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#### **Identifying the Role of Terahertz Vibrations in Metal-Organic Frameworks: From Gate-Opening Phenomenon to Shear-Driven Structural Destabilization**

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## **Supplemental Material**

**for**

# **Identifying the Role of Terahertz Vibrations in Metal-Organic Frameworks: From Gate-Opening Phenomenon to Shear-Driven Structural Destabilization**

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### **1 Materials Synthesis and Characterization**

We have synthesized the ZIF-4 sample using a modified route from that reported in ref.[1], this was to ensure that a phase pure compound was obtained for this work. 1.2 g  $Zn(NO<sub>3</sub>)<sub>2</sub>$  $\cdot$  6H<sub>2</sub>O and 0.9 g imidazole were dissolved in 90 mL DMF. The mixture was transferred into a 100 mL screw jar and tightly secured. The mixture was heated to 100°C for 3 days. After cooling the ZIF-4 crystals were filtered off, then washed 3 times with  $\sim$ 20 mL DMF and repeated 3 times with 20 mL DCM before air drying. Guest-free samples were prepared by immersing the dried crystals in 100 mL DCM and stirred gently for 3 days. Every 24 hours, the material was filtered off and the DCM was exchanged with 100 mL of fresh solvent. After the final filtering step, the sample was dried overnight in a vacuum oven at 120°C.

The ZIF-7 sample (i.e. ZIF-7-I [2]) was synthesized according to procedures reported by Park *et al.* [3]. The as-synthesized ZIF-7 was dried in the vacuum oven at room temperature overnight, and without further activation step to prevent (unwanted) phase transition from ZIF-7-I to ZIF-7-II [2]. Using FTIR spectroscopy (Fig. S3), we have established that the ZIF-7 sample contains DMF (*N*,*N*-dimethylformamide) solvent molecules used for synthesis, which is key to retaining the ZIF-7-I phase. We note that, recently Zhao *et al.* [2] reported heating of ZIF-7-I in vacuum at 77°C will yield ZIF-7-II due to the removal of entrapped DMF.

ZIF-8 sample was synthesized according to procedures reported in the literature [3, 4]. The activation of ZIF-8 was performed by solvent-exchange in methanol overnight followed by evacuation for 24 hours at room temperature, and finally the sample was dried overnight in a vacuum oven at 150°C.

Prior to the inelastic neutron scattering and synchrotron far-infrared spectroscopic experiments, the purity and crystallinity of the powdered samples were confirmed using powder X-ray diffraction (PXRD) on a Bruker D8 diffractometer, using Cu K<sub>α</sub> radiation ( $λ =$ 1.5418 Å) together with a LynxEye position-sensitive detector in the Bragg-Brentano geometry. All measurements were performed at ambient condition in the 2θ range of 5° to 50°. The PXRD patterns of ZIFs-4, -7, -8 are shown below.

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Fig. S1. PXRD of ZIF-4. Simulated pattern using crystal structure deposited by Park *et al*. [3] (CCDC code: VEJYUF [5]). (b) Activated sample used in this work.



Fig. S2. PXRD of ZIF-7 sample used in this study (black curve), this diffraction pattern was collected using powders that have been subjected to inelastic neutron scattering (INS). Our ZIF-7 sample matches the simulated pattern of ZIF-7-I phase (blue), which is significantly different from two other recently reported phases [2]: ZIF-7-II (red) and ZIF-7-III (green). Simulated pattern of ZIF-7-I was generated using crystal structure

deposited by Park *et al*. [3] (CCDC code: VELVIS [5]), while those of ZIF-7-II and ZIF-7-III based on crystal structures reported by Zhao *et al.* [2]



Fig. S3. FTIR spectra of the ZIF-7 sample (retrieved after INS experiment, see Fig.S2). The presence of the carbonyl peak (amides) confirms that the pores of the framework contain DMF solvents.



Fig. S4. PXRD of ZIF-8. (a) Simulated pattern using crystal structure deposited by Moggach *et al*. [6] (CCDC code: TUDKEJ [5]). (b) Activated sample used in this work.

### **2 Inelastic Neutron Scattering (INS)**

The INS spectra were recorded using the high resolution (∆E/E ~1.25%) broadband (-24 – 4000 cm<sup>-1</sup>) TOSCA spectrometer [7-9] at the ISIS Pulsed Neutron and Muon Source of the Rutherford Appleton Laboratory (Chilton, UK). TOSCA is an indirect geometry time-of-flight spectrometer where a pulsed, polychromatic beam of neutrons illuminates the sample at a distance of 17 m from the source (room temperature water moderator).

