16}$ ); Gemmi et al. 2007  
64 (minerals uvarovite and åkermanite); Boullay et al. 2009 (mineral brownmillerite); Mugnaioli  
65 et al. 2009 (inorganic salt  $\text{BaSO}_4$ ); Gemmi et al. 2010 (titanate  $\text{Li}_4\text{Ti}_8\text{Ni}_3\text{O}_{21}$ ); Hadermann et  
66 al. 2010 (perovskite related  $\text{Pb}_{13}\text{Mn}_9\text{O}_{25}$ ); White et al. 2010 (tin oxide  $\text{Sn}_3\text{O}_4$ ); Hadermann et

67 al. 2011 (mixed phosphate  $\text{Li}_2\text{CoPO}_4\text{F}$ ); Klein et al. 2011 (oxides  $\text{Mn}_2\text{O}_3$  and  $\text{PbMnO}_{2.75}$ ;  
68 Palatinus et al. 2011 (copper silicide-germanide  $\text{Cu}_3(\text{Si,Ge})$ ); Song et al. 2012  
69 (hydroxyapatite)). At this stage, solving a structure means determining its unit cell  
70 parameters, its space group and the position of most of the atoms within the unit cell.  
71 Nevertheless, another important goal in structural analysis is the structure refinement, i.e. the  
72 accurate determination of all the atomic positions and their occupancy. Unlike X-rays,  
73 electrons interact strongly with matter and continuous exchange of electrons between  
74 transmitted and diffracted beams occurs when they are passing through the crystal, leading to  
75 so-called *dynamical effects*. Accurate simulation of electron diffraction data thus requires the  
76 use of dynamical diffraction theory. In this context, the main advantage of PED for structure  
77 solving is the reduction of the dynamical effects (Gjønnnes et al. 1998; Eggeman et al. 2010;  
78 Sinkler and Marks, 2010), making the intensities more related to the square of the structure  
79 factors of reflections. Nevertheless, to date very few attempts have been made to treat PED  
80 data using dynamical theory for structural refinement (Own et al. 2006; Oleynikov et al. 2007;  
81 Dudka et al. 2007; Sinkler et al. 2010). In most other cases the refinement was based on the  
82 comparison of experimental diffracted intensities with simulated ones calculated in the  
83 kinematical approximation, i.e. considering that diffracted intensities as proportional to the  
84 square of the structure factors. The refinement results using kinematical approximation show  
85 that dynamical effects must be taken into account, if accurate structure parameters are needed.  
86 However, to our knowledge, only one structure refinement using dynamical theory has been  
87 reported (Dudka et al. 2007) with silicon as a test sample. In this work, we show that when  
88 the structure is partially known, dynamical analysis of intensities as obtained using PED leads  
89 to reliable and reasonably accurate determination of structural parameters such as atomic  
90 occupancy factors on specific sites of the structure. Applied to natural OPX samples, also  
91 characterized by single crystal XRD, our PED analysis enables an unambiguous

92 discrimination between an ordered sample (natural, untreated) and a disordered one (heat-  
93 treated and quenched).

94

95

## EXPERIMENTAL

96

### 97 **Specimen selection and heat treatments**

98 The studied specimens are natural OPX ( $\text{Mg}_{1.4}\text{Fe}_{0.6}\text{Si}_2\text{O}_6$ ) single crystals from  
99 granulite rocks of the Wilson Terrane, North Victoria Land, Antarctica (crystal label B22,  
100 Tribaudino and Talarico 1992). The ratio  $\text{Mg}/(\text{Fe}+\text{Mg})$  is close to 0.70 as previously  
101 determined by electron microprobe (Tarantino et al. 2002). Small amounts of Ca and other  
102 minor elements such as Ti, Al, and Cr are also present. They were not considered in the  
103 present analysis. Four crystals were selected and used for the X-ray single-crystal diffraction  
104 to check for the homogeneity of the samples. To enable a direct comparison with disordered,  
105 but otherwise similar sample, two of these crystals have also been heated for 48 hours at  
106  $1000^\circ\text{C}$ . They were sealed (after alternately washing with nitrogen flux and vacuuming) into a  
107 small silica tube together with an iron-wüstite buffer and then heated in a vertical furnace.  
108 Inside the silica tube, the crystals and the buffer were put into two small separate Pt crucibles  
109 to avoid contact between them. Heated samples were then quenched by dropping the tube into  
110 cold water. One untreated and one heat-treated crystals were then selected for TEM analysis.

111

### 112 **X-ray single-crystal diffraction and structure refinement**

113 Intensity data were collected at the Dipartimento di Scienze della Terra e  
114 dell'Ambiente, Università di Pavia, on a three-circle Bruker AXS SMART APEX  
115 diffractometer, equipped with a CCD detector (graphite-monochromatized  $\text{MoK}\alpha$  radiation,  $\lambda$   
116  $= 0.71073 \text{ \AA}$ , 55 kV, 30 mA) and a monocap collimator. The Bruker SMART software

117 package was used. A total of 3600 frames (frame resolution  $512 \times 512$  pixels) were collected  
118 with four different goniometer settings using the  $\omega$ -scan mode (scan width:  $0.2^\circ \omega$ ; exposure  
119 time: 5-10 s·frame<sup>-1</sup>; detector-sample distance: 4.02 cm). About 14500 reflections were  
120 collected. Completeness of the measured data was achieved up to  $37^\circ \theta$ . The Bruker SAINT+  
121 software was used for data reduction, including intensity integration, background and  
122 Lorentz-polarization corrections. The semi-empirical absorption correction of Blessing  
123 (1995), based on the determination of transmission factors for equivalent reflections, was  
124 applied using the program SADABS (Sheldrick, 1996). The unit-cell parameters were  
125 obtained by a least-squares procedure from the positions of about 8000 reflections in the  $\theta$ -  
126 range  $3 - 37^\circ$ . The observed  $F_o^2$  values were then treated with a full-matrix least-squares  
127 refinement in *Pbca* space group by SHELX-97 (Sheldrick, 2008), using individual weights  
128 and the weighting scheme suggested by the program. No threshold or cutting of low intensity  
129 reflections was applied, following the recommendations of Merli et al. (2002) suggested by  
130 the leverage analysis applied to the orthopyroxene. The atomic scattering curves were taken  
131 from International Tables for X-ray Crystallography (Ibers and Hamilton, 1974). Neutral vs.  
132 ionized scattering factors were refined in all sites that are not involved in chemical  
133 substitutions (Hawthorne et al. 1995) and complete ionization for Mg and Fe in M1 and M2  
134 sites was assumed. The extinction correction was applied with the procedures of program  
135 SHELX-97. In order to get a better comparability of the refinement results obtained using  
136 XRD and PED data, structure refinements from XRD data have also been achieved in the  
137 same conditions as previously described but limiting the resolution to that of PED data ( $d =$   
138  $0,7124 \text{ \AA}$ ).

139

140 **TEM observations and precession electron diffraction**

141 Thin foils for TEM observations were prepared from both the untreated and heat-  
142 treated samples. Slabs about 50 nm thick normal to the [001] orientation have been cut from  
143 the single crystal grains by focused ion beam (FIB) technique (FEI Strata DB 235 FIB-  
144 FESEM) at IEMN (Institute of Micro and Nano Electronics, University Lille 1).

145 TEM observations were performed at University Lille 1 with a LaB<sub>6</sub> FEI Tecnai G2-  
146 20 operated at 200 kV and equipped with a DIGISTAR precession system (Nanomegas). In  
147 the PED technique, the incident beam is scanned at a constant precession semi-angle (ranging  
148 typically from 1° to 4°) around the optical axis, in combination with an opposite and  
149 synchronized descans of the transmitted and diffracted beams below the specimen (Vincent  
150 and Midgley 1994). During the precession movement, the reciprocal lattice nodes are thus  
151 swept through the Ewald sphere and integrated intensities over a large range of deviation  
152 parameter  $S$  around the Bragg orientation are collected (compare Fig. 7). In PED, the incident  
153 beam is never directed along the zone-axis so that dynamical interactions are reduced.  
154 Microdiffraction (MD) and selected-area electron diffraction (SAED) patterns have been  
155 acquired with optical axis aligned parallel to the [001] zone axis of the crystal. MD patterns  
156 have been obtained using a nearly parallel probe of about 10-40 nm produced by a 10  $\mu\text{m}$   
157 condensor aperture. SAED patterns have been obtained using a defocused parallel beam and a  
158 circular aperture selecting an illuminated area of about 250 nm in diameter. Precession angles  
159 1.6° (heat-treated sample only), 2.4° and 2.8° have been used in order to test the sensitivity of  
160 the method to the precession angle.

