



UNIVERSITÀ DEGLI STUDI DI TORINO

AperTO - Archivio Istituzionale Open Access dell'Università di Torino

# Cyclodextrins as a Templating Agent in Solvent-Free Kneading- Based Syntheses of Nanosized SnO2 and ZnO

 This is a pre print version of the following article:

 Original Citation:

 Availability:

 This version is available http://hdl.handle.net/2318/1874582

 since 2022-10-04T13:23:30Z

 Published version:

 DOI:10.1021/acssuschemeng.2c02344

 Terms of use:

 Open Access

 Anyone can freely access the full text of works made available as "Open Access". Works made available under a Creative Commons license can be used according to the terms and conditions of said license. Use of all other works requires consent of the right holder (author or publisher) if not exempted from copyright protection by the applicable law.

(Article begins on next page)

Cyclodextrins as templating agent in solvent-free
 kneading-based syntheses of Nanosized SnO<sub>2</sub> and
 ZnO

- 4 Alberto Rubin Pedrazzo<sup>a</sup>\*, Andrea Jouve<sup>a</sup>, Sara Morandi<sup>a</sup>, Maela Manzoli<sup>b</sup>, Claudio Cecone<sup>a</sup>,
- 5 Pierangiola Bracco<sup>a</sup> and Marco Zanetti<sup>a,c</sup>

- 7 <sup>a</sup>Department of Chemistry, University of Torino, Via P. Giuria 7, NIS and INSTM Reference
- 8 Centres University of Torino, 10125 Torino, Italy
- 9 <sup>b</sup>Department of Drug Science and Technology, NIS and INSTM Reference Centres University of
- 10 Torino, Via Giuria 9, 10125 Torino, Italy
- 11 °ICxT Centre, University of Torino, Lungo Dora Siena 100, 10153 Torino, Italy
- 12
- 13 KEYWORDS
- 14 Kneading, cyclodextrins, MOS, semiconductor metal oxides, mechanochemistry
- 15 ABSTRACT
- 16 -----

#### 17 INTRODUCTION

18 In recent years, gas sensors have been increasingly used in industrial production and daily life:

19 efficient, low cost and small-sized sensors are nowadays requested in many fields of applications

20 such as environmental, food or biomedical [REF].

In this context, in the ultra-sensible sensors field there has been a growing interest and diffusion of metal oxide-based semiconductors (MOS): MOS are suitable materials for sensors for their outstanding physical and chemical properties that combine high efficiency, fast response, stability and, last but not least, simple preparation and low cost. <sup>1,2</sup>

In gas sensing applications (and in solar cell and photo-catalysis as well) MOS activity is strongly related to the specific surface area <sup>3,4</sup>. To improve the properties of MOS and, consequentially, to improve efficiency of semiconductor metal oxides-based devices, research is nowadays moving towards nano-sized structures with controllable crystalline phases.

There are many examples of synthetic methods for obtaining ceramic nanoparticles, such us thermal and physical deposition, hydro/solvo-thermal processes<sup>5,6</sup> and, over the last few years, electrospinning (ES) <sup>7–14</sup>.

ES is a widespread and relatively simple technique to obtain both micro and nano fibers of polymers usually showing high porosity and high specific surface area <sup>15–17</sup>. Recently<sup>18,19</sup>, our research group adopted the procedure of electrospinning of polymeric solutions containing precursors of ceramics, followed by thermal treatment, for the obtaining of ceramic nanofibers.

The addition of an oxide precursors to the ES solutions and a subsequent thermal treatment, above the thermal degradation of the polymer, will ablate the polymer and, at the same time, convert the precursor in the respective oxide<sup>10,20–22</sup>. Our last work in this field was focused on the influence of the polymer on morphology and microstructure<sup>18</sup>, confirming the concept of the polymer as a 40 templating agent, but also as variable capable of influencing morphological properties and the size 41 of the final oxide particles. In this work we are proposing a new synthetic route of nanosized MOS, 42 based on the same combination of oxide precursors and polymers (in this case oligomers) as 43 templating agent, exploiting a more simple and fast approach that does not involve an 44 electrospinning step. The ES, before calcination, is indeed replaced by the simple kneading of the 45 precursors in presence of an excess of cyclodextrins.

46 Cyclodextrins (CDs) are a family of cyclic oligosaccharides, composed of  $\alpha$ -D-glucopyranose units linked through  $\alpha$  -1,4 glycosidic bonds, with a peculiar toroidal structure<sup>23</sup>. CDs are obtained 47 48 from enzymatic conversion of starch and are renewable and eco-friendly materials. Moreover CDs are biocompatible and biologically degradable<sup>24-26</sup>. The three major and relevant from the 49 50 industrial point of view representatives of CDs family are the  $\alpha$ -CD,  $\beta$ -CD and  $\gamma$ -CD and are composed of 6, 7 and 8 glucopyranose units, respectively.<sup>24,26–28</sup>. The CDs structure, with the 51 52 presence of a slightly apolar cavity and a hydrophilic external part, determines the ability to 53 establish specific interactions with various types of molecules through the formation of 54 noncovalently bonded complexes, either in the solid phase or in aqueous solution. There are many examples in literature of CDs complexes with many different organic compounds <sup>24,27,29,30</sup> or metal 55 ions $^{31,32}$  and cyclodextrins are extensively used in a wide range of applications $^{26,33}$ . 56

In 2016 <sup>34</sup>, Zhang *et al.* employed host-guest inclusion complexes as versatile (the chemical
composition of host-guest inclusion complexes can be finely tuned) precursors for obtaining
hetero-doped carbon materials.