Energy analysis was performed after the interaction of the incident neutron beam with the sample under investigation. The sample scattered neutrons were Bragg reflected by a pyrolytic graphite analyser, while higher-order reflections beyond (002) were suppressed by a cooled  $(T < 50 K)$  Beryllium (Be) filter so as to define a unique final energy. Thus neutrons with a final energy of approximately 32  $cm^{-1}$  were passed towards the  $3$ He detector bank. The detector array was comprised of a total of ten banks each having thirteen <sup>3</sup>He tubes with an effective length (thickness) of 250 (2.5) mm. Five banks were located in forward scattering (scattering angle  $\sim$ 45°) and five in backscattering ( $\sim$ 135°) direction. The use of a low (fixed) final energy translated into a direct relationship between energy transfer (E<sub>T</sub>, cm<sup>-1</sup>) and momentum transfer (Q,  $A^{-1}$ ) such that E<sub>T</sub>  $\approx 16Q^2$ . Energy transfer and spectral intensity were then calculated using standard programs to convert to the conventional *S*(Q,ω).

A disc chopper to prevent frame overlap was positioned at 8 m from the moderator. It relied on the 40 ms timeframe during the operation of both target stations at ISIS to extend the incident-wavelength bandwidth of the instrument down to  $-24$  cm<sup>-1</sup> i.e. elastic line region. The samples (2–3 g) were wrapped in  $4 \times 4$  cm aluminium foil sachets (which filled the neutron beam at the sample position) and placed in thin walled aluminium cans. To reduce the impact of the Debye–Waller factor on the observed spectral intensity, the samples i.e. the sample chamber was cooled to approximately 10 K by a closed cycle refrigerator (CCR) and the spectra were recorded for 6 to 12 hours.

TOSCA outstanding spectral resolution ( $ca$ . 1.25%  $E<sub>T</sub>$ ), arises from the combination of several factors: the narrow bandpass of the PG002/Be analyser, tight moderator pulse widths (~10  $\mu$ s/Å), a long incident flight path and a time- and energy-focused detector geometry. Quantitatively, the above design choices translate into an absolute spectral resolution of  $\sim$ 2.4 cm<sup>-1</sup> for elastic events (E<sub>T</sub> = 0 cm<sup>-1</sup>) and  $\sim$ 50 cm<sup>-1</sup> at E<sub>T</sub> = 4000 cm<sup>-1</sup>. Instrument backgrounds are negligible, thus low-cross section measurements beyond hydrogen-containing materials are not only feasible but also a growing area of research on the instrument.

## **3 Synchrotron Radiation Far-Infrared (SR FIR) Absorption Spectroscopy**

Infrared (IR) absorption spectra of all compounds were recorded at the Multimode InfraRed Imaging and Microspectroscopy (MIRIAM) Beamline B22 located in the Diamond Light Source synchrotron [10]. The great advantage of the synchrotron radiation (SR) IR source was mainly the brightness of the beam (photon flux density several order of magnitude higher than conventional sources) and broadband spectral range (covering simultaneously from the visible to the sub-THz region). The storage ring was operating in the standard user mode, i.e. 900 bunches filling pattern for a total circulating current of 300 mA and top up at every 10 minutes (lifetime circa 18 h and photon flux fluctuation limited to 0.3%). IR spectroscopy was performed in vacuum via a Bruker Vertex 80 V Fourier Transform IR (FTIR) interferometer (Bruker Optics, Ettlingen, Germany) equipped with an RT DLaDTGS detector. The measurements were performed at a resolution of 1 cm<sup>-1</sup> with a 6 µm thick Mylar broad-band multilayer beamsplitter (allowing the spectral range below 700 cm<sup>-1</sup>) at a scanner velocity of 10 kHz (with respect to the 15,800 cm<sup>-1</sup> laser reference).