161

162

## ANALYSIS OF PED DATA

163

164 **Dynamical calculations of intensities**

165 In a first approach, dynamical diffracted intensities have been calculated in the Bloch-  
166 wave formalism using the JEMS software by P. Stadelmann (2004). Then, for systematic  
167 comparison of simulated data with experimental ones and search for the best agreement, an  
168 auxiliary program also using the Bloch-wave approach has been used. The full description of  
169 the program and simulation conditions is presented in a dedicated paper (Palatinus *et al.*  
170 2013). Basically, the simulation of diffracted intensities is obtained as an incoherent  
171 summation of intensities sequentially calculated for a number  $N_{\text{or}}$  of orientations of the  
172 incident beam along the precession circuit.  $N_{\text{or}}$  is an important parameter of the simulation:  
173 the larger is  $N_{\text{or}}$ , the more accurate is the result. A few tests have been performed probing the  
174 sensitivity of the simulated intensities on the choice of  $N_{\text{or}}$ . These tests showed that fixing  $N_{\text{or}}$   
175 to 150 is appropriate, as no improvement of the match could be obtained with larger  $N_{\text{or}}$ . For a  
176 given structure file (see next paragraph), other main simulation parameters are the sample  
177 thickness  $t$ , the orientation of the precession hollow cone axis with respect to the crystal  
178 lattice and the number of diffracted beams to be taken into account for convergence of results.  
179 In this preliminary work, no refinement of the beam orientation with respect to the crystal  
180 orientation has been performed. We thus assumed that the crystal zone-axis used for the  
181 diffraction pattern collection was perfectly parallel to the precession cone axis (normally  
182 aligned along the optical axis of the microscope). This is generally not exactly fulfilled  
183 experimentally, but we will see in the results section that this approximation leads to  
184 reasonably accurate results provided the precession angle is sufficiently large. The number of  
185 beams included in the calculation is described by two parameters, the maximum length of the  
186 diffraction vectors  $g^{\text{max}}$  (in  $\text{\AA}^{-1}$ ) and the maximum excitation error  $S_g^{\text{max}}$ . Following our  
187 preliminary tests and results from Palatinus *et al.* (2013), the values of  $g^{\text{max}}$  and  $S_g^{\text{max}}$  have  
188 been fixed to  $2.0 \text{\AA}^{-1}$  and  $0.02 \text{\AA}^{-1}$ , respectively, leading to a good compromise between  
189 computation time and accuracy.

190

191 **Comparison of simulated and experimental data**

192 For comparison with simulated data, integrated intensities were extracted from  
 193 experimental zone-axis patterns using the program PETS (Palatinus 2011, Palatinus et al.  
 194 2013). The output of the program consists of the list of reflections with their indices,  
 195 intensities and estimated standard deviations of the intensities  $\sigma(I)$  calculated using the  
 196 standard background-signal-background method. Intensities were extracted up to  $g^{\max} = 1.4$   
 197  $\text{\AA}^{-1}$ . Typical values of the number of observed reflections ( $I > 3\sigma$ ) are about 400, for about  
 198 500 total reflections. All the PED hkl files used in the present work are available as  
 199 supplementary material<sup>1</sup>. The experimental data sets were then compared with several sets of  
 200 simulated intensities calculated from the OPX structure with variable Fe molar fraction  
 201  $X_{\text{Fe}}(\text{M1})$  and  $X_{\text{Fe}}(\text{M2})$  on the M1 and M2 sites. Mg content is given by  $X_{\text{Mg}}(\text{M1})=1-X_{\text{Fe}}(\text{M1})$   
 202 and  $X_{\text{Mg}}(\text{M2})=1-X_{\text{Fe}}(\text{M2})$  as required by the pyroxene stoichiometry  $(\text{Mg}_{2-x}\text{Fe}_x)\text{Si}_2\text{O}_6$ , with  $x$   
 203  $= X_{\text{Fe}}(\text{M1}) + X_{\text{Fe}}(\text{M2})$  the total Fe content (considering that minor elements are not taken into  
 204 account for this study; they account for  $< 0.04$  apfu, i.e.  $< 2\%$ ). All other structural  
 205 parameters are kept equal to the values deduced from XRD analysis. No variation of the cell  
 206 parameters as a function of the order parameter has been considered since this effect is  
 207 negligible (Tarantino et al. 2002). The structure parameters used for the calculations are  
 208 given in Table 1 for both the untreated and heat-treated crystals.

209 The present method is not a refinement method based on a least square procedure but  
 210 rather a grid search method. The best match between experimental and simulated intensities is  
 211 assessed by the lowest value of the weighted residual value  $wR2$  given by:

212

$$wR2 = \sqrt{\frac{\sum w_g (I_g^o - I_g^c)^2}{\sum w_g (I_g^o)^2}}$$

---

<sup>1</sup> Deposit items are available via the MSA web site at <http://www.minsocam.org>.

213 where  $I_g^o$  and  $I_g^c$  are the observed and calculated intensities,  $w_g = \sigma^{-2} (I_g^o)$  and the summations  
214 run over all reflections from the experimental data set.

215

216

## RESULTS

217

### XRD structure refinements

218

219

220

221 Table 2 reports the atomic fractions of Mg and  $\text{Fe}^{2+}$  at the M1 and M2 sites and the  
222 degree of order expressed as  $Q = X_{\text{Fe}}(\text{M2}) - X_{\text{Fe}}(\text{M1})$  of the untreated and heat-treated  
223 crystals, together with the refinement parameters results. Results obtained limiting the  
224 resolution to that of PED data ( $g^{\text{max}} = 1.4 \text{ \AA}^{-1}$ ) overlap within their error bars with those  
225 obtained with the full set of XRD data. The four crystallographic data of both crystals have  
226 been deposited<sup>1</sup>

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230

231 The untreated crystals are characterized by a high degree of  $\text{Fe}^{2+}$ -Mg order on the  
232 octahedral sites M1 and M2 of the OPX structure, with M2 sites mainly occupied by larger  
233  $\text{Fe}^{2+}$  cations. This ordered state is characteristic for slow cooling rate and low closure  
234 temperature of the diffusion process (around 200°C) associated with the metamorphic origin  
235 of the parent rocks (Tribaudino and Talarico 1992).

231

232

233

234 For the heat-treated samples, structure refinement results confirmed that the structure  
235 was disordered, with a higher degree of mixing of the  $\text{Fe}^{2+}$  on both M1 and M2 sites.

234

### TEM samples description

235

236

237

238 At the TEM scale, the untreated sample exhibits a homogenous microstructure made  
239 of OPX containing a few planar defects and dislocations (Fig. 1a). The heat-treated sample  
240 shows evidence of incongruent melting located at the very surface of the sample, leading to a

238 mixture of melt SiO<sub>2</sub> and Fe-rich olivine. The TEM study was performed in the lower part of  
239 the sample, for which OPX is found to be homogeneous (Fig. 1b).

240

#### 241 **Determination of cation occupancies by PED**

242 PED [001] zone-axis patterns have been acquired on both samples at several defect-  
243 free areas separated by about 0.5  $\mu\text{m}$ . Results are first presented for the heat-treated sample  
244 and then for the untreated one.