60 The present work demonstrates the possibility to exploit inclusion complexes of CDs and metal 61 precursors to obtain nanostructured oxide: cyclodextrins plays in this case the role of a sacrificial 62 template. We selected precursors of SnO<sub>2</sub> and ZnO (two MOS traditionally obtained in form of 63 nanostructures via sol-gel synthesis or, as said, more recently, via electrospinning) because they 64 have been widely investigated in solar cells, photocatalysts and gas sensors  $^{6,19,35,36}$ : the inclusion 65 complexes were prepared using Zinc Oxide (ZnAc) and Tin (II) - ethylexanoate (SnEx) and 66 exploiting a sustainable and solvent-free approach, completely based on the simple kneading of α-67 CD, β-CD and γ-CD and precursors in ball mill.

According to the Green Chemistry Principles, published in 1998 by Anatastas *et al.*, chemical processes must be designed in order to "minimize the quantity of final waste and to avoid hazardous or toxic solvents". <sup>37,38</sup> Unfortunately, toxic solvents are present in most of processes: the recovery and reuse are often expensive, and the disposal is a major concern. Even if in much smaller quantities, the process reported in our previous article<sup>18</sup> for obtaining oxides with electrospinning also involves the use of dimethylformamide for the preparation of the polymeric solution to be electrospun.

The use of kneading for the preparation of inclusion complexes with CDs is quite established, permitting to overcome all the limitations related to poorly soluble or insoluble compounds<sup>29,39,40</sup>; in our case permitted to completely avoid the use of solvents and to speed-up the preparation of the MOS precursor.

The complexes after kneading were characterized by thermogravimetric analyses and, subsequentially, thermally treated for the obtaining of the oxide. After the synthesis the so obtained SnO<sub>2</sub> and ZnO nanoparticles were characterized by scanning electronic microscopy (SEM), highresolution transmission electron microscopy (HR-TEM) and by X-ray diffraction (XRD) and by measuring the BET surface area, evidencing a possible influence of the CDs dimension on the finale oxide.

## 86 EXPERIMENTAL

Materials

87

88	$\beta$ -CD were provided by Roquette Frères (Lestrem, France) with purity > 95%. $\alpha$ -CD, $\gamma$ -CD
89	(research grade, purity > 95.0%) were purchased from Cyclolab (Budapest, Hungary).
90	Cyclodextrins were dried in oven at 75 °C up to constant weight, before use.
91	SnO <sub>2</sub> and ZnO precursors, respectively Tin(II) 2-ethylhexanoate (SnEx, purity grade of 94.4%)
92	and Zinc Acetate (dihydrate, ZnAc), were purchased from Sigma-Aldrich (Steinheim, Germany)
93	and used without further purifications.
94	
95	Ball Mill (BM)
96	The ball mill used for the kneading step was a Retsch PM200 High Speed Planetary Ball Mill
97	(Haan, Germany), with planetary configuration and Zirconia jars, inner volume 50 mL.
98	
99	Thermogravimetric analyses
100	Thermogravimetric Analyses (TGA) were performed on a TA Instruments (New Castle, DE, USA)
101	Hi-res Q500 Thermogravimetric Analyzer. TG Analyses parameters are the same for all samples:
102	100 mL/min air-flow, heating rate 10 °C/min, RT to 700 °C. The thermograms were elaborated
103	using TA Instruments Universal Analysis 2000 software and, subsequentially, OriginLab2019b.
104	
105	
106	
107	

