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LONG-TERM *IN-VITRO* WEAR PERFORMANCE OF AN INNOVATIVE THERMO-COMPRESSED CROSS-LINKED POLYETHYLENE

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NOMENCLATURE

PE-A	conventional PE GUR-1020 (EtO-sterilized)
PE-B	conventional PE GUR-1020 (γ-sterilized)
PE-C	conventional PE GUR-1050 (γ-sterilized)
XLPE	cross-linked PE GUR-1020 (EtO-sterilized)
XLPE-RT	cross-linked-Thermo-Compression PE GUR-1050 (EtO-sterilized)
Mc	Million cycles
KW	Kruskal-Wallis non-parametric statistical tests
KS	Kolmogorov-Smirnov non-parametric statistical tests

1 ABSTRACT

were compared with conventional and traditional commercially available cross-linked polyethylene in terms of wear behavior in a hip simulator for 10 millions cycles using bovine calf serum as lubricant. Gravimetric measurements revealed significant differences between the wear behaviors of the five sets of acetabular cups. In particular, this new type of thermo-compression crosslinked wore more than the traditional cross-linked polyethylene but exhibited a wear rate about four times lower than conventional UHMWPE. The FTIR analyses indicate that oxidation to various extents appears as a consequence of γ -irradiation in presence of oxygen. Keywords: PE GUR-1020; PE GUR-1050; Wear, Long-term wear behavior; Conventional UHMWPE; FTIR analysis; Oxidation.

New cross-linked polyethylene acetabular cups, obtained by thermo-compression process,

19 2 INTRODUCTION

20 The research to improve the life of a patient that had a hip implant remains a major concern of 21 scientists in the world. It is well know that ultra-high-molecular-weight-polyethylene 22 (UHMWPE) is the most commonly used material in orthopaedics implants even if wear due 23 to this material represents a serious clinical problem [1]. Excessive wear may lead to gross 24 mechanical failure such as fracture and disassociation [2] and the release of particulate wear 25 debris may induce biological responses that cause implant loosening [3, 4]}. Medical grade 26 resins are described as GUR-1020 e GUR-1050 depending on their molecular weights. The requirements for medical grade UHMWPE powder are specified in ASTM standard F648 and 27 28 ISO standard 5834-1. These standards characterize powders on molecular weight, trace 29 impurities of titanium, aluminium, and chlorine (residuals from catalysts), and ash content. 30 No analysis is required on the final form that can be achieved by different processes [5-7]; 31 these may not produce uniform conditions in the manufactured shape and there is still no 32 consensus on which resin and conversion method would be universally proposed as the best choice for all the orthopaedic applications. 33

34 At the moment, the focus of research in total joint arthroplasty has shifted to improve the wear performance of conventional UHMWPE and increase its mechanical properties, such as 35 36 yield stress and Young's modulus. To this end, increasing the cross-linked density is well 37 established as a promising solution to improve the wear resistance and durability of 38 polyethylene [3, 8]. The molecular chain structure at the articulating surface of polyethylene 39 is believed to undergo a reorganization process due to strain accumulation caused by surface 40 traction [9]. This altered polyethylene responds to stresses favourably if loaded along the fiber 41 axis but unfavourably if loaded off axis [10]. Several different manufacturing processes have 42 been developed to increase the wear characteristic of conventional polyethylene in order to 43 achieve the so-called cross-link polyethylene (XLPE). The industrial process consists to treat 44 the UHMWPE with high-energy radiation, peroxide, or silane treatments [3, 8, 11]. The 45 common factor of these aforementioned processes, is the treatment with γ -irradiation or e-46 beam at dose between 40 to 100 kGy [1, 3, 8, 9, 11-14]) before the machining of PE bar into 47 the final form; this increase the crosslinking density and improve wear resistance of the 48 conventional polyethylene. Radiation-induced cross-linking of UHMWPE has been shown to 49 improve wear resistance by up to 5 fold if compared to conventional polyethylene [15-17]. 50 This decrease in wear has been observed in clinical studies [18, 19], in laboratory tests using 51 hip joint simulators [8, 20, 21], and under both ideal and severe wear conditions [14, 22].