##### 245 *Heat-treated sample*

246 Selected area PED patterns have been acquired at three areas of the sample (located by  
247 circles on Fig. 1b) and for precession angles of 1.6°, 2.4° and 2.8° (Fig. 2). As described in  
248 the experimental section, the best match between experimental and simulated intensities is  
249 searched by varying three parameters: the sample thickness  $t$  and the occupancies  $X_{\text{Fe}}(\text{M1})$   
250 and  $X_{\text{Fe}}(\text{M2})$ . Results giving the best agreement (lowest  $wR2$  values) are summarized in Table  
251 3 and plotted in Fig. 3. Uncertainty of the thickness is taken as a half of the thickness step  
252 between individual simulations (3 nm). In the present work, uncertainties of the occupancies  
253 are estimated as the variations of  $X_{\text{Fe}}$  leading to 0.1% variation on the minimum  $wR2$  value.  
254 This estimation may appear as somewhat artificial, but it is directly related to the curvature of  
255 the  $wR2$  surface as a function of  $X_{\text{Fe}}(\text{M1})$  and  $X_{\text{Fe}}(\text{M2})$  (Figs. 4a and b) and so to the actual  
256 sensitivity of the method as a function of the experimental parameters (mainly the precession  
257 angle). A more rigorous treatment based on a statistical analysis of the data as described in  
258 Palatinus *et al.* (2013) leads to the same range of values for the uncertainties.

259 Results obtained with precession angle 1.6° are inconsistent with those obtained with  
260 2.4 and 2.8°, leading to quite different  $X_{\text{Fe}}(\text{M1})$  and  $X_{\text{Fe}}(\text{M2})$  values (Fig. 3). A higher  
261 dispersion of the results is also observed with precession angle 1.6° together with larger  
262 estimated errors of the three parameters. The larger errors are associated with the shape of the

263  $wR2$  surface at  $1.6^\circ$  precession angle, which is much flatter than those obtained at  $2.4^\circ$  and  
264  $2.8^\circ$  (Fig. 4), making  $wR2$  less sensitive to  $X_{Fe}(M1)$  and  $X_{Fe}(M2)$ .

265 Results obtained with  $2.4^\circ$  and  $2.8^\circ$  data sets overlap within their standard deviation  
266 for  $X_{Fe}(M1)$  and  $X_{Fe}(M2)$  whatever the observed area and the precession angle. The resulting  
267 mean values and dispersions are  $X_{Fe}(M1) = 0.144 \pm 0.008$  and  $X_{Fe}(M2) = 0.447 \pm 0.010$ .  
268 These values are consistent with those derived from XRD data ( $X_{Fe}(M1) = 0.155(2)$  and  
269  $X_{Fe}(M2) = 0.438(2)$ ). The thicknesses as deduced from data sets with precession angles  $2.4^\circ$   
270 and  $2.8^\circ$  are also consistent, giving  $t = 49 \pm 1.5$  nm for area 1,  $t = 43 \pm 1.5$  for area 2 and  $t =$   
271  $47.5 \pm 1.5$  nm for area 3.

272

### 273 *Untreated sample*

274 On the untreated sample, analysis has been performed using precession angles  $2.4^\circ$   
275 and  $2.8^\circ$  for three areas of the TEM specimen. Results are summarized in Table 4 and plotted  
276 on Fig. 5. Note that for the first area, 5 data sets are available: 3 of them have been acquired  
277 using microdiffraction (oplt1Ap24, oplt1Ap28 and oplt1Bp28) and the remaining using  
278 selected area diffraction. For areas 2 and 3, all the data have been acquired using selected area  
279 diffraction.

280 Slight discrepancies are obtained at the three areas as a function of the precession  
281 angles. Discrepancies are also observed between microdiffraction and selected area data sets  
282 taken on area 1 with  $2.4^\circ$  precession angle (compare oplt1Ap2.4 (microdiffraction) and  
283 oplt1Bp2.4 (selected area)). Results are more consistent using  $2.8^\circ$  precession angle (compare  
284 oplt1Ap2.8 and oplt1Bp2.8 (microdiffraction) with oplt1Cp2.8 (selected area)). Overall, a  
285 larger dispersion of the results is observed compared to the heat-treated sample, leading to  
286 mean values and dispersions  $X_{Fe}(M1) = 0.069 \pm 0.016$  and  $X_{Fe}(M2) = 0.551 \pm 0.028$ . Despite

287 the small discrepancy between  $X_{\text{Fe}}(\text{M1})$  obtained with PED and with XRD ( $X_{\text{Fe}}(\text{M1}) =$   
288  $0.029(2)$  and  $X_{\text{Fe}}(\text{M2}) = 0.554(2)$ ), the agreement is once again satisfactory.

289

290

## DISCUSSION

291

292 It follows clearly from Figs. 3 and 5 that the present method enables the distinction of  
293 the OPX samples as a function of their ordering state. This distinction is emphasized in Fig. 6,  
294 where all the data have been plotted together (only the inconsistent data with precession angle  
295  $1.6^\circ$  have been removed). Furthermore, values obtained for site occupancies are globally  
296 consistent with those obtained using XRD at the millimeter scale. To our knowledge, this is  
297 the first successful demonstration that site occupancies can be determined quantitatively at  
298 submicron scale using precession electron diffraction. Even if the dispersion of the results is  
299 still high compared to that obtained using XRD and has to be lowered for quantitative  
300 exploitation, this result opens the door to a wide range of applications in the field of the study  
301 of minerals at the sub-micron scale and their potential use as geothermometers and  
302 speedometers. In this section, we discuss the influence of experimental parameters on the  
303 accuracy of the results, namely the precession angle and the initial orientation of the sample.  
304 Possible structural heterogeneity in the natural sample is then inferred.

305

### *Influence of the precession angle and of the sample orientation*

306  
307 Two points require detailed discussion. First, concerning the heat-treated sample  
308 (Table 3 and Fig. 3), results obtained with precession angle  $1.6^\circ$  are inconsistent with those  
309 obtained with  $2.4^\circ$  and  $2.8^\circ$  and should be discarded. Second, results obtained on the various  
310 areas of the natural sample with  $2.4^\circ$  and  $2.8^\circ$  precession angles do not strictly overlap within  
311 their uncertainties (Table 4 and Fig. 5). These points suggest that the occupancy

312 determination could depend on the precession angle. However, calculation of diffracted  
313 intensities for comparison with experimental data takes into account the value of the  
314 precession angle and results should therefore not depend on it. Nevertheless, as described in  
315 the section on data analysis, another important experimental parameter has not been taken into  
316 account in the simulations of the PED intensities, namely the accurate orientation of the  
317 sample with respect to the precession hollow cone axis. Let us call  $\Theta$  the value of the angle  
318 between the steady incident beam direction and the crystal zone axis (Fig. 7). When  $\Theta = 0$   
319 (Fig. 7a), the on-axis orientation is perfect and for each diffraction vector, the excitation error  
320  $S_g$  is equal to that of the opposite vector  $S_{-g}$ . When  $\Theta \neq 0$  (Fig. 7b),  $S_g \neq S_{-g}$  and consequently  
321  $I_g \neq I_{-g}$ . It is one of the principal advantages of the precession method to suppress the  
322 influence of the sample misorientation by acquiring the integrated value of intensities  $I_g^{int}$   
323 (Fig. 7c) instead of a particular value  $I_g(S_g)$  as in the steady beam configuration. To fully  
324 exploit this advantage, the precession angle  $\phi$  should be high with respect to  $\Theta$ , otherwise  
325 integration of the intensities is not complete and still depends on the orientation of the sample.  
326 This is particularly true for the intense reflections close to the center of the diffraction pattern  
327 (small  $\mathbf{g}$  vectors). It is thus likely that the dispersion of the results as a function of the  
328 precession angle occurs due to the imperfect alignment of the zone axis with respect to the  
329 non-precessed electron beam, which is indeed not exactly known and difficult to quantify for  
330 a given data set. This effect is most important for low precession angles, since the integration  
331 of the intensities is then only partial. As a matter of fact, results obtained with the same data  
332 sets but including beam orientation refinement (Palatinus *et al.* 2013) reveal a lower  
333 sensitivity of the refined occupancies to the precession angle, and thus support the present  
334 interpretation. Therefore, in a first approach, we suggest using high precession angles (larger  
335 than  $2^\circ$ ) for the data collection, and orienting the crystal very carefully. The residual effect of

336 misalignment should then be very small. Repeating the experiment several times on the same  
337 area is a further means of improving the accuracy.