108 *Preparation of the samples* 

109 The preparation is the same for all oxides samples and can be divided in 2 parts:

110

111	Kneading: an appropriate amount of cyclodextrins (dried powder) is kneaded with the precursor,
112	approximately respecting the molar ratio 2:1, expressing an excess of cyclodextrins.
113	For Tin(II) 2-ethylhexanoate (viscous liquid): 3 g of $\alpha$ -CD, kneaded with 0.624 g of SnEx; 3 g of
114	$\beta$ -CD, kneaded with 0.535 g of SnEx; 4 g of $\gamma$ -CD, kneaded with 0.624 g of SnEx.
115	For Zinc Acetate (powder): 3 g of $\alpha$ -CD, kneaded with 0.338 g of ZnAc; 3 g of $\beta$ -CD, kneaded
116	with 0.290 g of ZnAc; 4 g of $\gamma$ -CD, kneaded with 0.338 g of ZnAc.
117	
118	The mixtures are kneaded for 1 h at 300 rpm. The size of the balls is 5 mm (declared diameter).
119	Approximately 12 mL (volume, measured with graduated cylinder) of sintered zirconium oxide
120	balls, 5 mm of diameter, in each jar. Quantities of kneaded reactants are optimized for this specific
121	volume of balls and, consequently free volume in the jars.
121 122	volume of balls and, consequently free volume in the jars.
121 122 123	volume of balls and, consequently free volume in the jars. <i>Calcination/Elimination of Organic Fraction</i> : after calcination all samples were thermally treated
121 122 123 124	volume of balls and, consequently free volume in the jars. <i>Calcination/Elimination of Organic Fraction</i> : after calcination all samples were thermally treated to eliminate the cyclodextrin matrix and convert the precursor into the correspondent oxide.
121 122 123 124 125	volume of balls and, consequently free volume in the jars. <i>Calcination/Elimination of Organic Fraction</i> : after calcination all samples were thermally treated to eliminate the cyclodextrin matrix and convert the precursor into the correspondent oxide. In a typical procedure approximately 8-10 g of mixture cyclodextrin/precursor contained in a
<ol> <li>121</li> <li>122</li> <li>123</li> <li>124</li> <li>125</li> <li>126</li> </ol>	<ul> <li>volume of balls and, consequently free volume in the jars.</li> <li><i>Calcination/Elimination of Organic Fraction</i>: after calcination all samples were thermally treated</li> <li>to eliminate the cyclodextrin matrix and convert the precursor into the correspondent oxide.</li> <li>In a typical procedure approximately 8-10 g of mixture cyclodextrin/precursor contained in a</li> <li>Coors combustion boat were inserted in a Lenton (Hope, UK) tubular furnace.</li> </ul>
121 122 123 124 125 126 127	volume of balls and, consequently free volume in the jars. <i>Calcination/Elimination of Organic Fraction</i> : after calcination all samples were thermally treated to eliminate the cyclodextrin matrix and convert the precursor into the correspondent oxide. In a typical procedure approximately 8-10 g of mixture cyclodextrin/precursor contained in a Coors combustion boat were inserted in a Lenton (Hope, UK) tubular furnace. Parameters were optimized via Thermogravimetric Analyses; used conditions were the following:
<ol> <li>121</li> <li>122</li> <li>123</li> <li>124</li> <li>125</li> <li>126</li> <li>127</li> <li>128</li> </ol>	volume of balls and, consequently free volume in the jars. <i>Calcination/Elimination of Organic Fraction</i> : after calcination all samples were thermally treated to eliminate the cyclodextrin matrix and convert the precursor into the correspondent oxide. In a typical procedure approximately 8-10 g of mixture cyclodextrin/precursor contained in a Coors combustion boat were inserted in a Lenton (Hope, UK) tubular furnace. Parameters were optimized via Thermogravimetric Analyses; used conditions were the following: 3h, 500° C, under 100 mL/min air-flow.
<ol> <li>121</li> <li>122</li> <li>123</li> <li>124</li> <li>125</li> <li>126</li> <li>127</li> <li>128</li> <li>129</li> </ol>	<ul> <li>volume of balls and, consequently free volume in the jars.</li> <li><i>Calcination/Elimination of Organic Fraction</i>: after calcination all samples were thermally treated</li> <li>to eliminate the cyclodextrin matrix and convert the precursor into the correspondent oxide.</li> <li>In a typical procedure approximately 8-10 g of mixture cyclodextrin/precursor contained in a</li> <li>Coors combustion boat were inserted in a Lenton (Hope, UK) tubular furnace.</li> <li>Parameters were optimized via Thermogravimetric Analyses; used conditions were the following:</li> <li>3h, 500° C, under 100 mL/min air-flow.</li> <li>A white powder was obtained from all thermal treatments.</li> </ul>

131 Scheme 1, following, shows a schematic representation of the complete procedure.

132 Further details of the synthesis will be discussed more in deep in the next sections (Results and



133 Discussion).

- 138
- 139 X-ray diffraction
- 140 X-ray diffraction (XRD) patterns of samples SnO<sub>2</sub> and ZnO obtained after thermal treatment were
- 141 collected to determine the crystalline phases and the average crystallite sizes by applying the
- 142 Scherrer's formula on the (xxx) diffraction peak.
- 143 XRD) analysis were carried out on a PW3050/60 X'Pert PRO MPD (PANalytical, Malvern, United
- 144 Kingdom), using as X-ray source a PW3373/10 LFF with Cu Anode,  $\lambda = 0.541$  Å.

- 146 *BET*
- Gas-volumetric analysis Nitrogen adsorption-desorption isotherms were measured using a
  Micromeritics ASAP2010 volumetric sorption analyzer (Norcross, GA, USA). Before

149	measurements, the samples were outgassed at 100 $^{\circ}\mathrm{C}$ under vacuum (10 mm Hg) for 15 hours (for
150	removing water and eventual adsorbates). The specific surface areas were determined through the
151	Brunauer-Emmett-Teller (BET) method to the adsorption/desorption isotherms of $N_2$ at 77 K.
152	
153	SEM
154	Morphological characterization of the samples was performed by scanning electron microscopy
155	(SEM), using a Tescan VEGA 3 SEM (Brno, Czech Republic) working with secondary electrons
156	and 10keV accelerating voltage. The samples were analyzed without any previous metal coating.
157	
158	TEM
159	High-resolution transmission electron microscopy (HR-TEM) was utilized to achieve further
160	morphological and structural information: the instrument is a Jeol (Akishima, Tokyo, Japan) JEM
161	3010 UHR (300 kV, LaB6 filament). The synthesized samples were deposited on a Cu grid, coated
162	with a porous carbon film. All digital micrographs were acquired by an Ul-traSscan 1000 camera,
163	and the images were processed by Gatan digital micro-graph (Pleasanton, CA, US).
164	
165	
166	
167	
168	
169	RESULTS AND DISCUSSION
170	The starting point for this project is the idea of using a polymer as a sacrificial template to obtain
171	very small oxide particles for specific applications. The focus is therefore on the final resulting

172 oxide but, as we recently demonstrated <sup>18</sup>, the influence of the polymer on the precursor conversion 173 is dramatic. The choice of the templating polymer, consequentially, is of primary importance. Our 174 choice fell on cyclodextrins for different reasons: CDs are a widely studied oligomer <sup>25</sup>, with an 175 extensive literature that supports many of our claims, CDs are a renewable because obtained from 176 enzymatic conversion of starch <sup>24,25</sup>. Thus, CDs are renewable, eco-friendly materials and, last but 177 not least, cheap. In Figure 1, following, are reported a simplified chemical structure of a  $\alpha$ ,  $\beta$  and 178  $\gamma$  CDs and the respective dimensions of the CDs employed in the present work.