52 However, as the level of polyethylene cross-linking increases, the mechanical properties, such 53 as the ductility, yield strength, and elongation to failure decreases [1, 11, 23-25]. Moreover, 54 this result in a decrease in the toughness [1, 26] and acetabular cups fabricated from these 55 materials might present a risk for wear increase and catastrophic failure [24]. At this concern, 56 improvements in the forming process of XLPE acetabular cups were implemented. In 57 particular, a thermo-compression process, at a temperature above the melting point, follows 58 the irradiation phase completing the manufacturing of the articulating surface of the 59 acetabular cup. As a consequence, mechanical machining of the articulating surface is 60 avoided and higher material homogeneity and superficial finish achieved. Absence of 61 machining marks on the surface of the polyethylene should reduce the abrasion and this could 62 end in improved wear resistance of the XLPE cups.

With this in mind, in this work we asked whether new cross-linked PE acetabular cups,
obtained by thermo-compression process (XLPE-RT), would result in less wear than XLPE
and conventional UHMWPE.

66 In particular, two outcomes were used as the basis for comparing wear behavior:

5

- Wear rate (determined from sequential gravimetric measurements during the course of the
- test) my be influenced by material composition (different GUR) and sterilization;
- 69 Degree of oxidation of the acetabular cups (determined from FTIR measurements).

On the basis of an initial analysis [27] it was decided to continue testing up to 10 million cycles (Mc) in order to evaluate the long-term wear effects. This study provides additional important information to that already described [27], since it clarifies the characteristics of the long-term wear resistance of the materials.

74 **3 MATERIALS AND METHODS**

75 **3.1** Specimens tested

The wear behaviour of 20 different polyethylene acetabular cups (28-mm inner x 44-mm outer; 4 specimens for each batch) coupled with 28-mm CoCrMo femoral heads were investigated using a hip joint simulator. The conventional PE and the XLPE resins studied were surgical grade consolidated by compression moulding according to the ISO 5834/2. More details are shown in Table 1.

Following a standardized procedure [22, 28-30], another 10 acetabular cups (two for each type of material used) were stored (non-loaded) in bovine calf serum to compensate for weight chances due to fluid absorption. All specimens were tested three months after their sterilization. All polyethylene acetabular cups were pre-soaked for four weeks prior the wear tests.

86

3.2 Experimental wear details

Wear tests were performed using a 12-station hip joint simulator (Shore Western, USA). The simulator set-up followed is described in details elsewhere [27, 31, 32]. Each articulating station was subjected to a sinusoidal loading (max 2 kN) with a frequency of 1.1 Hz, according to the rotation test frequency under room temperature (23 ± 1 °C) conditions.

The weight loss of the cups was determined every 0.5 Mc until five Mc and every one Mc from 5 to 10 Mc. A SARTORIUS microbalance (AG, Germany) with a precision of ± 0.1 mg was used to measure the weight loss during the experiments.

94 The effects of the different polyethylenes on wear were evaluated using *KW* and *KS* statistical 95 tests. Statistical significance was set at P < 0.05. 96

3.3 Roughness measurements

97 The surface roughness of all the femoral heads was measured using a contact profilometer 98 Hommel Tester T8000 (Hommel Werke, Germany). Scanning operations were performed 99 identifying three planes according to previous standardized protocol [33]. Sampling lengths 100 were taken using a cut-off of 0.08 mm. Two parameters (R_a and R_t), used to qualify the 101 surface roughness, were taken into consideration as previously reported [33]. In particular, R_a 102 is the arithmetic mean of the sum of roughness profile values (DIN EN ISO 4287 - 1997) and 103 does not indicates if valleys or peaks cause the roughness. R_t is the vertical distance from the 104 deepest valley to the highest peak of the roughness profile over the sampling length. The 105 measurements were acquired at 0, 5, and 10 Mc.

