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Hydrogen storage of Mg–Zn mixed metal borohydrides

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Abstract

On the basis of DFT results on the ideal amount of Zn required to dissolve into the α -phase of $\text{Mg}(\text{BH}_4)_2$ (space group P6122) and experimental results recently presented in the literature, mixtures of α - $\text{Mg}(\text{BH}_4)_2$ and ZnCl_2 have been prepared by ball-milling, reactive milling and cryo-milling in Mg:Zn starting ratios of 1:0.33, 1:0.2 and 1:0.1. Samples have been characterized with different experimental techniques (PXRD, TGA, TPD with RGA and desorption kinetics in a Sieverts apparatus). All the as-milled samples exhibited a H_2 release beginning approximately at 125 °C, indicating a significant decrease in the decomposition temperature compared to pure $\text{Mg}(\text{BH}_4)_2$. With the exception of the $\text{Mg}(\text{BH}_4)_2$ -0.33 ZnCl_2 sample ball-milled under 1 bar Ar for 3 h that showed a small B_2H_6 release around 120 °C, all samples are free of B_2H_6 emissions. Results from desorption kinetics at 195 °C suggest a strong relation between the amount of ZnCl_2 in the starting mixture and the weight loss.

1. Introduction

$\text{Mg}(\text{BH}_4)_2$ is an interesting candidate material for hydrogen storage applications due to its high gravimetric capacity of 14.9 H_2 wt% [1]. Recently, it was demonstrated that depending on the $\text{Mg}(\text{BH}_4)_2$ polymorph, the total H_2 content can be further increased by adding the amount of H_2 that physisorbs under cryogenic conditions, resulting in a total capacity up to 17.9 wt% [2]. Despite its high gravimetric capacity, $\text{Mg}(\text{BH}_4)_2$ exhibits slow desorption kinetics and its decomposition temperature is still too high to allow coupling with Proton-Exchange Membrane Fuel Cells (PEM-FC) in hydrogen storage applications. Different methods have been attempted to destabilize pure complex hydrides: anion substitution, cation substitution with mixed metal borohydrides formation, nanoconfinement and formation of reactive hydride composites [3–5]. Very few examples exist in the literature regarding destabilization of $\text{Mg}(\text{BH}_4)_2$ by substitution of either Mg^{2+} or $\frac{1}{2}\text{BH}_4^-$. Mixing different ratios of $\text{Mn}(\text{BH}_4)_2$ and α - $\text{Mg}(\text{BH}_4)_2$ Mn–Mg substitution in α - $\text{Mg}(\text{BH}_4)_2$ up to 10 mol% at room temperature was observed [6]. Hino et al. observed halide substitution of $\frac{1}{2}\text{BH}_4^-$ by Cl or Br in the lattice of β - $\text{Mg}(\text{BH}_4)_2$ after annealing a milled mixture of α - $\text{Mg}(\text{BH}_4)_2$ and MgX_2 ($X = \text{Cl}, \text{Br}$) resulting in a reduction in the temperature of hydrogen desorption [7]. Recently, theoretical calculations combined with experimental work on Mg–Zn mixed borohydrides showed that α - $\text{Mg}(\text{BH}_4)_2$ can be efficiently destabilized by adding a suitable amount of Zn [8]. Selecting ZnCl_2 as Zn source and ball-milling it with α - $\text{Mg}(\text{BH}_4)_2$ in a Mg:Zn 1:0.7 ratio, we observed the formation of a mixed metal borohydride

with Zn^{2+} substituting Mg^{2+} in the α - $Mg(BH_4)_2$ lattice and forming a $Mg(1-x)Zn_x(BH_4)_2$ solid solution with $x = 0.4$. Theoretical calculations predicted a value of $x = 0.2$ as the optimum ratio for the mixed system, with an estimated enthalpy of decomposition close to $30 \text{ kJ mol}^{-1} H_2$, which could provide hydrogen sorption at room conditions. In this study, the destabilization effect of $ZnCl_2$ on α - $Mg(BH_4)_2$ is investigated by ball-milling the starting materials in Mg:Zn ratios of 1:0.10, 1:0.20 and 1:0.33. The as-milled systems are characterized by means of Powder X-ray Diffraction (PXD), Thermo-Gravimetric Analysis (TGA), Thermal Programmed Desorption (TPD) with Residual Gas Analysis (RGA) and the results are discussed and compared with the Mg:Zn 1:0.7 samples reported previously. Finally, the desorption kinetics of the systems at $195 \text{ }^\circ\text{C}$ is assessed with a Sieverts apparatus.