180 Figure 1 - Simplified structure and dimensions of  $\alpha$ ,  $\beta$  and  $\gamma$  cyclodextrins

181

182  $\alpha$ ,  $\beta$  and  $\gamma$  cyclodextrins exhibit a different size and a different cavity volume (Figure 1, details).

The principle behind exploiting CDs as a sacrificial "container" is also associated the relatively high degradation temperature of the pristine CDs, that is roughly around 300°C for  $\alpha$ ,  $\beta$  and  $\gamma$ cyclodextrins: this is a crucial parameters, because the polymer "matrix", in this case the oligomer, needs to be completely removed in an efficient way, but still needs to protect e to assist in a synergic way the formation of the oxide. In order to obtain the SnO<sub>2</sub> and ZnO, it is necessary to achieve the complete volatilization of the polymer and the degradation organic precursor <sup>19</sup>. An incomplete elimination of the organic fraction will lead to a grey/black powder, with an amorphous carbon residue. At the same time, a high temperature can affect the final size of the oxide, which could favor crystal growth during calcination rather than nucleation, leading to bigger particle and lower surface area. In Table 1, following, are reported all the synthetic information and the BET surface area,  $[m^2/g]$ . A careful correlation of the results shown in Table 1 with the thermal measurements shown in Figure 2 and Figure 3 (and with data already present in literature), allows us to make some assumptions.

196

197 Table 1 – Oxides ( $\alpha$ ,  $\beta$  and  $\gamma$  SnO<sub>2</sub> and  $\alpha$ ,  $\beta$  and  $\gamma$  ZnO), relative preparative conditions (kneading step and 198 subsequential thermal treatment) and BET Surface Area (m<sup>2</sup>/g)

OXIDE		Precursor	Kneading	Thermal Treatment	BET Surface Area [m²/g]	
	$\alpha$ -SnO <sub>2</sub>	Tin (II) ethylexanoate	1 h 300 rpm 5,0 mm Ø balls 10 min inversion time	3 h 500°C ramp 10 °C/min	44,2	
SnO <sub>2</sub>	β-SnO <sub>2</sub>				47,1	
	γ-SnO <sub>2</sub>				50,4	
	α-ZnO	Zinc Acetate		1 h	2 h	12,0
ZnO	β-ZnO		300 rpm 5,0 mm Ø balls 10 min inversion time	3 n 500°C ramp 10 °C/min	10,5	
	γ-ZnO				17,5	

199

As shown in Table 1, the preparative conditions are the same for all samples: nevertheless, outcomes are quite different. The BET Surface area for what concerns the SnO<sub>2</sub> is comprised between  $\approx$  44 and 50 m<sup>2</sup>/g, for all CDs template, a quite interesting result, comparable with the results reported in the literature with oxides and similar treatments REF. It is necessary to underline, however, that these very interesting results were obtained through an objectively very simple preparation, without complex procedures and, even more noteworthy, without any solvent

206 in any step. However, the same reasoning cannot be made with regard to ZnO. The oxide obtained 207 has a much lower surface area, indeed. SnO<sub>2</sub> and ZnO are certainly two completely different oxides 208 but the dramatic difference in the final result is clear. Still, the presence of a trend in both oxides 209 from different precursors is evident: the surface area appears to be (with the sole exception of  $\beta$ -210 ZnO, Table 1 for reference) inversely proportional to the size of the cyclodextrin cavity. It is worth 211 to say that the formation of an inclusion complex is not the only possible way of CDs to form 212 complexes: stable interactions are sometimes established between a guest molecule and the 213 external part of CDs. Indeed, hydroxy group on the outer surface of cyclodextrins can lead to the 214 formation of hydrogen bonds and, consequentially, to water-soluble complexes, in a similar way to what is seen with non-cyclic oligo or polysaccharides <sup>27,41</sup>. These supramolecular assemblies 215 216 (there are no covalent bonds) are called *non-inclusion* complexes. They can, eventually, protect 217 guest molecules from the environment and achieve anyway an efficient increase of solubility. In literature many examples of non-inclusion compounds are reported <sup>42–45</sup>. Among the others there 218 219 is, of course, the size compatibility between host and guest molecules. The size of CD has to be 220 large enough to permit the guest entrance but if the guest is too small if compared to the cavity, 221 CD-guest interactions will be very weak, and the dissociated form will prevail on the associated 222 one. The same applies for the opposite situation: a molecule larger than the cavity will enter 223 partially or will arrange outside the cavity. An example in literature is, for example, the 224 naphthalimide, where the apolar chain fit preferentially in a smaller  $\alpha$ -cyclodextrin cavity, wherein the aromatic moiety of the same molecule will arrange inside a  $\beta$ -cyclodextrin <sup>46</sup>. 225

It is therefore possible to hypothesize that there is not the real formation of a 1: 1 complex, but rather the formation of "clusters". Given the interesting results these clusters are necessarily of small size; thus, it is possible to assume that in any case the ion or the precursor are, statistically, partially or completely incorporated by the cyclodextrins and this is more probable with the  $\gamma$ CD, whereas the cavity is larger. This hypothesis can also be supported by a possible interpretation of the thermogravimetric analyses, Figure 2 and Figure 3, following.