106 A statistical analysis (factorial ANOVA, analysis of variance, statistical significance was set 107 at p < 0.05)[34] was performed to find out whether the various surface profile parameters (R_a 108 and R_t) of the femoral heads could predict the variance observed in the weight loss of the 109 acetabular cups measurements. A linear regression analysis was performed between each 110 roughness parameter and the weight loss measurements; the linear regression intercept 111 coefficient and squared linear regression coefficient R² are reported.

112

113 **3.4 FTIR spectroscopy**

The most worn polyethylene acetabular cups (one for each batch) were cut perpendicular to the articulating surface; from the cross section a series of ca. 180 μ m thick slices was obtained using a PolyCuts Microtome (Reichert-Jung) at 10 mm/s in air at room temperature. A FTIR Microscope (Spectrum Spotlight 300, Perkin-Elmer, Shelton, Connecticut, USA) was used to map the oxidation. For collecting the line-scan spectra, the area of analysis was set at 100 x 100 μ m² and the spectra were recorded every 100 μ m along the mapping direction, starting

120 from the articulating surface towards the bulk. Each line scan was collected starting from the worn area of the bearing surface. All spectra were run in the transmission mode with a 4 cm^{-1} 121 resolution and 16 scans per spectrum. All the spectra were normalised at 2020 cm-1 at 122 absorption of 0.05, corresponding to a film thickness of 100 µm. The peak at 2020 cm⁻¹, a 123 124 combination band associated with the twisting of CH₂, was used as an internal standard, since 125 it can be regarded as unaffected by minor changes in the polymer structure [35]. The degree of oxidation of samples was determined by the ketones absorption at 1718 cm⁻¹. 126 127 Hydroperoxides were detected after reaction with nitrogen monoxide (NO) for 15 hours in a 128 dark reaction vessel. Under these conditions, hydroperoxides in polyethylene are converted into more easily detectable nitrates at 1630 cm⁻¹ as reported by Lacoste et al. [36, 37]. 129

4 RESULTS

131 All the acetabular cups and the femoral heads completed the planned 10 Mc.

132 Compared to the different cup configurations, the XLPE-RT combination wore more than the

133 XLPE but maintained a lower weight loss than the other conventional PE during the whole

134 test (Fig. 1).

Significant statistical differences (p < 0.0001) were observed between all the polyethylene cups using the KS statistical test. The same significance (KW, p < 0.05) was also observed between the two cross-linked PE materials and between the three conventional PE acetabular cups. Significant statistical differences (p < 0.05) were also found considering the type of resin (PE-A vs. PE-B) and between the groups with the same sterilization (PE-B vs. PE-C).

140 The mean roughness measurements (\pm Standard Deviation) of the parameters (R_a , R_t) for the 141 femoral heads under investigation at zero Mc, 5 Mc, and 10 Mc are reported in Table 2.

142 The coefficient of correlation between the weight loss and the two roughness parameters (R_a 143 and R_t) for the femoral heads coupled with the various different polyethylenes are plotted in 144 Figs. 2 & 3.

The average roughness R_a for the heads coupled with the cross-linked PE showed the lower coefficient of linear regression ($R^2 = 0.45$) in comparison with the conventional PE ($R^2 = 0.81$). The same correlation was observed with the R_t parameter ($R^2 = 0.27 \text{ vs. } R^2 = 0.84$).

The FTIR line-scan spectra measured on the aforementioned polyethylene acetabular cups are shown in Fig. 4. All samples show diffusion of apolar products from the bovine serum, as indicated by the ester absorption at 1740 cm⁻¹ found in all spectra. However, this phenomenon was found to be more superficial than in samples retrieved after in vivo implantation [38, 39]. The FTIR spectrum of PE-A shows only a slight oxidation, evidenced by the hydroperoxides absorption at 1630 cm⁻¹. Samples B and C show the same oxidation profile, but the oxidation level of sample B is much higher. The presence of oxidation is also confirmed by the absorption of ketones at 1718 cm⁻¹. The oxidation found in these samples can be attributed to a poor packaging storage, in agreement with other study that observed this phenomenon [40].

The two-crosslinked samples show different oxidation level: a minimal absorption of both ketones and hydroperoxides indicates quite a low oxidation for XLPE. On the contrary, a high concentration of ketones and hydroperoxides indicates a high oxidation level of XLPE-RT, comparable to that of PE-B.