2. Experimental methods

Mixtures of α - $Mg(BH_4)_2$ (synthesized at Karlsruhe Institute of Technology, Karlsruhe, Germany) with $ZnCl_2$ (99% purity, Sigma Aldrich) were ball-milled either at room temperature or at liquid nitrogen temperature (cryo-milling). Room-temperature ball milling was carried out in a Fritsch Pulverisette 6 Monomill at 400 rpm either under inert (1 bar Ar for 3 h) or reactive (50 bar H_2 in an Evico Magnetics high-pressure vial for 12 h) atmosphere using stainless steel balls and vials (ball-to-powder ratio 100:1). Cryo-milling in Ar at liquid N_2 temperature was performed in a SPEX 6770 Freezer/Mill using a specially designed stainless steel vial, in order to minimize oxygen and nitrogen contaminations. The mass ratio between the stainless steel cylindrical impactor and the powder mixture was 15:1 and the impact frequency during milling was set at 30 Hz. All sample handling and preparation were carried out under inert Ar atmosphere in an MBraun Unilab glove box fitted with a recirculation system and gas/humidity sensors. Oxygen and water levels were kept below 1 ppm at all times.

Powder X-ray Diffraction (PXD) measurements were performed in a Debye–Scherrer geometry using a Panalytical X'Pert Pro Multipurpose Diffractometer, equipped with Ni filtered Cu source. Powder samples were sealed into boron silica glass capillaries of internal diameter 0.8 mm in nitrogen atmosphere, which give a broad peak in the pattern centered around $2\theta = 22^\circ$. Patterns were taken by spinning capillaries in a 2θ range $2^\circ - 100^\circ$ at a scanning step of 0.016° and with a scan speed of $0.010^\circ/\text{s}$. Rietveld refinement of PXD patterns has been carried out with MAUD software [9].

Thermo Gravimetric Analysis (TGA) experiments were performed with a Netzsch STA 449 F3 Jupiter instrument. Samples were measured in Al crucibles with pierced lids of the same material. The samples were heated between $30 \text{ }^\circ\text{C}$ and $600 \text{ }^\circ\text{C}$, with a heating rate of $2 \text{ }^\circ\text{C min}^{-1}$ under argon gas (50 ml/min). The measurements

were baseline corrected by the Proteus software package.

Temperature-Programmed Desorption (TPD) with Residual Gas Analysis (RGA) spectra were measured in an in-house developed setup under vacuum (10^{-5} mbar) and between RT and $600 \text{ }^\circ\text{C}$ with a constant heating rate of $2 \text{ }^\circ\text{C min}^{-1}$. Simultaneous RGA was measured with a MULTIVISION IP mass spectrometer detector system coupled to a PROCESS Eye analysis package from MKS Instruments.

Dehydrogenation and hydrogenation kinetic measurements were carried out in a calibrated in-house built volumetric apparatus. Approximately 200 mg of sample

was inserted in a sample holder and placed under static vacuum. The sample holder is coupled with a pneumatic valve that keeps the sample in Ar during its transport from the glove box to the Sieverts apparatus. During dehydrogenation, each sample was heated from room temperature up to 195 °C at 20 °C min⁻¹ and then remained under isothermal conditions for 24 h. Hydrogenation took place after the end of dehydrogenation without evacuating the sample-holder and under the same isothermal conditions. For each hydrogenation experiment the H₂ pressure was 95 bar. Ultra-pure 6.0 grade H₂ was used.