Figure 2 - Comparison of thermo-gravimetric analysis (TGA) and derivative TGA (DTGA) of the different
 component of the precursor mixture in air, from RT to 700 °C, 10 °C/min. Respectively a) Zinc Acetate b) Tin
 (II) ethylexanoate c) α, β and γ Cyclodextrins



236

Figure 3 - Comparison of thermo-gravimetric analysis (TGA) and derivative TGA (DTGA) ir, from RT to 700 °C, 10 °C/min. a)  $\alpha$ ,  $\beta$  and  $\gamma$  SnO<sub>2</sub> from Tin (II) ethylexanoate/CDs and b)  $\alpha$ ,  $\beta$  and  $\gamma$  ZnO from Zinc Acetate/CDs

241 In Figure 2, the thermo-gravimetric analyses and derivative TGA of the different component of 242 the precursor mixture in air, from RT to 700 °C, are reported. The degradation of ZnAc, inset a) 243 Figure 2, shows a thermal degradation that proceeds in two different steps: the first one at 100°C 244 reasonably related to the water and the second one, that starts before 300°C and that leads to a 245 residue of 17%, can be related to the decomposition of ZnAc itself, that is only partially converted into ZnO (ZnAc is reported to sublimate at 234 °C<sup>47</sup>). Assuming a total conversion of the precursor 246 247 into ZnO the residual should amount to approximately 37% wt. A similar situation is shown with 248 the Tin (II) ethylexanoate, inset b) Figure 2, where the residue from TG is around 21%: also in this 249 case the expected quantity of residue, with a complete conversion of the precursor in SnO<sub>2</sub>, is 37%.

In both precursors, there is a competition between degradation and oxidation, indeed. A different situation is shown in Figure 2, inset c), concerning CDs. The degradation can be divided in three stages<sup>48</sup>: the first one, below 100°C, due to loss of absorbed water and water of crystallization (in this case, since CDs were dried for at least 24h, the weight loss is barely detectable), the second, that develops in a temperature range that start at roughly 250 °C, which is associated with a weight loss of 70-80%, with the formation of a char residue and the third, at T > 300 °C, that leads to complete oxidation (combustion) of the residue.

The final residue, after the third stage, is  $\approx 0\%$  for all cyclodextrins. Cyclodextrins alone, therefore, in an oxidizing atmosphere, do not leave any residue.

259 Consequentially, looking at the thermogram reported in Figure 3, showing a residue around 4.8 % 260 for SnO2 and 4.0 % for ZnO, it is possible to state that the simultaneous presence of CDs and 261 precursor leads to the formation of an oxide and that the cyclodextrins are completely eliminated 262 from the mixture above 450 ° C. In both thermograms, inset a) and b) Figure 3, most of the weight 263 loss profile is related to the volatilization of the cyclodextrin matrix, and this is consistent with 264 what it was possible to hypothesize considering the initial composition of the mixture after 265 kneading. What is necessary to highlight is the amount of residual oxide, because it could confirm 266 a protective effect of the cyclodextrins: the theoretical residue in oxide, assuming a total 267 conversion of the precursor present in the post-kneading mixture is 4.5 % for ZnO and 5.6 % for 268 SnO<sub>2</sub>, respectively. Thus, results are remarkably close to the 4.0 % and 4.8 % reported before. The 269 yield is low, but we must consider that the precursor/CDs kneaded mixture contains approximately 270 15% wt. of precursor: tin (II) 2-ethylhexanoate and zinc acetate, if heated in the same conditions, 271 volatilize, leaving a considerably smaller amount of residue, while in this case the precursor is 272 almost completely converted into oxide.

Cyclodextrins, therefore, do not only play a role of simple support but rather protect and assist the
formation of oxide (preventing the volatilization of the precursor), influencing both its yield and
final size.

However, it is necessary to make further considerations, about the possible reasons why the same interesting results, concerning the BET surface area (and consequentially particle size) were not obtained as for SnO<sub>2</sub> with ZnO. In Figure 2 it is evident that cyclodextrins have their principal weight loss at a temperature comparable to that of zinc acetate, while it is about 50 °C higher than that of tin (II) 2-ethylhexanoate: given the results previously commented and assuming therefore that there is a protective effect, this is effective only if the "shield" molecule does not degrade before the host molecule.

283 Moreover, a less effective protective effect can also be hypothesized from an analysis of the TG 284 curves reported in inset b) of Figure 3: in the area around 400 degrees, where in presence of oxygen 285 presumably occurs the oxidation/combustion of the cyclodextrins, it is possible to notice an 286 anomalous behavior (very evident in the peaks of the first derivative). These small weight losses 287 can be related to volatiles that are quickly released as soon as the protective "cage" is thermally 288 oxidized. Given the tendency of ZnAc to sublimate, reported in the literature, it can be 289 hypothesized that part of the precursor is actually not converted but rather simply "blocked" and 290 subsequently lost during the heating.