The experiments were performed at room temperature (RT) using the Attenuated Total Reflection (ATR) method for the advantage of measuring samples without any preparation [11]. Specifically, the MOF powder was placed on top of a diamond ATR crystal into a Bruker Diamond ATR accessory, and the sample was held in position by pressure applied onto a HDPE disk *via* a clamp mechanism. Schematically, the synchrotron IR beam enters the ATR crystal from the bottom at an angle of circa 45° to be totally reflected at the crystal-sample interface, then re-directed *via* a circa 45° mirror to the detector pupil. Provided the sample refractive index is lower than the ATR material (diamond *n* = 2.4) only a fraction of radiation reaches into the sample (evanescent wave). At the wavelength where the sample absorbs energy the evanescent wave will be attenuated, and consequently the beam exiting is modulated like in a transmission measurement. The beam penetration depth in ATR mode depends linearly on the wavelength (for diamond and the geometry used this penetration is below 2 mm for soft materials), plus the ATR spectrum is influenced by the relative refractive index change of the sample. For these reasons the spectra shown have been corrected via Extended ATR correction algorithms in the OPUS 7.2 software used for data acquisition and subsequent analysis.

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## **4 Vibrational Modes for Spectral Region < 700 cm-1 (~21 THz)**

Table S1: Assignment of vibrational modes up to 87 meV, based on DFT calculations (§5).







### **5** *Ab Initio* **Quantum Mechanical Simulations**

## **5.1 DFT calculations of vibrational frequencies of periodic ZIF structures**

Density functional theory (DFT) calculations were carried out at the PBE [12] level of theory augmented with a damped empirical dispersion term (PBE-D) [13] and performed with the periodic *ab initio* code CRYSTAL14 [14]. Crystalline orbitals were represented as linear combinations of Bloch functions (BF), and were evaluated over a regular threedimensional (3D) mesh in reciprocal space. Each BF was constructed from local atomic orbitals (AOs), which are linear combinations (with constant coefficients) of Gaussian-type functions (GTFs). Each GTF is the result of a Gaussian multiplied by a solid spherical harmonic.

All electron TZVP basis sets were used for Zn, N, C and H atoms, consistent with our previous work on ZIF-8 [15]. The adopted basis sets contain 4352, 2736 and 2016 basis functions for ZIF-4, ZIF-7 and ZIF-8, respectively. For the exact basis sets used in this study, please see other supplemental information (filename: BasisSets-Ryder-PRL.txt).

A full relaxation of both lattice parameters and atomic coordinates was allowed. The geometry optimisation at constant symmetry was performed by means of a quasi-Newtonian algorithm in which the quadratic step (BFGS Hessian updating scheme) is combined with a linear one (parabolic fit) as proposed by Schlegel [16-20]. Convergence was tested on the root mean square (RMS) and the absolute value of the largest component of the gradients and the estimated displacements. The threshold for the maximum and RMS gradient, and the maximum and RMS atomic displacement of all atoms was set to 1.5 x 10<sup>-4</sup>, 1.0 x 10<sup>-4</sup>, 3.0 x 10<sup>-4</sup> and 2.0 x 10<sup>-4</sup> a.u., respectively. The optimisation was considered to have completed when all four conditions were simultaneously satisfied.

The mass-weighted Hessian matrix for the calculation of the vibrational frequencies was obtained by numerical differentiation of the analytical first derivatives, calculated at geometries obtained by slightly displacing in turn each of the 3N nuclear coordinates with respect to the equilibrium geometry. Vibrational frequencies were calculated at the Γ point, as the volume of each system is sufficient to expect negligible phonon dispersion.

The IR intensities were calculated through the Berry Phase approach [21], by evaluating the atomic Born tensors as polarization differences between the original and the distorted geometries. This approach assumes the polarization difference is equal to the timeintegrated transient macroscopic current that flows through the sample during the vibrations.[14, 22]

Summarized below are the comparisons of lattice parameters calculated from DFT (for ideal crystalline structures) versus experimental values reported in literature.





• Average error in lattice parameters using PBE is 1.56%. We note the error is relatively higher for PBE-D (5.57%) due to dispersion correction overestimating the interactions caused by extended aromaticity of the bulky benzimidazolate ligands. Fig. S5 shows that the predicted vibrational spectra are consistent with experiments.



#### *Notes:*

- <sup>†</sup>Crystallographic information file (CIF) deposited to CCDC [5] by ref.[3], in which singlecrystal XRD data were collected at 283(2) K.
- ‡ Experimental cell parameters of ZIF-8 determined from (i) single-crystal X-ray diffraction [6]: *a* = 16.9920(8) Å at 293 K, (ii) neutron powder diffraction [23]: *a* =16.9900(2) Å at 3.5 K and *a* = 17.0117(4) at 200 K.
- $\diamond$ Symmetry lowered in DFT optimization of ZIF-8, due to imaginary frequencies being present when using the experimentally assigned space group (293 K). The imaginary modes can be linked to the rotation of the methyl groups observed at low temperatures (see ref.[23]). To account for this effect, the symmetry of the system was relaxed to allow for a stable structure with minimum potential energy to be optimized.