3. Results and discussion

On the basis of calculations recently presented in the literature [8], mixtures of a-Mg(BH₄)₂ and ZnCl₂ have been ball-milled in different ratios. Pure compounds with a Mg:Zn ratio 1:0.33 have been ball-milled at room temperature under 1 bar Ar for 3 h (S1) and under 50 bar H₂ for 12 h (S2). Mixtures with Mg:Zn ratios of 1:0.2 (S3) and 1:0.1 (S4) were cryo-milled for 1 h.

Fig. 1 reports the PXD patterns for samples S1, S3 and S4. The patterns of pure a-Mg(BH₄)₂ and ZnCl₂ are also provided for reference. Prolonged milling (3 h in Ar–S1 or 12 h in H₂–S2) results in the conversion of the starting materials to Zn and MgCl₂. This is clearly seen in the PXD pattern of S1 in Fig. 1. A small fraction of unreacted a-Mg(BH₄)₂ is also observed. The PXD pattern of S2 (not shown) displays similar features. The same behavior was reported for the a-Mg(BH₄)₂ + ZnCl₂ mixture with a 1:0.7 Mg:Zn ratio and ball milled in Ar for 12 h [5]. The PXD patterns of S3 and S4 (Fig. 1) are dominated by the starting materials. In those samples, a partial amount of a-Mg(BH₄)₂ and ZnCl₂ has reacted towards formation of Zn and MgCl₂, similar to S1. The Rietveld refinement of PXD patterns of S3 and S4 samples provided accurate values of lattice parameters for a-Mg(BH₄)₂ phase, as reported in Table 1. Results indicate a reduction of unit cell parameters with respect to pure a-Mg(BH₄)₂, suggesting a small substitution of Mg²⁺ by Zn²⁺ in the borohydride lattice with formation of a Mg(1-x)Znx(BH₄)₂ solid solution. From the observed volume contractions (see Table 1) compared to those calculated for solid solutions as a function of composition [5], a value of x around 0.10 and 0.06 is estimated for samples S3 and S4, respectively. These values are much smaller than those observed for different Mg:Zn ratios, as reported in Ref. [5]. Finally, it should be noted that anion substitution, that is Cl⁻ replacing ½BH₄⁻ anions, cannot be ruled out. However, considering the ratio between anionic radii of ½BH₄⁻ (0.203 nm) and Cl⁻ (0.171 nm), a stronger reduction of the cell volume would be expected. The contraction of the unit cell observed experimentally is therefore more likely related to a cation rather than an anion substitution and the formation of a Mg(1-x)Znx(BH₄)₂ solid solution. The results of TGA measurements on as-milled samples are illustrated in Fig. 2, where the mass loss profile obtained for pure a-Mg(BH₄)₂ is also reported for reference. All the as-milled samples show mass-loss events which begin at approximately 125 °C and continue up to the end of the temperature scale at 600 °C. Samples S1 and S2 show a similar mass-loss profile, suggesting a

2-step decomposition mechanism with individual steps occurring in the temperature ranges of 150–180 °C and 230–320 °C. S1 and S2 have a maximum mass-loss of 7 and 6 wt%, respectively. This result suggests that H₂ and/or B₂H₆ were released during the prolonged milling, similarly to the sample with Mg:Zn 1:0.7 ballmilled under the same experimental conditions used for S2 [8].

Sample S3 shows a 1-step mass-loss mechanism that yields 3 wt% at 250 °C and 11 wt% at 450 °C. A similar behavior is observed for S4, exhibiting 4 and 14.5 wt% at 250 °C and 450 °C, respectively. The mass loss profile is similar to that of the pure borohydride with a clear shift towards lower temperatures. These results suggest that small amount of ZnCl₂ in the initial mixture is enough to initiate the gas release at 125 °C without compromising significantly the total H₂ capacity of the borohydride.