338

339 *Untreated sample heterogeneity*

340 For the untreated sample, there is a systematic discrepancy between the  $X_{\text{Fe}}(\text{M1})$  and  
341  $X_{\text{Fe}}(\text{M2})$  values deduced from PED and XRD. Furthermore, independently of the inaccuracy  
342 of the sample orientation, results on  $X_{\text{Fe}}(\text{M1})$  and  $X_{\text{Fe}}(\text{M2})$  are more dispersed for the  
343 untreated sample than for the heat-treated one (Fig. 6). All parameters for PED data  
344 acquisition and analysis being equivalent for both samples (except for the actual beam  
345 orientation, *cf* the previous section), this strongly suggests an influence of the samples  
346 themselves. Indeed, the heat-treated (disordered) sample has been thermally homogenized at  
347 high temperature, whereas no treatment has been made on the natural sample (ordered). The  
348 untreated sample may thus present local composition or ordering heterogeneities. Such  
349 structural heterogeneities may explain both the larger dispersion of the PED results and the  
350 discrepancy between XRD and PED results obtained on this sample.

351 The heterogeneity of the untreated sample is highlighted when plotting the line of  
352 constant composition in the graph of  $X_{\text{Fe}}(\text{M2})$  as a function of  $X_{\text{Fe}}(\text{M1})$  (Fig. 6). This line is  
353 obtained using the relation  $X_{\text{Fe}}(\text{M2}) = 2(1-y) - X_{\text{Fe}}(\text{M1})$ , where  $y$  is the ratio  $\text{Mg}/(\text{Fe}+\text{Mg})$ .  
354 Obviously, the dispersion of the results around the line drawn for  $y = 0.70$  (as given by the  
355 electron microprobe analysis at the grain scale) is much more pronounced for the untreated  
356 sample than for the heat-treated one. At this point, two types of dispersion should be  
357 distinguished: dispersion along the line corresponds to the variation of site occupancies (order  
358 parameter) at constant composition, whereas results deviating from the line correspond to  
359 compositional variations. In the case of the untreated sample, both kinds of dispersion are  
360 present, suggesting order parameter variation as well as composition variation along the

361 sample at a submicronic scale. The maximum Mg/(Mg+Fe) variation deduced from our  
362 analysis is around 4% (see Table 5). While order parameter variation involving short-range  
363 diffusion processes is plausible at this scale, composition variation is more unlikely. EDX  
364 composition profile acquired across the studied areas revealed no composition fluctuation  
365 higher than the sensitivity of the EDX method, i.e. around 2% on the Mg/(Fe+Mg) ratio. This  
366 suggests that data sets resulting in a too high deviation (superior to 2%) from the constant  
367 composition line are probably influenced by the imperfection of the model, especially by  
368 neglecting the variation of the sample orientation. This is confirmed by results obtained using  
369 orientation refinement (Palatinus et al. 2013), which are mainly dispersed along the constant  
370 composition line, corresponding to ordering variations at a microscopic scale.

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## CONCLUSIONS

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374 To our knowledge, this work on the structural ordering in orthopyroxene is the first  
375 demonstration of a quantitative determination of site occupancies at submicron scale using  
376 precession electron diffraction.. Even if quantitative exploitation of the results for deciphering  
377 thermal history of the sample is still doubtful due to the high dispersion of the results,  
378 precision is largely sufficient to distinguish between a natural metamorphic OPX ordered  
379 structure from a disordered one obtained after annealing at high temperature and rapid  
380 quenching. The method should be sensitive enough to characterize even possible intermediate  
381 states of ordering.

382

383 There are other minerals in which the cationic distribution on non-equivalent sites  
384 depends on the cooling rate and closure temperature. This is for instance the case of  
385 clinopyroxene, for which equilibrium and kinetics of the disordering process has been already  
well studied by single-crystal XRD for augitic compositions (Brizi et al. 2000; 2001) and for

386 low-Ca pigeonitic compositions (Pasqual et al. 2000; Domeneghetti et al. 2005; Alvaro et al.  
387 2011). Along with orthopyroxenes, the latter are thus considered as potential  
388 geospeedometers. However, microtextural features present in many pyroxenes must be taken  
389 into account when dealing with accurate determination of cation distributions by XRD. For  
390 instance, orthopyroxenes and clinopyroxenes commonly show exsolution phenomena; in  
391 some favorable cases these can be assessed properly and the presence of exsolution products  
392 can be corrected for (Domeneghetti et al. 1996). Unfortunately, this has not been possible for  
393 pigeonite crystals bearing augite exsolutions, which is by far the most common case for  
394 pigeonite samples. This situation prevents the use of ordering processes in pigeonite as  
395 geospeedometer for calculating cooling rates in meteorites. The use of PED thus opens an  
396 immense field of application of geospeedometry using pigeonites, and may shed light on  
397 many complicated cooling histories of terrestrial rocks or of planetary bodies.

398

399

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401

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566 **Table and Figure captions**

567

568 **Table 1:** Parameters of the orthopyroxene structures used for the dynamical calculation of  
569 intensities.

570

571 **Table 2:** Refined molar fractions of Fe and Mg on M1 and M2 sites of the orthopyroxene  
572 structure as deduced from XRD.

573

574 **Table 3:** Refinement results for PED data sets obtained on the heat-treated sample. Labels  
575 include the area location (1, 2 or 3) and the precession angle. All data sets were collected  
576 using selected area diffraction.

577

578 **Table 4:** Refinement results for PED data sets obtained on the untreated sample. Labels  
579 include the area location (1, 2 or 3) and the precession angle. All data sets were collected  
580 using selected area electron diffraction except oplt1Ap2.4, oplt1Ap2.8 and oplt1Bp2.8, which  
581 correspond to microdiffraction.

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585 **Figure 1:** TEM images of (a) the natural sample and (b) the annealed sample. Circles indicate  
586 the analyzed areas (1, 2 and 3).

587

588 **Figure 2:** [001] zone-axis PED pattern (precession angle  $2.8^\circ$ ) obtained on the heat-treated  
589 sample. The dashed circle corresponds to the resolution limit  $g^{\max} = 1.4 \text{ \AA}^{-1}$  for data  
590 extraction.

591

592 **Figure 3:** Plot of  $X_{\text{Fe}}(\text{M2})$  versus  $X_{\text{Fe}}(\text{M1})$  for the heat-treated sample. Squares: precession  
593 angle  $1.6^\circ$ , triangles:  $2.4^\circ$  and circles:  $2.8^\circ$ . Colors correspond to studied areas on the TEM  
594 sample (see Fig. 1) (red: area 1, green: area 2, blue: area 3). The black star corresponds to  
595 XRD data as obtained on the single crystal (error bars ca. size of the symbol).

596

597 **Figure 4:** Plot of  $wR2$  as a function of  $X_{\text{Fe}}(\text{M1})$  and  $X_{\text{Fe}}(\text{M2})$  for the heat-treated sample. a)  
598 Precession angle  $2.8^\circ$ , area 3. b) Precession angle  $1.6^\circ$ , area 2.

599

600 **Figure 5:** Plot of  $X_{\text{Fe}}(\text{M2})$  versus  $X_{\text{Fe}}(\text{M1})$  for the untreated sample. Triangles: precession  
601 angle  $2.4^\circ$  and circles: precession angle  $2.8^\circ$ . Colors correspond to studied areas on the TEM  
602 sample (see Fig. 1) (red: area 1, green: area 2, blue: area 3). The black star corresponds to  
603 XRD data as obtained as obtained on the single crystal (error bars ca. size of the symbol).

604

605 **Figure 6:** Plot of  $X_{\text{Fe}}(\text{M2})$  versus  $X_{\text{Fe}}(\text{M1})$  for the untreated (blue) and heat-treated (red)  
606 samples. Black stars correspond to XRD data (error bars ca. size of the symbol). The dashed-  
607 line corresponds to the constant composition line with  $\text{Mg}/(\text{Mg}+\text{Fe}) = 0.70$ .