Another plausible reason in the different superficial area from a comparison of the two different precursors can be related to the fact that in the case of the SnO<sub>2</sub>, the kneading phase is carried out with a liquid precursor, tin (II) 2-ethylhexanoate, instead of a powder such as in the case of the Zinc Acetate: the liquid form of the precursor leads to a much more effective grinding / mixing,

- leading to the obtainment of a sort of "paste". From a practical point of view, the kneading phase
- 296 therefore seems to have a significant influence on the final result.
- 297
- 298 We also characterized the final oxides from the morphological point of view by scanning electron
- 299 microscopy, results are shown below (Figure 4):



- 300
- Figure 4 SEM images of the oxide obtained from different precursors and CDs. Instrumental magnification
   5kx, 10keV
- 303

The suitable temperatures for calcining the mixture after the kneading step were chosen on the results obtained by TG analyses, considering the temperatures at which the weight losses are concluded: the same temperature, 500 °C was, used for the calcination of all kneaded mixtures,

307 leading to a white and uniform oxide residue. As shown in Figure 4, there are no particular 308 differences that emerge from the SEM comparison of the samples, even if obtained from different 309 mixtures of oxides and oligomers. Higher magnification are reported in Supporting Informations.



#### 310

#### 311 Figure 5 XRD patterns of SnO<sub>2</sub> and ZnO, from precursot/yCD mixtures.

The XRD confirms that the samples are constituted by crystalline cassiterite  $(SnO_2 \text{ in the} 1313 \text{ tetragonal crystal phase, JCPDS file number 00-001-0625})$ . The calculated size (D, via Debye-Sherrer equation) is .....

- 315 TEM
- 316 CONCLUSIONS
- 317

319

320

321	CORRESPONDING	AUTHOR
-----	---------------	--------

322 \*alberto.rubinpedrazzo@unito.it, Post-Doc Fellow

323 ABBREVIATIONS

- 324  $\alpha$ CD, alpha-cyclodextrin;  $\beta$ CD, beta-cyclodextrin;  $\gamma$ CD, gamma-cyclodextrin; BM, ball-mill;
- 325 SnEx, Tin(II) 2-ethylhexanoate; ZnAc, Zinc Acetate;
- 326

- 328
- 329 REFERENCES
- Bracco, P., Scalarone, D., Trotta, F. Electrospun Membranes for Sensors Applications, in:
   Smart Membranes and Sensors: Synthesis, Characterization, and Applications. *Ed. Scrivener Publ.* 2014, 301–336.
- Eranna, G.; Joshi, B. C.; Runthala, D. P.; Gupta, R. P. Oxide Materials for Development of
  Integrated Gas Sensors A Comprehensive Review. *Crit. Rev. Solid State Mater. Sci.* 2004,
  29 (3–4), 111–188. https://doi.org/10.1080/10408430490888977.
- 336 (3) Sun, Y.-F.; Liu, S.-B.; Meng, F.-L.; Liu, J.-Y.; Jin, Z.; Kong, L.-T.; Liu, J.-H. Metal Oxide
   337 Nanostructures and Their Gas Sensing Properties: A Review. *Sensors* 2012, *12* (12), 2610–
   338 2631. https://doi.org/10.3390/s120302610.
- 339 (4) Zaera, F. Nanostructured Materials for Applications in Heterogeneous Catalysis. *Chem.*340 Soc. Rev. 2013, 42 (7), 2746–2762. https://doi.org/10.1039/C2CS35261C.
- 341 (5) Yu, X.; Marks, T. J.; Facchetti, A. Metal Oxides for Optoelectronic Applications. *Nat. Publ.*342 *Gr.* 2016, *15* (4), 383–396. https://doi.org/10.1038/nmat4599.

- 343 (6) Das, S.; Jayaraman, V. SnO2: A Comprehensive Review on Structures and Gas Sensors.
  344 *Prog. Mater. Sci.* 2014, *66*, 112–255. https://doi.org/10.1016/j.pmatsci.2014.06.003.
- 345 (7) Li, D.; Xia, Y. Fabrication of Titania Nanofibers by Electrospinning. *Nano Lett.* 2003, *3*346 (4), 555–560. https://doi.org/10.1021/nl034039o.
- 347 (8) Choi, S. S.; Lee, S. G.; Im, S. .; Kim, S. H.; Joo, Y. L. Silica Nanofibers from
   348 Electrospinning/Sol-Gel Process. *J. Mater Sci. Lett.* 2003, 22 (12), 891–893.
- 349 (9) Morandi, S.; Cecone, C.; Marchisio, G.; Bracco, P.; Zanetti, M.; Manzoli, M. Shedding 350 Light on Precursor and Thermal Treatment Effects on the Nanostructure of Electrospun 351 TiO2 Fibers. Nano-Structures and *Nano-Objects* 2016, 7, 49-55. 352 https://doi.org/10.1016/j.nanoso.2016.05.003.
- (10) Chandraiah, M.; Sahoo, B.; Panda, P. K. Preparation and Characterization of SnO2
  Nanofibers by Electrospinning. *Trans. Indian Ceram. Soc.* 2014, 73 (4), 266–269.
  https://doi.org/Doi 10.1080/0371750x.2014.923786.
- Madhugiri, S.; Sun, B.; Smirniotis, P. G.; Ferraris, J. P.; Balkus, K. J. Electrospun
  Mesoporous Titanium Dioxide Fibers. *Microporous Mesoporous Mater.* 2004, 69 (1–2),
  77–83. https://doi.org/10.1016/j.micromeso.2003.12.023.
- Bazargan, A. M.; Fateminia, S. M. A.; Ganji, M. E.; Bahrevar, M. A. Electrospinning
  Preparation and Characterization of Cadmium Oxide Nanofibers. *Chem. Eng. J.* 2009, *155*(1–2), 523–527. https://doi.org/10.1016/j.cej.2009.08.004.
- 362 (13) Dai, Y.; Liu, W.; Formo, E.; Sun, Y.; Xia, Y. Ceramic Nanofibers Fabricated by
   363 Electrospinning and Their Applications in Catalysis, Environmental Science, and Energy
   364 Technology. *Polym. Adv. Technol.* 2011, 22 (3), 326–338. https://doi.org/10.1002/pat.1839.
- Masa, S.; Hontanon, E.; Santos, J. P.; Sayago, I.; Lozano, J. Chemiresistive Sensors Based
  on Electrospun Tin Oxide Nanofibers for Detecting NO2 at the Sub-0.1 Ppm Level. *Proc. 2019 5th Exp. Int. Conf. exp.at 2019, 2019, Funchal, Madeira Island, Port.* 2019, No. 2,
  310–314. https://doi.org/10.1109/EXPAT.2019.8876485.