In order to identify the nature of the gas released by the samples during heating and to make sure that the mass-loss observed can be indeed ascribed to H₂ desorption, TPD-RGA measurements were conducted and the results are reported at Fig. 3. Only data for mass 2 (H₂) and mass 26 (B₂H₆) are illustrated due to the negligible intensity observed for other masses. Fig. 3a shows the results obtained for pure α-Mg(BH₄)₂ confirming the stepwise desorption observed in the TGA profile (Fig. 2). The delivery of H₂ begins at 190 °C, becomes more significant at 250 °C and eventually the main desorption peak is observed at 300 °C. At the same time, no B₂H₆ is detected in the released gas. On the other hand, the ball-milled mixed samples show a hydrogen release that begins at much lower temperatures than for pure α-Mg(BH₄)₂ and depends on the amount of ZnCl₂ contained in the starting mixture. Sample S1 (Fig. 3b) exhibits a wide H₂ release distributed from 100 up to 400 °C, that is the widest temperature range of H₂ release among the examined samples. At 100 °C a bump is observed for both mass 2 and mass 26 suggesting a limited B₂H₆ release. This is the only instance among the samples investigated in this work where B₂H₆ release is detected. Two main desorption peaks are seen at 180 °C and 280 °C, in very good agreement with the TGA results. H₂ desorption peaks occurring at the same temperature range as at S1 are observed for S2 (Fig. 3c). H₂ release begins at 100 °C, however it is more intense between 150 and 200 °C. No B₂H₆ release was detected throughout the heating treatment. Finally, samples S3 and S4 clearly show one main desorption event at 283 and 288 °C, respectively (Fig. 3d and e, respectively). The minor desorption peaks observed for pure α-Mg(BH₄)₂ are missing, suggesting an alteration in the decomposition path of the borohydride. For both samples, the intensity of signal for mass 26 was negligible in all the temperature range.

Finally, in order to quantify the effect of the different amounts of ZnCl₂ in the parent mixture on the desorption kinetics of α-Mg(BH₄)₂, desorption measurements were carried out in a Sieverts apparatus. Fig. 4 shows the desorption profile for the pure α-Mg(BH₄)₂ and the as-milled samples, acquired at 195 °C. At this temperature, pure α-Mg(BH₄)₂ releases 0.5 wt% after 24 h

(Fig. 4a). Sample S4 releases 1 wt%, that is double the amount released by the pure material (Fig. 4b). Under the same conditions, sample S3 releases 1.4 wt%, that is almost three times higher compared to the pure borohydride (Fig. 4c). Practically no difference is observed in the desorption kinetics of S1 and S2, with both samples delivering 2.8 wt% after 24 h at 195 °C (Fig. 4d and e, respectively). Following the dehydrogenation, each sample was hydrogenated at the same temperature (195 °C) under 95 bar H₂ for 48 h without any evident absorption taking place.

The results above show that a careful selection of milling conditions and milling time is crucial for producing Mg(1-x)Zn_x(BH₄)₂ solid solutions from Mg(BH₄)₂ and ZnCl₂. If the milling is performed at room temperature and for an extended time (over 3 h), the starting materials react to form MgCl₂ and Zn with H₂ and B₂H₆ likely being released during milling. The resulting composites (S1 and S2 in this work) contain residual Mg(BH₄)₂ which desorb H₂ and in some cases B₂H₆. The desorption temperature is much lower than that for pure Mg(BH₄)₂ and the desorption kinetics is faster. However, the amount of hydrogen released during desorption is significantly less than the theoretical capacity of the starting mixtures. A reduced H₂ capacity and the B₂H₆ release are major setbacks for practical hydrogen storage applications.