608

609 **Figure 7:** Sketch of the variation of the intensity of a  $\mathbf{g}$  diffraction vector as a function of the  
610 orientation of the incident beam.  $\mathbf{S}_g$  is the vector pointing from the reciprocal lattice node to  
611 the Ewald sphere.  $\mathbf{S}_g$  is positive when oriented along the beam direction and negative  
612 elsewhere. (a) The incident beam is perfectly aligned along the zone axis and  $\mathbf{S}_g = \mathbf{S}_{\cdot g}$ . (b) The  
613 incident beam is tilted with an angle  $\phi$  from the zone-axis orientation. Then  $\mathbf{S}_g \neq \mathbf{S}_{\cdot g}$ . (c)  
614 When the beam is rotated, the intensities are integrated along the S values. For the integration  
615 to be sufficiently complete, the precession angle has to be high enough.  
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622**Table 1****Untreated crystal**

Space group			<i>P b c a</i>			
Lattice parameters (Å)			alpha	beta	gamma	
<i>a</i>	<i>b</i>	<i>c</i>	90	90	90	
18.2810	8.8732	5.2070	90	90	90	
Unit-cell volume (Å <sup>3</sup> )			844.6			
Structure parameters :						
Atom type	x/a	y/b	z/c	Occupancy	U (Å <sup>2</sup> )	Site multiplicity
1 Si	0.27154	0.34091	0.05149	1	0.007	8c
2 Si	0.47389	0.33680	0.79705	1	0.007	8c
3 O	0.18347	0.33841	0.04145	1	0.008	8c
4 O	0.56252	0.33763	0.79837	1	0.008	8c
5 O	0.31119	0.50097	0.04912	1	0.010	8c
6 O	0.43369	0.48496	0.69471	1	0.010	8c
7 O	0.30269	0.22822	-0.17331	1	0.010	8c
8 O	0.44726	0.19933	0.59739	1	0.010	8c
9 Mg (M1)	0.37558	0.65446	0.87127	1-X <sub>Fe</sub> (M1)	0.008	8c
10 Fe (M1)	0.37558	0.65446	0.87127	X <sub>Fe</sub> (M1)	0.008	8c
11 Mg (M2)	0.37803	0.48345	0.36509	1-X <sub>Fe</sub> (M2)	0.010	8c
12 Fe (M2)	0.37803	0.48345	0.36509	X <sub>Fe</sub> (M2)	0.010	8c

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630**Heat-treated crystal**

Space group			<i>P b c a</i>			
Lattice parameters (Å)			alpha	beta	gamma	
<i>a</i>	<i>b</i>	<i>c</i>	90	90	90	
18.3022	8.8816	5.2082	90	90	90	
Unit-cell volume (Å <sup>3</sup> )			846.6			
Structure parameters :						
Atom type	x/a	y/b	z/c	Occupancy	U (Å <sup>2</sup> )	Site multiplicity
1 Si	0.27165	0.34084	0.05070	1	0.007	8c
2 Si	0.47378	0.33678	0.79655	1	0.007	8c
3 O	0.18360	0.33902	0.03944	1	0.009	8c
4 O	0.56235	0.33842	0.79762	1	0.009	8c
5 O	0.31119	0.50097	0.04912	1	0.010	8c
6 O	0.43342	0.48423	0.69311	1	0.010	8c
7 O	0.30274	0.22748	-0.17301	1	0.011	8c
8 O	0.44740	0.19873	0.59763	1	0.011	8c
9 Mg (M1)	0.37572	0.65432	0.87013	1-X <sub>Fe</sub> (M1)	0.008	8c
10 Fe (M1)	0.37572	0.65432	0.87013	X <sub>Fe</sub> (M1)	0.008	8c
11 Mg (M2)	0.37783	0.48402	0.36400	1-X <sub>Fe</sub> (M2)	0.010	8c
12 Fe (M2)	0.37783	0.48402	0.36400	X <sub>Fe</sub> (M2)	0.010	8c

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635 **Table 2**

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	Untreated crystal		Heat treated crystal	
	Full data Set	Low Res. 0.714 Å	Full data set	Low Res. 0.714 Å
X <sub>Fe</sub> (M1)	0.029(2)	0.028(3)	0.155(2)	0.154(3)
X <sub>Mg</sub> (M1)	0.971(2)	0.972(3)	0.845(2)	0.846(3)
X <sub>Fe</sub> (M2)	0.554(2)	0.555(3)	0.438(2)	0.439(3)
X <sub>Mg</sub> (M2)	0.446(2)	0.445(3)	0.562(2)	0.561(3)
Mg/(Fe+Mg)	0.709(3)	0.709(3)	0.704(3)	0.704(3)
$Q = X_{Fe}(M2) - X_{Fe}(M1)$	0.525(3)	0.527(3)	0.283(3)	0.285(3)
R <sub>i</sub> (%)	2.73	2.41	3.07	2.66
wR2	6.64	5.99	7.41	6.81
n. of $I/\sigma > 4$	2090	1187	2039	1172
n. relf. tot.	2209	1209	2219	1215
ref. param.	93	93	93	93
Goof	1.199	1.172	1.144	1.178

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641 **Table 3**

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dataset	wR2	t (nm)	$\Delta(t)$	$X_{\text{Fe}}(\text{M1})$	$\Delta(X_{\text{Fe}}(\text{M1}))$	$X_{\text{Fe}}(\text{M2})$	$\Delta(X_{\text{Fe}}(\text{M2}))$	$Q^*$	$\Delta Q$	Mg/(Fe+Mg)
precession angle: 1.6°										
opht1p1.6	12.88	52	1.5	0.200	0.015	0.420	0.015	0.220	0.015	0.69
opht2p1.6	13.18	46	1.5	0.235	0.015	0.415	0.015	0.180	0.015	0.68
opht3p1.6	20.58	49	1.5	0.280	0.015	0.420	0.015	0.140	0.015	0.65
precession angle: 2.4°										
opht1p2.4	6.77	49	1.5	0.147	0.010	0.430	0.010	0.283	0.015	0.71
opht2p2.4	9.53	43	1.5	0.134	0.010	0.441	0.010	0.307	0.015	0.71
opht3p2.4	12.30	49	1.5	0.140	0.010	0.452	0.010	0.312	0.015	0.70
precession angle: 2.8°										
opht1p2.8	7.38	49	1.5	0.154	0.010	0.442	0.010	0.288	0.020	0.70
opht2p2.8	8.66	43	1.5	0.134	0.010	0.458	0.010	0.324	0.020	0.70
opht3p2.8	11.04	46	1.5	0.154	0.010	0.458	0.010	0.304	0.020	0.69

644

645  $*Q = X_{\text{Fe}}(\text{M2}) - X_{\text{Fe}}(\text{M1})$ 

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649650 **Table 4**

651

652

dataset	wR2	t (nm)	$\Delta(t)$	$X_{\text{Fe}}(\text{M1})$	$\Delta(X_{\text{Fe}}(\text{M1}))$	$X_{\text{Fe}}(\text{M2})$	$\Delta(X_{\text{Fe}}(\text{M2}))$	$Q^*$	$\Delta Q$	Mg/(Fe+Mg)
precession angle: 2.4°										
oplt1Ap2.4 <sup>1</sup>	12.80	43	1.5	0.030	0.010	0.582	0.010	0.552	0.015	0.69
oplt1Bp2.4	9.42	40	1.5	0.067	0.010	0.544	0.010	0.447	0.015	0.69
oplt2p2.4	7.39	40	1.5	0.092	0.010	0.502	0.010	0.410	0.015	0.70
oplt3p2.4	17.89	40	1.5	0.081	0.010	0.572	0.010	0.491	0.015	0.67
precession angle: 2.8°										
oplt1Ap2.8 <sup>1</sup>	9.60	43	1.5	0.072	0.010	0.544	0.010	0.472	0.015	0.69
oplt1Bp2.8 <sup>1</sup>	9.35	40	1.5	0.072	0.010	0.535	0.010	0.463	0.015	0.70
oplt1Cp2.8	9.26	40	1.5	0.067	0.010	0.563	0.010	0.496	0.015	0.69
oplt2p2.8	15.11	40	1.5	0.072	0.010	0.595	0.010	0.523	0.015	0.67
oplt3p2.8	10.90	40	1.5	0.072	0.010	0.526	0.010	0.454	0.015	0.70