- 369 (15) MacDiarmid, A. G.; Jones, W. .; Norris, J. D.; Gao, J.; Johnson, A. T.; Pinto, N. J.; Hone,
  370 J.; Han, F. K. Electrostatically-Generated Nano Fibers of Electronic Polymers. *Synth. Met.*371 2001, *119* (1–3), 27–30.
- 372 (16) Theron, S. A.; Zussman, E.; Yarin, A. L. Experimental Investigation of the Governing
  373 Parameters in the Electrospinning of Polymer Solutions. 2004, 45 (2004), 2017–2030.
  374 https://doi.org/10.1016/j.polymer.2004.01.024.
- 375 (17) Liu, S.; White, K. L.; Reneker, D. H. Electrospinning Polymer Nanofibers with Controlled
  376 Diameters. *IEEE Trans. Ind. Appl.* 2019, 55 (5), 5239–5243.
  377 https://doi.org/10.1109/TIA.2019.2920811.
- 378 (18)Rubin Pedrazzo, A.; Cecone, C.; Morandi, S.; Manzoli, M.; Bracco, P.; Zanetti, M. 379 Nanosized Sno2 Prepared by Electrospinning: Influence of the Polymer on Both 380 2021. Morphology and Microstructure. *Polymers* (Basel). 13 (6). 381 https://doi.org/10.3390/polym13060977.
- 382 (19) Fioravanti, A.; Morandi, S.; Rubin Pedrazzo, A.; Bracco, P.; Zanetti, M.; Manzoli, M.;
  383 Mazzocchi, M.; Carotta, M. C. Ultrasensitive Gas Sensors Based on Electrospun TiO2 and
  384 ZnO. *Proceedings* 2017, *1* (4), 485. https://doi.org/10.3390/proceedings1040485.
- Wang, X.; Fan, H.; Ren, P. Electrospinning Derived Hollow SnO2 Microtubes with Highly
  Photocatalytic Property. *Catal. Commun.* 2013, 31, 37–41.
  https://doi.org/10.1016/j.catcom.2012.11.009.
- 388 Sun, X.; Huang, Y.; Zong, M.; Wu, H.; Ding, X. Preparation of Porous SnO2/ZnO (21)389 Nanotubes via a Single Spinneret Electrospinning Technique as Anodes for Lithium Ion 390 Batteries. J. Sci. Mater. Electron. 2016. 27 (3), 2682-2686. Mater. 391 https://doi.org/10.1007/s10854-015-4077-x.
- 392 (22) Sun, Y.; Wang, J.; Du, H.; Li, X.; Wang, C.; Hou, T. Formaldehyde Gas Sensors Based on
  393 SnO2/ZSM-5 Zeolite Composite Nanofibers. *J. Alloys Compd.* 2021, 868, 159140.
  394 https://doi.org/10.1016/j.jallcom.2021.159140.
- 395 (23) Kurkov, S. V.; Loftsson, T. Cyclodextrins. Int. J. Pharm. 2013, 453 (1), 167-180.

396

https://doi.org/10.1016/j.jpharm.2012.06.055.

- 397 (24) Duchêne, D. Cyclodextrins and Their Inclusion Complexes. *Cyclodextrins Pharm. Cosmet.* 398 *Biomed. Curr. Futur. Ind. Appl.* 2011, 1–18. https://doi.org/10.1002/9780470926819.ch1.
- 399 (25) Szejtli, J. Introduction and General Overview of Cyclodextrin Chemistry. *Chem. Rev.* 1998,
  400 98 (5), 1743–1754. https://doi.org/10.1021/cr970022c.
- 401 (26) Szejtli, J. Utilization of Cyclodextrins in Industrial Products and Processes. J. Mater. Chem.
  402 1997, 7 (4), 575–587. https://doi.org/10.1039/a605235e.
- 403 (27) Loftsson, T.; Duchêne, D. Cyclodextrins and Their Pharmaceutical Applications. *Int. J.*404 *Pharm.* 2007, 329 (1–2), 1–11. https://doi.org/10.1016/j.ijpharm.2006.10.044.
- 405 (28) Szejtli, J. Cyclodextrin Technology; s.l.:Springer Netherlands, 1988.
- 406 (29) Marques, H. M. C. A Review on Cyclodextrin Encapsulation of Essential Oils and Volatiles.
   407 *Flavour Fragr. J.* 2010, *25* (5), 313–326. https://doi.org/10.1002/ffj.2019.
- 408 (30) Saenger, W. Cyclodextrin Inclusion Compounds in Research and Industry. *Angew. Chemie*409 *Int. Ed. English* 1980, *19* (5), 344–362. https://doi.org/10.1002/anie.198003441.
- 410 (31) Prochowicz, D.; Kornowicz, A.; Lewiński, J. Interactions of Native Cyclodextrins with
  411 Metal Ions and Inorganic Nanoparticles: Fertile Landscape for Chemistry and Materials
  412 Science. *Chem. Rev.* 2017, *117* (22), 13461–13501.
  413 https://doi.org/10.1021/acs.chemrev.7b00231.
- 414 (32) Norkus, E. Metal Ion Complexes with Native Cyclodextrins. An Overview. J. Incl. Phenom.
  415 Macrocycl. Chem. 2009, 65 (3), 237–248. https://doi.org/10.1007/s10847-009-9586-x.
- 416 (33) Erem Bilensoy. Cyclodextrins in Pharmaceutics, Cosmetics, and Biomedicine: Current and
  417 Future Industrial Applications; John Wiley and Sons Inc, Ed.; 2011.
- 418 (34) Zhang, J.; Xu, D.; Qian, W.; Zhu, J.; Yan, F. Host-Guest Inclusion Complexes Derived
  419 Heteroatom-Doped Porous Carbon Materials. *Carbon N. Y.* 2016, 105, 183–190.
  420 https://doi.org/10.1016/j.carbon.2016.04.034.