Reducing the milling time down to 1 h and performing milling at liquid nitrogen temperature limits the extent of the reaction between the starting Mg(BH₄)₂ and ZnCl₂. Although limited in time, milling still promotes an intimate mixing of the starting materials which results in the formation of Mg(1-x)Zn_x(BH₄)₂ solid solutions (x = 0.10 and x = 0.06 for S3 and S4, respectively). At the same time, in analogy with the synthesis by ball milling of other hydrides [10], the low temperature maintained during milling likely prevents the decomposition of the newly formed solid solutions. The composites containing Mg(1-x)Zn_x(BH₄)₂ solid solutions show the best hydrogen desorption performances among the samples investigated in this work. A relative fast desorption kinetics is coupled with desorption temperatures which are significantly reduced when compared to pure Mg(BH₄)₂, without compromising the weight capacity. In addition, no B₂H₆ is detected in the released gas. This work confirms the destabilization effect reported previously for cryo-milled Mg(1-x)Zn_x(BH₄)₂ with x = 0.4 and predicted by the theoretical calculations on the dissolution of Zn in Mg(BH₄)₂. It is worth noting that the optimal destabilization, with an estimated enthalpy of decomposition close to 30 kJ mol⁻¹ H₂, is predicted for x = 0.2, whereas experimental results are only shown for x = 0.06 and 0.10 (this work) and x = 0.40 [8]. This discrepancy is due to the difficulty of accurately tune the degree of Zn substitution in the final solid solution, since a fraction of the starting ZnCl₂ is either left unreacted or transformed to metallic Zn. Therefore, the relative content of Zn in the starting mixture does not correspond to the actual content of Zn found in the Mg(1-x)Zn_x(BH₄)₂ solid solution after cryo-milling, as shown in Fig. 5 where the amount of Zn in the solid solution is plotted against the Mg:Zn molar ratio

in the starting materials for the cryo-milled samples. The data for Mg:Zn ratio 0.7 come from Ref. [8]. An almost linear relation between the starting Zn content and the Zn substituted in $\text{Mg}(\text{BH}_4)_2$ can be noticed in Fig. 5. From such a linear relation, we should expect the formation of a $\text{Mg}(1-x)\text{Zn}_x(\text{BH}_4)_2$ solid solution with $x = 0.2$ for a Mg:Zn starting ratio of about 1:0.35.

In spite of the beneficial effect that the dissolution of Zn has on the hydrogen desorption in $\text{Mg}(\text{BH}_4)_2$, the reversibility of hydrogen sorption in the $\text{Mg}(1-x)\text{Zn}_x(\text{BH}_4)_2$ solid solution still remains unfulfilled. Since no B_2H_6 was detected in the released gas, we can ascribe the irreversibility to the formation of stable B-containing phases.

4. Conclusions

On the basis of DFT results on the ideal amount of Zn required to dissolve into the α -phase of $\text{Mg}(\text{BH}_4)_2$ and experimental results that we recently presented in the literature, mixtures of α - $\text{Mg}(\text{BH}_4)_2$ and ZnCl_2 have been processed by ball-milling, reactive milling and cryo-milling in Mg:Zn ratios of 1:0.33, 1:0.2 and 1:0.1, respectively. All the as-milled samples exhibited mass loss beginning at 125 °C. Rietveld refinement showed the formation of Mg–Zn mixed metal borohydride. With the exception of the $\text{Mg}(\text{BH}_4)_2$ –0.33 ZnCl_2 sample ball-milled under 1 bar Ar for 3 h that showed a small B_2H_6 release around 120 °C, all the samples were found free of B_2H_6 emissions. Desorption kinetics obtained at 195 °C suggest a strong relation between the amount of ZnCl_2 in the starting mixture and the weight loss at this temperature. It was found that ZnCl_2 can significantly destabilize the α - $\text{Mg}(\text{BH}_4)_2$ and create systems that in most cases are free of B_2H_6 emissions during the thermal decomposition.