653

654 \* $Q = X_{\text{Fe}}(\text{M2}) - X_{\text{Fe}}(\text{M1})$ 655 <sup>1</sup>microdiffraction

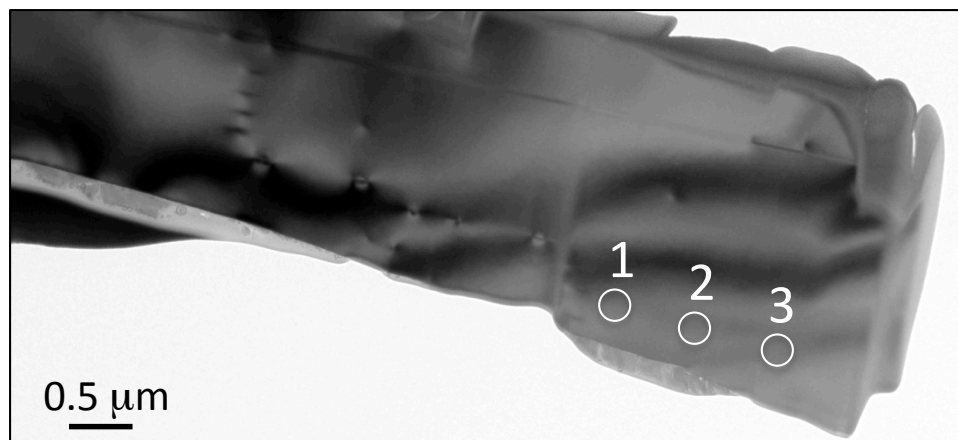
656

657

658

659

(a)



(b)

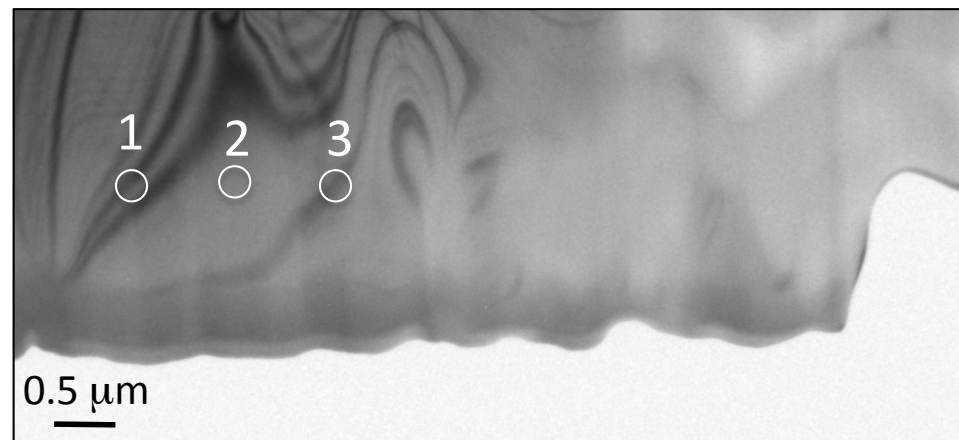


Figure 1

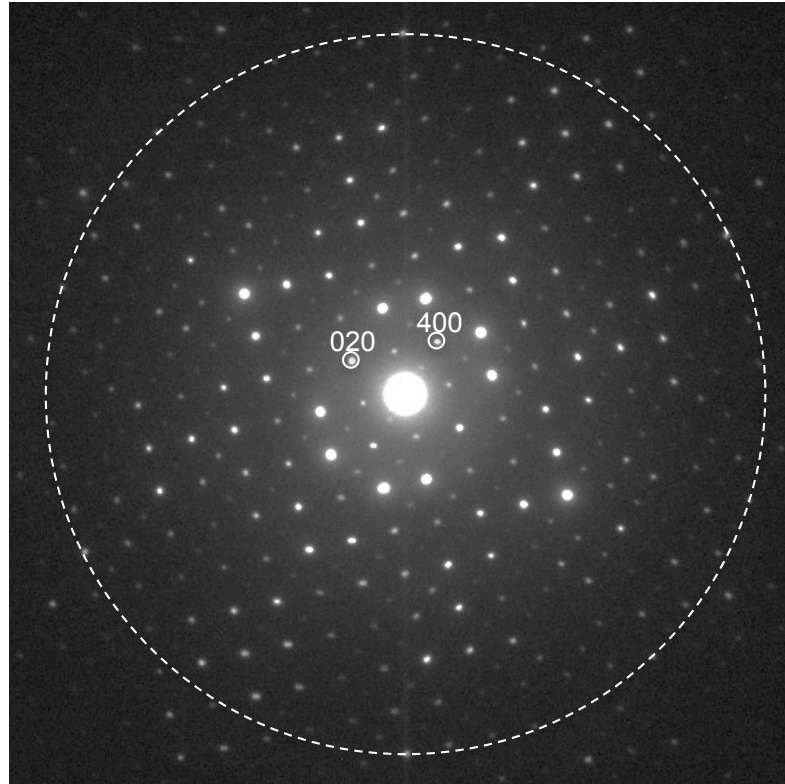


Figure 2

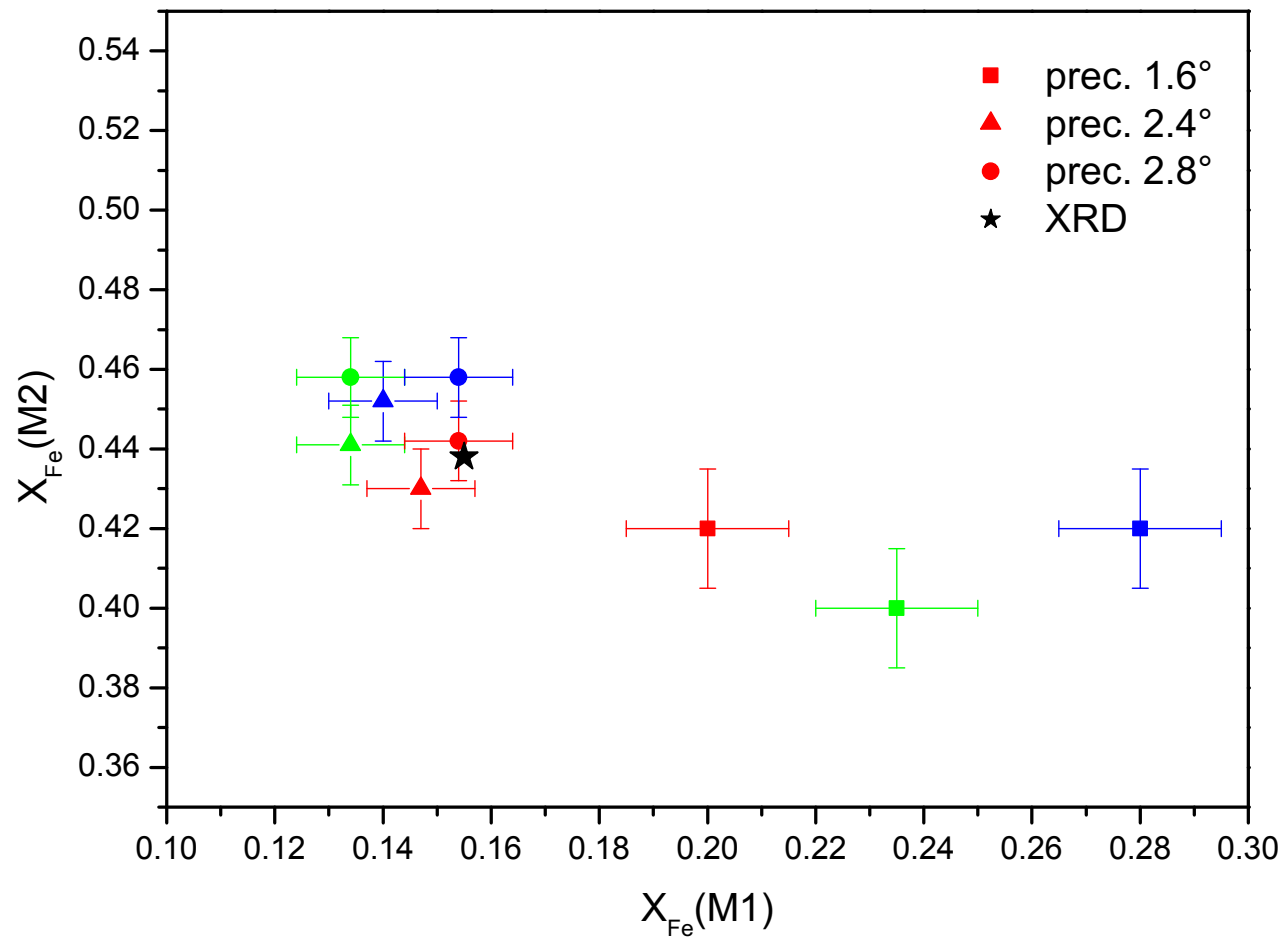
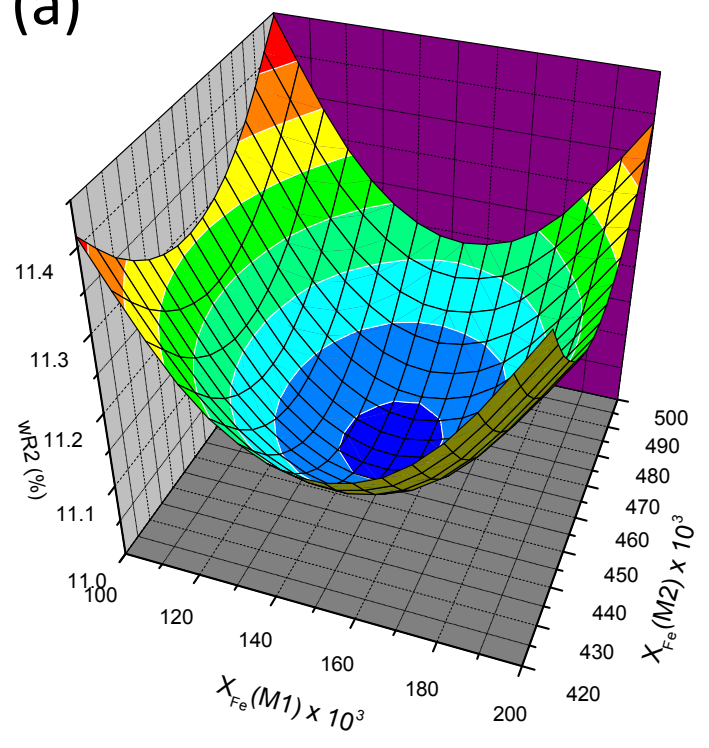