Fig. S5. Comparison of far-IR spectra in the region of 25-700 cm<sup>-1</sup> between (a) DFT PBE, (b) DFT PBE-D and (c) synchrotron radiation FIR measurements.

#### **5.2 Overtones associated with higher-order vibrational excitations**

The theoretical results place the highest peaks at 3196, 3180 and 3195  $cm^{-1}$  for ZIF-4, ZIF-7 and ZIF-8 respectively. In reality, however, the experimental spectrum is more complex (see Fig.2 in main manuscript), as higher orders of vibrational excitation take place and even though the probability is far less, the collective contribution to the spectral results is significant. It is, therefore, necessary to include in the theoretical results the overtones that would be expected from these excitations, which was achieved using the vibrational eigenvectors, generated from the DFT calculations, with the aClimax code [24, 25]. The theoretical spectra include up to  $10<sup>th</sup>$  order overtones with fully anisotropic external Debye-Waller effects. This approximation of the probability of higher order excitations has allowed for the theoretical results to match exceptionally well with that of experiment. To account for the overtone peaks being shifted to higher regions, we have shown each spectra up to 4000 cm $^{\text{-}1}.$ 



Fig. S6. Comparison of DFT spectra, for ZIF-4, with and without the inclusion of higher order vibrational excitations.



Fig. S7. Comparison of DFT spectra, for ZIF-7, with and without the inclusion of higher order vibrational excitations.



Fig. S8. Comparison of DFT spectra, for ZIF-8, with and without the inclusion of higher order vibrational excitations.

## **5.3 MP2 calculations to determine harmonic frequencies of imidazole molecules**

To further pinpoint the exact nature of the vibrational modes, we calculated the theoretical spectra of the isolated gas-phase molecules that constitute the organic bridging linkers in each framework. The minimum energy structure and harmonic frequency calculations for the three individual organic linker molecules: Imidazole (Im), benzimidazole (bIm), and 2 methylimidazole (mIm), were carried out with the Gaussian 09 code [26]. The calculations were performed at the MP2 [27] level of theory with the aug-cc-pVDZ [28] basis set.

The predictions obtained are tabulated in Table S2 below, which are found to be consistent with spectroscopic measurements available in the literature, see for example [29-31]. Vibrational signatures of the isolated molecular systems enabled us to confirm that, the major vibrational modes of ZIFs are indeed those that involved a contribution from the inorganic building units, i.e.  $ZnN<sub>4</sub>$  tetrahedra. On this basis, certain modes corresponding to organic linker vibrations are being shifted to lower energy levels upon formation of 3-D extended frameworks.



Table S2: Theoretical IR-active vibrational modes of the imidazole-based molecules.



## **6 Nature of THz Vibrational Modes < 100 cm-1 (~3 THz)**

Table S3: Assignment of the THz vibrational modes obtained from DFT calculations in §5.







### **7 ZIF-4 Pore Modification Mechanisms Associated with THz Vibrations**



Fig. S9. (a) DFT optimized structure where the solvent accessible volume (SAV) is represented by the yellow surfaces. (b) Increase of SAV by 2~3% due to the 0.2 THz soft mode. (c) Coalescence of adjacent SAV at 1 THz ('gate-opening' mode) to yield further raise in the SAV.

### **8 Animations of ZIFs Lattice Dynamics**

Web links to simulations calculated using Density Functional Theory (DFT), by applying the CRYSTAL14 code [14].

#### **ZIF-4:**

FIG.4a - Soft mode at 0.2 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF4SoftModeAnimA.gif

FIG.4b - Gate-opening mechanism at ~1 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF4GateOpeningAnimB.gif

#### **ZIF-7:**

FIG.5a - Soft mode at 0.652 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF7SoftModesAnimA.gif

FIG.5b - Breathing and shearing mechanisms at 0.654 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF7BreathingShearingAnimB.gif

FIG.5c - Breathing mechanism at 1.47 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF7BreathingAnimC.gif

#### **ZIF-8:**

FIG.6a - Soft mode at 0.57 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF8SoftModeAnimA.gif

FIG.6b - Shear mechanism at 0.65 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF8ShearModeAnimB.gif

FIG.6c - Gate-opening mechanism at 1 THz http://www.eng.ox.ac.uk/tan/thz-anim/ZIF8GateOpeningAnimC.gif

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