- 421 (35) Q. Qi, T. Zhang, L. L. and X. Z. Synthesis and Toluene Sensing Properties of SnO2
  422 Nanofibers. *Sensors Actuators B Chem.* 2009, *137*, 471.
- 423 (36) Wang, W.; Huang, H.; Li, Z.; Zhang, H.; Wang, Y.; Zheng, W.; Wang, C. Zinc Oxide
  424 Nanofiber Gas Sensors via Electrospinning. *J. Am. Ceram. Soc.* 2008, *91* (11), 3817–3819.
  425 https://doi.org/10.1111/j.1551-2916.2008.02765.x.
- 426 (37) Anastas, P. T.; Warner, J. C. *Green Chemistry: Theory and Practice*; Oxford University
  427 Press, 1998.
- 428 (38) Anastas, P. T.; Zimmerman, J. B. Design through the 12 Principles of Green Engineering.
  429 *IEEE Eng. Manag. Rev.* 2007, *35* (3), 16. https://doi.org/10.1109/EMR.2007.4296421.
- 430 (39) Verma, S.; Rawat, A.; Kaul, M.; Saini, S. Solid Dispersion: A Strategy for Solubility
  431 Enhancement. *Int. J. Pharm. Technol.* 2011, *3* (2), 1062–1099.
- 432 (40) Patil, J. S.; Kadam, D. V.; Marapur, S. C.; Kamalapur, M. V. Inclusion Complex System; a
  433 Novel Technique to Improve the Solubility and Bioavailability of Poorly Soluble Drugs: A
  434 Review. *Int. J. Pharm. Sci. Rev. Res.* 2010, *2* (2), 29–34.
- 435 (41) Crini, G. Recent Developments in Polysaccharide-Based Materials Used as Adsorbents in
  436 Wastewater Treatment. *Prog. Polym. Sci.* 2005, 30 (1), 38–70.
  437 https://doi.org/10.1016/j.progpolymsci.2004.11.002.
- 438 (42) Loftsson, T.; Saokham, P.; Sá Couto, A. R. Self-Association of Cyclodextrins and
  439 Cyclodextrin Complexes in Aqueous Solutions. *Int. J. Pharm.* 2019, *560* (5), 228–234.
  440 https://doi.org/10.1016/j.ijpharm.2019.02.004.
- 441 (43) Loftsson, T.; Magnúsdóttir, A.; Másson, M.; Sigurjónsdóttir, J. F. Self-Association and
  442 Cyclodextrin Solubilization of Drugs. *J. Pharm. Sci.* 2002, *91* (11), 2307–2316.
  443 https://doi.org/10.1002/jps.10226.
- 444 (44) Mazzaglia, A.; Sciortino, M. T.; Kandoth, N.; Sortino, S. Cyclodextrin-Based
  445 Nanoconstructs for Photoactivated Therapies. J. Drug Deliv. Sci. Technol. 2012, 22 (3),
  446 235–242. https://doi.org/10.1016/S1773-2247(12)50034-1.

447	(45)	Zhang, J.; Ma, P. X. Cyclodextrin-Based Supramolecular Systems for Drug Delivery:
448		Recent Progress and Future Perspective. Adv. Drug Deliv. Rev. 2013, 65 (9), 1215–1233.
449		https://doi.org/10.1016/j.addr.2013.05.001.
450	(46)	Brochsztain, S.; Politi, M. J. Solubilization of 1,4,5,8- Naphthalenediimides and 1, 8-
451		Naphthalimides through the Formation of Novel Host-Guest Complexes with a-
452		Cyclodextrin. Langmuir 1999, 15, 486–4494.
453	(47)	Mar, G. L.; Timbrell, P. Y.; Lamb, R. N. Formation of Zinc Oxide Thin Films by the
454		Thermal Decomposition of Zinc Acetate. 1993, 73, 177–192. https://doi.org/10.1007/978-
455		3-642-84933-6_15.
456	(48)	Trotta, F.; Zanetti, M.; Camino, G. Thermal Degradation of Cyclodextrins. 2000, 69, 373-
457		379.
458		

### 459 TOC/GRAPHIC ABSTRACT