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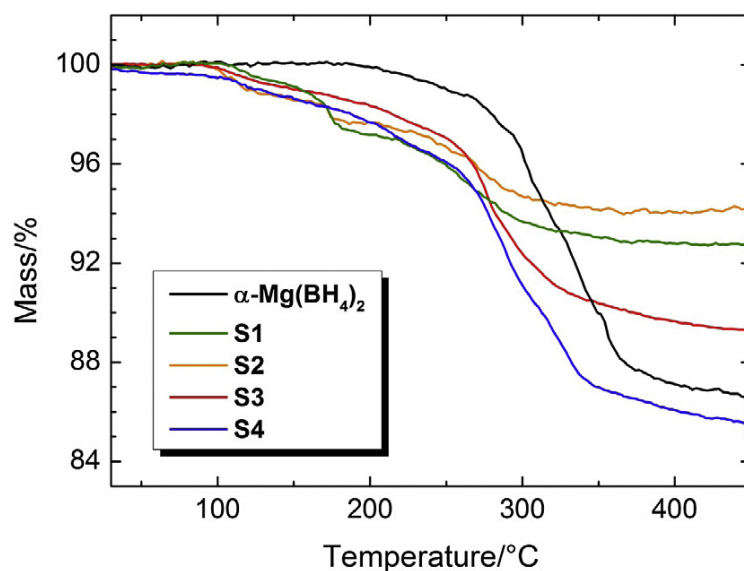
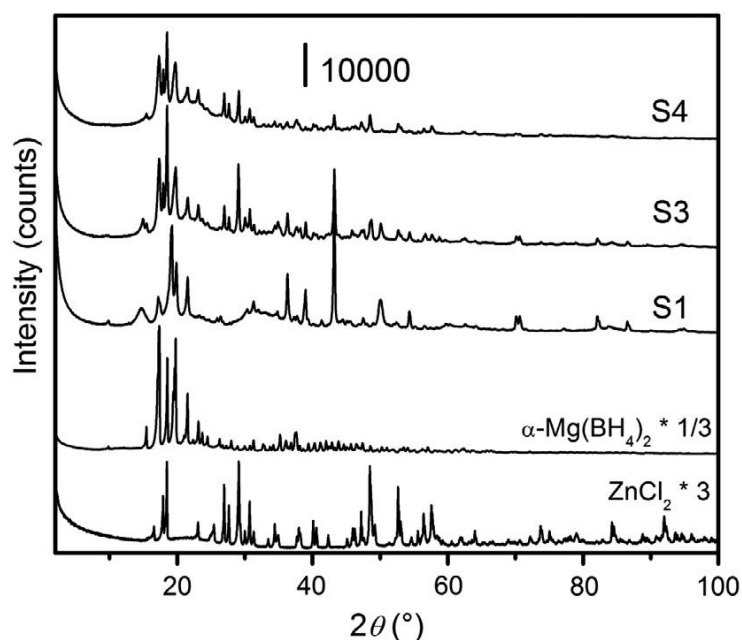
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