Figure 3

(a)



(b)

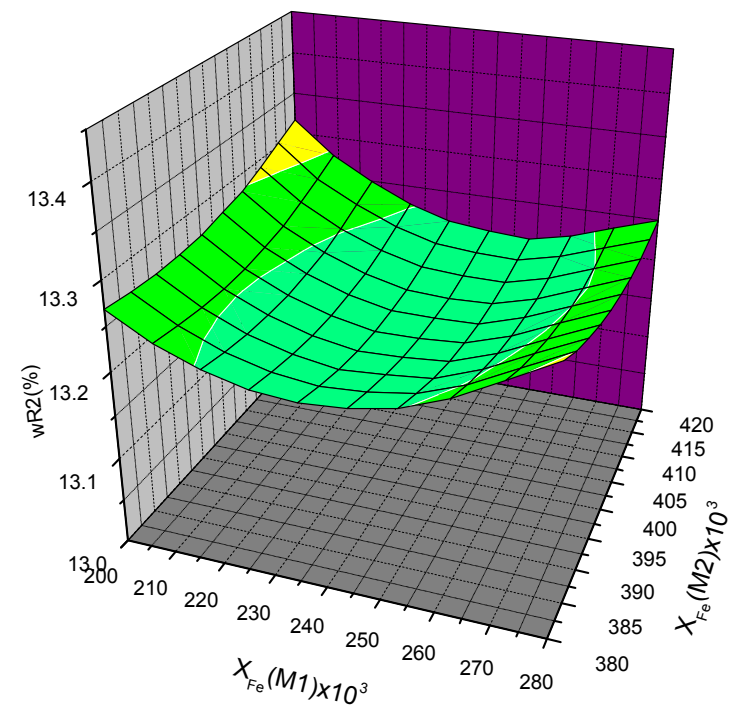


Figure 4

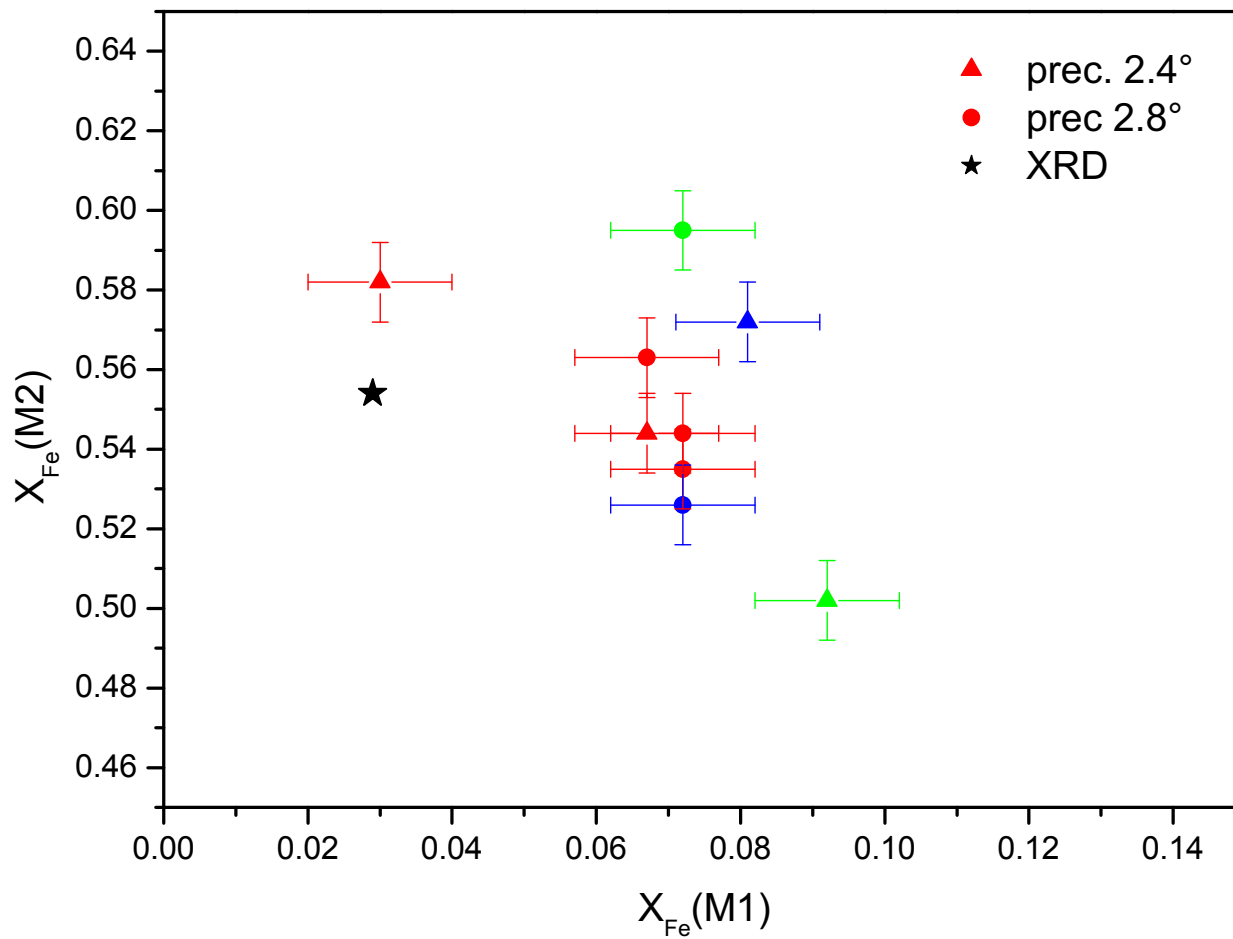