Evidently, not all hydroperoxides were eliminated due to a too short thermal treatment. In addition, the higher oxidation level found in XLPE-RT can be explained with the lack of mechanical machining after irradiation. Thus, if from one hand the absence of machining marks on the articulating surface could reduce the abrasion, on the other end, eliminating the mechanical machining also means failing in removing the superficial layer of polyethylene, which have been oxidized during irradiation.

167 **5 DISCUSSION**

Interest in an improved polyethylene that present wear resistance for orthopaedic implants is an open challenge. The wear performance of five different polyethylenes coupled with CoCrMo femoral heads was evaluated for 10 million cycles. In particular, we asked whether new XLPE-RT acetabular cups, after long-term wear tests, would result in less wear than long-term wear tests would result in less wear than XLPE and conventional UHMWPE after UHMWPE and to determine the degree of oxidation of these specimens.

We found a reduced wear rate for the XLPE configurations in comparison to the conventional PE. In particular, the XLPE and XLPE-RT cups showed a significant 80% to 90% wear reduction compared with the conventional PE (Fig. 1). XLPE-RT acetabular cups showed higher weight losses than XLPE (50%). It was stressed that our initial hypothesis related to the absence of machining marks on the surface of the polyethylene should be reduce the abrasion and this could ends in improved wear resistance of the cups doesn't verified but the high oxidation level observed on these new XLPE-RT cups could altered the results observed.

181 Significant differences (p < 0.05) were observed between all the polyethylene cups. In 182 particular, the results of this investigation clearly showed a reduced wear for the two types of 183 XLPE with respect to the conventional UHMWPE components.

Hip simulator studies have indicated that cross-linking can reduce the type of wear (>95%) that occurs in acetabular components in *in-vitro* tests [11, 41-45]. These results are also confirmed from *in-vivo* follow-up studies: a significant decrease in the femoral head penetration into the highly cross-linked polyethylene acetabular liners compared with the conventional UHMWPE liners was observed by Martell and D'Antonio [15, 46]. Regarding the sterilisation treatments, we found that our γ -irradiated PE-B specimens continued to wear about four times faster than the EtO sterilised of the same composition (Fig. 1). The oxidation level of PE-B, partially crosslinked by radiation sterilization, is much higher than that of the
EtO sterilized sample (PE-A). Thus, the oxidation induced by irradiation in air led to a
decrease in the molecular mass.

A concern about the same sterilisation method but different material show the PE-B cups (γ irradiated) wore about 3 times than the PE-C cups sterilised under the same conditions. This result can be explained by the different oxidation levels found in the two samples, according to the mechanism proposed by Costa et al. [4]. In particular, a decrease of molecular weight may involve abrasion.

199 Concerning the roughness measurements, R_a and R_t were taken into account because, as better 200 explained in a previous work [33], the femoral heads implanted in patients could be scratched 201 at various stages and variations of the roughness during the life of the prosthesis may not 202 change in a systematic manner [47]. Profiles of different surfaces could have the same 203 roughness average (R_a) and wavelength but have different shapes. In this sense, R_t is the other 204 parameter that can better correlate with the wear rate, when femoral heads with substantially 205 different surface conditioning are compared. A slight increase in R_a may be observed from 5 206 to 10 Mc for all the combinations (Table 2). Conversely, an appreciable increase in Rt is observed for all the configurations. It was stressed that the PE-B group presents higher Ra and 207 higher Rt with respect to the other polyethylenes. Probably that isolated scratches on the PE-B 208 209 acetabular cups increased the wear rate significantly more than generalized roughness of the 210 metallic counter-face and could also change the wear performance ranking of various 211 polyethylene formulations. A correlation between wear that occurs at 10 Mc and weight loss is observable only for high weight loss (conventional polyethylenes with $R^2 > 0.8$). Regarding 212 213 the cross-linked PE the model is not able to describe a correlation between the roughness and 214 the weight loss because the variations are smaller than measurements accuracy.