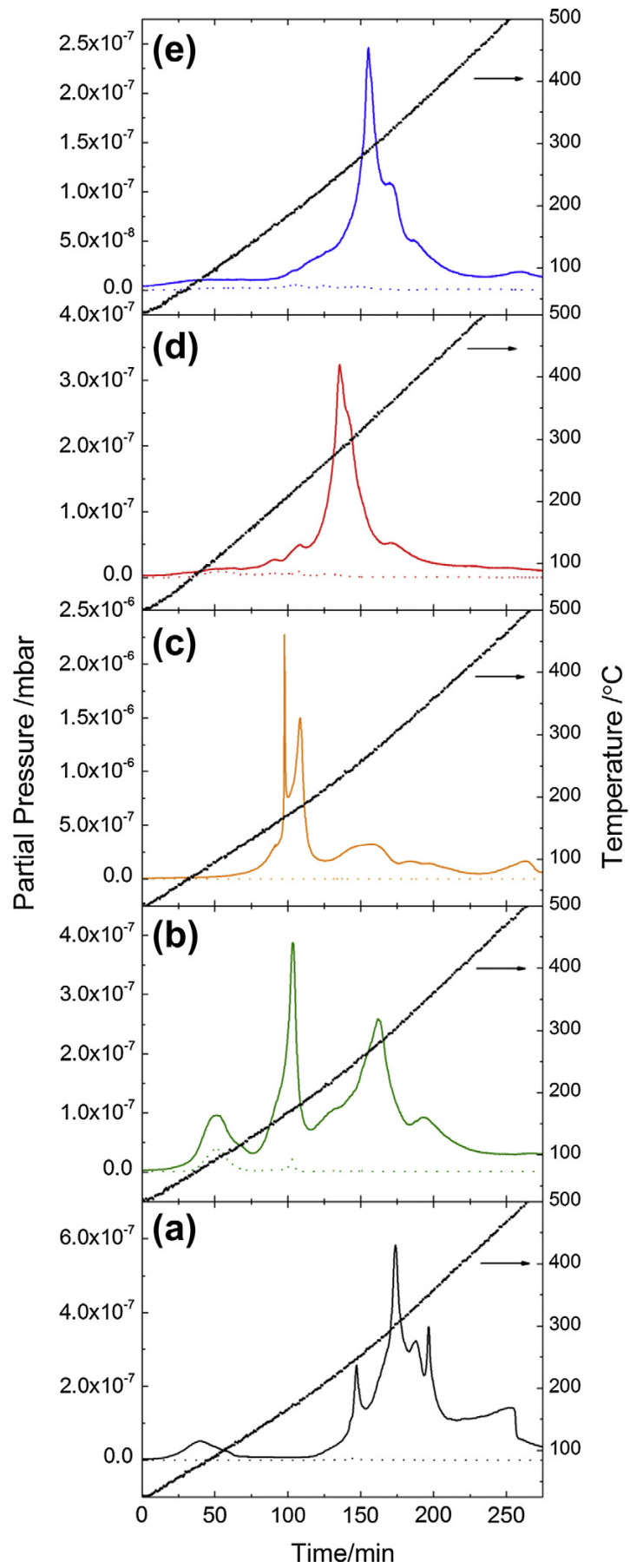
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Table 1
Lattice constants and unit cell volumes obtained from Rietveld refinement of PXD patterns.