Figure 5

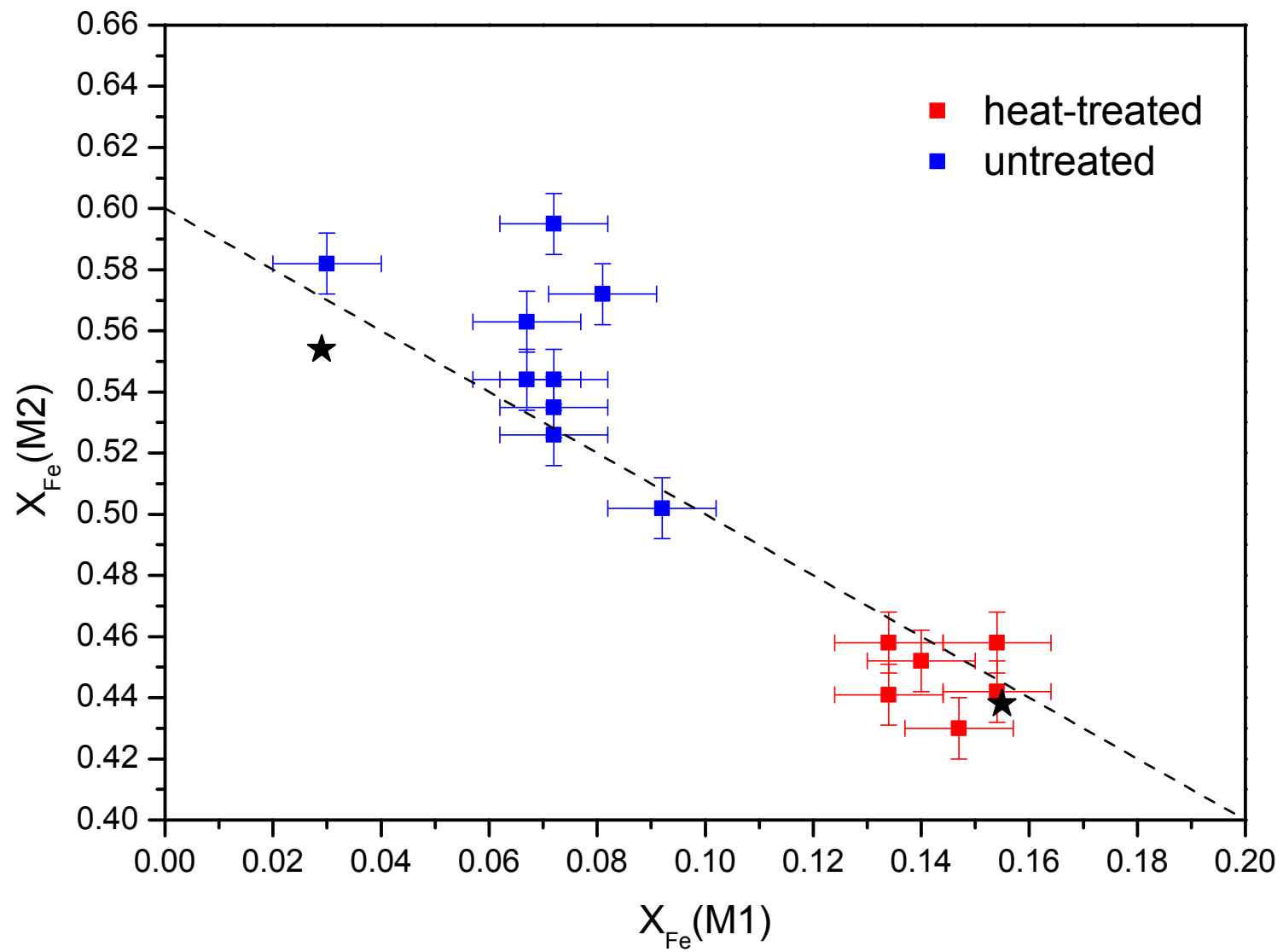


Figure 6

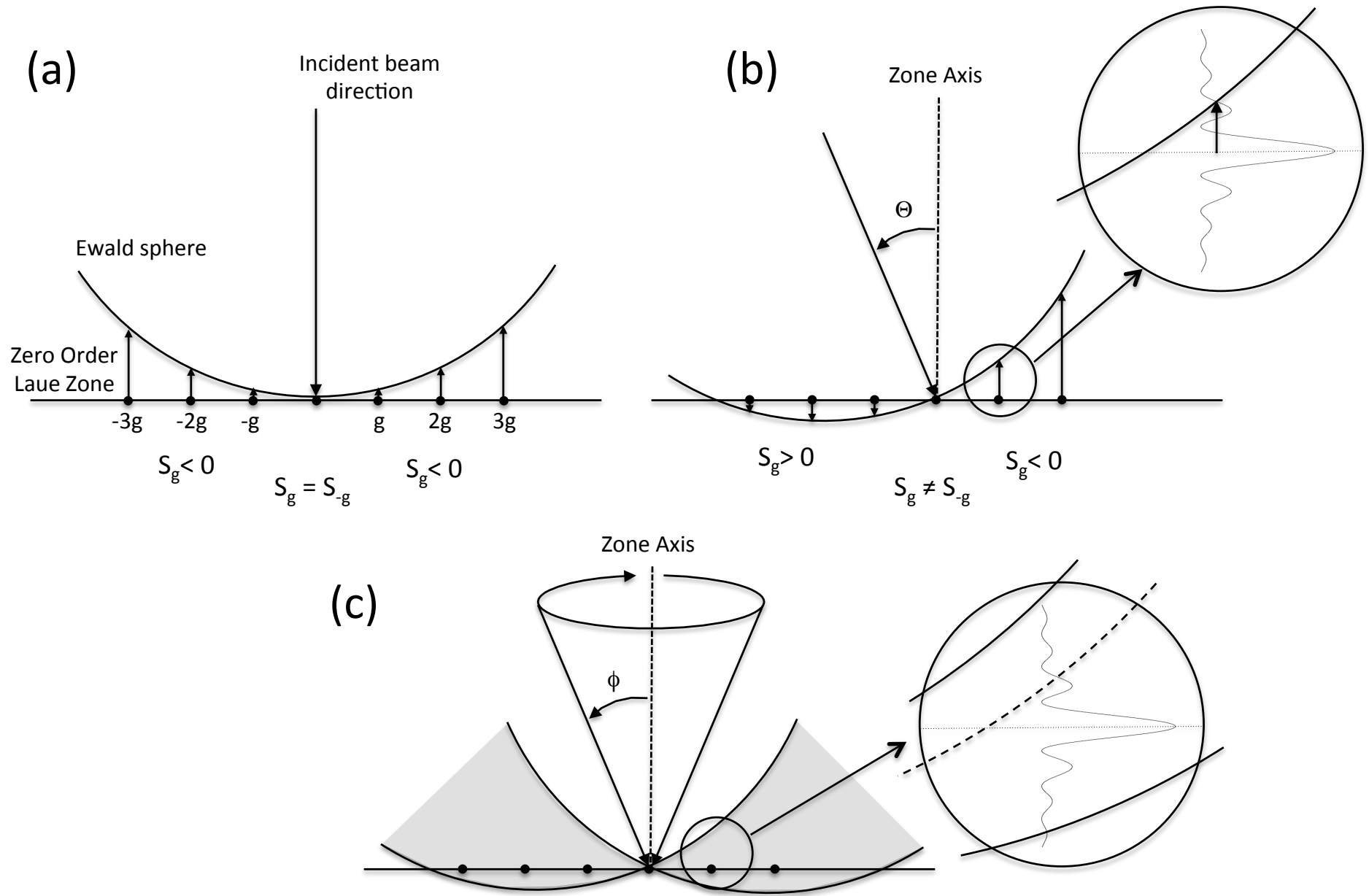


Figure 7