2156CONCLUSIONS

216 All the acetabular cups (cross-linked and conventional PE) showed significant differences 217 between them in respect to the wear behavior. Weight loss was found to decrease for the 218 XLPE than the other configurations. However, this study is limited in predicting clinical wear 219 rates. It is stressed that these wear tests were conducted in a laboratory under controlled conditions on a hip simulator. The FTIR analyses indicate that oxidation to various extents 220 221 appears as a consequence of γ -irradiation in presence of oxygen and therefore the XLPE cups 222 in our study showed the most variability in wear rates, whereas the conventional PE cups 223 showed much more uniform wear behavior.

Differences in the wear behaviour could be correlated to different oxidation levels. In fact, while these levels increase for the cups, the roughness values keep constant as to indicate that improvement in manufacturing process influences the quality of substrate rather then the surface. Further studies are in progress to better investigate the reasons of different oxidation levels and to correlate the wear behaviour to the spatial distribution of the oxidation.

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230

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Type of resin	Manufacturing process	Irradiation dose	Final sterilization	
PE-A**	Mechanical machining from bar extrusion to obtain the final form.		EtO	
PE-B**	Mechanical machining from bar extrusion to obtain the final form.		γ 3 ±0.5Mrad (Dose in vacuum packed)	
PE-C**	Mechanical machining from bar extrusion to obtain the final form.		γ 3 ±0.5 Mrad (Døse in vacuum packed)	
XLPE**	Machining into the final form. Washing and packaging under controlled environment (standard procedure for biomedical components).	y-irradiation dose in the range 67.5 ÷ 82.5 kGy (nominal dose 100kGy) Annealing at 150°C under nitrogen for 6 hours.	EtO	
XLPE-RT**	Thermo-compression according to ISO 5834/2 into a cylindrical pre-form, which allows obtaining the concavity of the cup with no need of further machining.	$\begin{array}{l} \gamma \text{-radiation dose in the} \\ \text{range $2.9 \div 97.5 kGy} \\ \text{(nominal dose 100kGy)} \\ \text{Annealing at 150°C} \\ \text{under nitrogen for 6} \\ \text{hours.} \end{array}$	EtO	

** More details about the type of resin is given in the Nomenclature section.

Cycles		$R_a(\mu m)$			$R_t(\mu m)$	
(Mc)	0 Mc	5 Mc	10 Mc	0 Mc	5 Mc	10 Mc
PE-A	0.01 ± 0.01	0.02 ± 0.01	0.02 ± 0.01	0.05 ± 0.01	0.17 ± 0.09	0.16 ± 0.05
PE-B	0.01 ± 0.01	0.02 ± 0.01	0.05 ± 0.03	0.07 ± 0.01	0.15 ± 0.04	0.36 ± 0.21
PE-C	0.01 ± 0.01	0.01 ± 0.01	0.01 ± 0.01	0.05 ± 0.01	0.11 ± 0.03	0.10 ± 0.01
XLPE	0.01 ± 0.01	0.01 ± 0.01	0.02 ± 0.01	0.05 ± 0.02	0.16 ± 0.07	0.21 ± 0.03
XLPE-RT	0.01 ± 0.01	0.01 ± 0.01	0.02 ± 0.01	0.05 ± 0.01	0.11 ± 0.04	0.17 ± 0.05

Table 2 - Mean roughness (\pm Standard Deviation) for R_a and R_t parameters at zero, 5, and 10 million cycles.



Figure 1 – The picture shows the weight loss (\pm standard deviation) for five different polyethylenes. The cross-linked PE wore less than the other ones. Between 5 and 10 million cycles, wear reduced with respect the initial trend. The PE-B continued to show the worst wear behavior.



Figure 2 – Weight loss of the different polyethylenes vs. R_a from zero to 10 Mc. The solid line represents a simple linear regression fit, whereas the points represent each data value.



Figure 3 – Weight loss of the different polyethylenes vs. Rt from zero to 10 Mc. The solid line represents a simple_linear regression fit, whereas the points represent each data value.



 $\label{eq:Figure 4} Figure \ 4-FTIR \ spectra \ of \ the \ most \ worn \ polyethylene \ acetabular \ cups.$