Sample	Mg:Zn	$a = b$ (Å)	c (Å)	Vol (Å ³)	$-\Delta V$ (%)
α -Mg(BH ₄) ₂	0	10.336	37.089	3962	0
S3	0.2	10.299	36.997	3924	0.96
S4	0.1	10.306	37.086	3938	0.60
S3 from Ref. [8]	0.7	10.216	36.673	3827	3.41





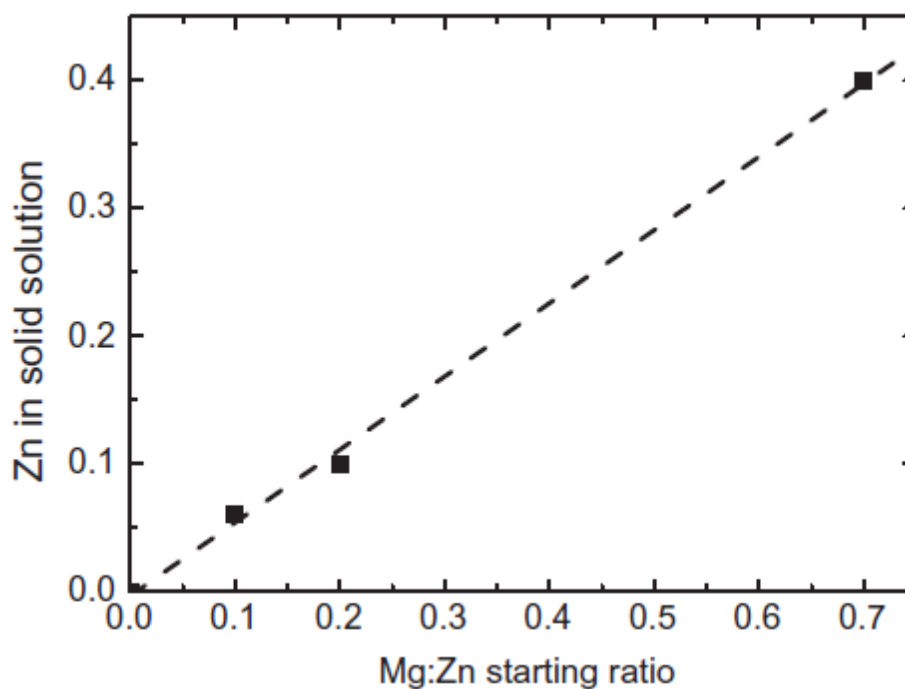
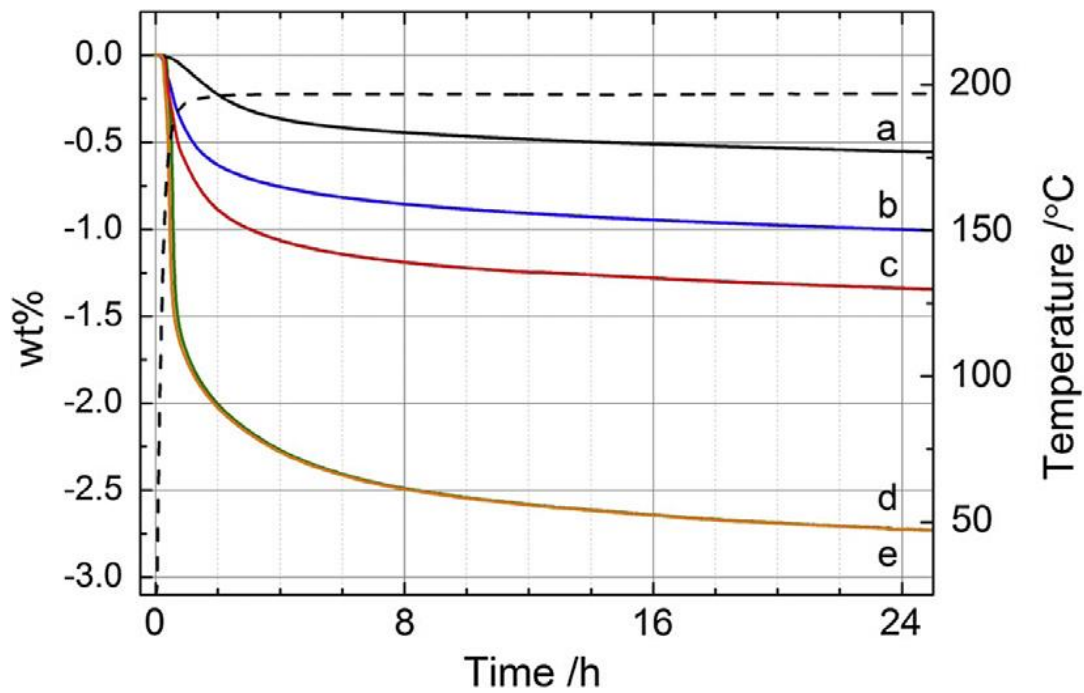


Figure captions

Fig. 1. X-ray diffraction patterns (Cu K α 1.540598 Å) of pure a-Mg(BH₄)₂, pure ZnCl₂ and of samples S1, S3 and S4.

Fig. 2. TGA profiles of pure a-Mg(BH₄)₂ (black) and of samples S1 (green), S2 (orange), S3 (red) and S4 (blue). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

Fig. 3. TPD-RGA spectra of pure a-Mg(BH₄)₂ (a), S1 (b), S2 (c), S3 (d) and S4 (e). Continuous line is used for mass 2 (H₂) and dotted line for mass 26 (B₂H₆). The

temperature ramps used for the desorption are indicated by continuous black lines.

Fig. 4. Desorption kinetics during the first 25 h at 195 C for pure α -Mg(BH₄)₂ (a), S1 (b), S2 (c), S3 (d) and S4 (e). The dashed line indicates the temperature ramp.

Fig. 5. Amount of Zn in the solid solution vs Mg:Zn molar ratio in the starting materials for the cryo-milled samples. The data for Mg:Zn ratio 0.7 are from Ref. [8].