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## High-pressure behaviour of zoisite

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#### Abstract

A high-pressure single-crystal X-ray diffraction (XRD) study has been carried out on two natural zoisite samples $\mathrm{Ca}_{2} \mathrm{Al}_{3-\mathrm{x}} \mathrm{Fe}_{x} \mathrm{Si}_{3} \mathrm{O}_{12} \mathrm{OH}$, one Fe -free $(x=0)$ and Fe -rich $(x=0.12)$. The unit-cell parameters were determined for the Fe-free sample at 18 different pressures up to 7.76 GPa and for the Fe -rich sample at 13 different pressures up to 7.63 GPa. The $P(V)$ data for both of the samples were fitted by a third-order Birch-Murnaghan equation of state (BM3 $\mathrm{EoS})$. The equation of state coefficients are: $V_{0}=903.39(5) \AA^{3}, K_{T 0}=122.1(7) \mathrm{GPa}$ and $K_{0}^{\prime}=6.8(2)$ for the Fe-free sample and $V_{0}=906.95(5) \AA^{3}, K_{T 0}=119.1(7) \mathrm{GPa}$ and $K_{0}^{\prime}=7.3(2)$ for the Fe-rich sample. This shows that the addition of Fe in to the crystal structure of zoisite leads to a slight softening of the structure

Both compositions exhibit axial compressibilities $\beta_{\mathrm{c}}>\beta_{\mathrm{a}} \gg \beta_{\mathrm{b}}$, with the compressibilities of the $a$ and $b$ axes of the two samples being indistinguishable. The softening of the bulk modulus of zoisite with Fe content follows from softening of the $c$-axis of the structure. A high-pressure structural study of the Fe -free sample showed that the main compression mechanisms in the structure are the compression of thesoft inter-octahedral distance along [001] and soft intra-octahedral distances along [010] directions, while along [100] the main compression occurs because of the compression of the-stiff intra-octahedral distances. The substitution of Fe on to the $M 3$ octahedral site of the structure leads to an increase of the intra-octahedral distance of the $M 3$ that triggers the rotation of M12 and therefore leads to the softening of the M12 inter-octahedral distances that accounts for the softening of the $c$-axis of the structure.


## INTRODUCTION

Several authors (i.e. Forneris and Holloway, 2003; Kirby et al. 1996) have argued that the correct model for explaining the generation of many volcanisme are and intermediate earthquakes in subduction zones involves the progressive dehydration of the subducting slab through a series of reactions that release $\mathrm{H}_{2} \mathrm{O}$ into the mantle wedge. These fluids released from subducting slabs could trigger hydration-driven partial melting reactions, inducing partial melting of the mantle wedge above the slab, which in turn is considered responsible for the intermediate earthquakes (i.e. Peacock, 2001 and Kirby et al. 1996), as well as for arc volcanism (i.e. Forneris and Holloway 2003) in subduction zones. A key role in these dehydration reactions is played by the hydrous mineral phases that are mainly contained in the subducting slab. Therefore a detailed study of the properties of the hydrous phases, and especially the thermo-elastic behavior, will be an important requirement for understanding properties and processes within subduction zones, in particular for constraining their stability and the related dehydration reactions in which they are involved as a function of pressure and temperature (i.e. Hacker et al. 2003 and Mao et al. 2007). Minerals of the epidote group for example, participate in numerous relevant phase equilibria, which need to be accurately evaluated in order to understand these geological processes. Zoisite belongs to the epidote group and occurs in high and ultrahigh-pressure metamorphic rocks from a wide variety of geological settings, including continental collisions and subduction zones (i.e. Hacker et al. 2003; Mao et al. 2007 and Enami et al. 2004).

Zoisite is the only member of the epidote group that is orthorhombic, instead of monoclinic. The structure of zoisite (Figure 1) was firstly determined by Fesenko et al. (1955), Fesenko et al. (1956), and was later refined by Dollase (1968). Monoclinic epidotes, including clinozoisite, have two distinct edge-sharing octahedral chains that run along [010]. Zoisite has only one type of octahedral chain parallel to [010] built up of edge sharing octahedra designated M12 and occupied by $\mathrm{Al}^{3+}$ (Ghose and Tsang, 1971). That chain of octahedrat is decorated by a second octahedral site, M3, which can be occupied by both $\mathrm{Al}^{3+}$ and $\mathrm{Fe}^{3+}$ (Ghose and Tsang 1971). This site is more distorted and is attached to M12 octahedra by edge sharing. The octahedral chains are linked by isolated $\mathrm{SiO}_{4}$ tetrahedra (T3) in the $\mathbf{c}$ direction and by $\mathrm{Si}_{2} \mathrm{O}_{7}$ groups ( $T 1$ and $T 2$ ) in the $\mathbf{a}$ and $\mathbf{c}$ directions (see Fig. 1). In this framework of interconnected octahedral chains and bridging tetrahedra there are two distinct irregularly shaped, sevenfold-coordinated cavities (Ca1 and Ca2) occupied by Ca . Hydrogen is bonded to oxygen O10 which coordinates cations in the octahedral chains (Franz and Liebscher 2004).

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Several authors have measured the equation of state of zoisite by means of X-ray diffraction on samples with different composition and have obtained several different values for the bulk moduli; 125.1(2.1) GPa was measured by Grevel et al. (2000) on a synthetic zoisite sample with composition $\mathrm{Ca}_{2} \mathrm{Al}_{3} \mathrm{Si}_{3} \mathrm{O}_{12} \mathrm{OH} ; 127(4) \mathrm{GPa}$ was measured by Pawley et al. (1998) on a natural zoisite sample containing 2.3-4.3\% of ferric end member; and 102(7) GPa was obtained by Comodi and Zanazzi (1997) on a zoisite sample with composition $\mathrm{Ca}_{2} \mathrm{Al}_{2}\left(\mathrm{Al}_{0.9} \mathrm{Fe}_{0.1}\right) \mathrm{Si}_{3} \mathrm{O}_{12} \mathrm{OH}$. Recently Mao et al. (2007) measured the elastic constants at ambient conditions by Brillouin scattering on a Fe-free crystal of zoisite from Merelani Hills, to obtain a value of $K_{\mathrm{S} 0}=125.3(4) \mathrm{GPa}$ which, however, is not in agreement with the compression studies, as we discuss below.

In order to clarify the discrepancies among the literature data, and to determine the effect of the substitution of Fe on the bulk modulus, we investigated the high pressure behavior of two samples with different Fe contents, Fe-free $\left(\mathrm{X}_{\mathrm{ps}}=0\right)$ and Fe -rich ( $\mathrm{X}_{\mathrm{ps}}=0.12$ ) up to 7.76 and 7.63 GPa respectively, by means of single-crystal X-ray diffraction. We determined the unit-cell parameter evolution on both samples in order to calculate the equation of state and we measured intensity data on the Fe -free sample to follow the evolution of the structure with pressure.

## EXPERIMENTAL METHODS

## Samples

Two single crystals of natural zoisite with general formula $\mathrm{Ca}_{2}\left(\mathrm{Al}, \mathrm{Fe}_{3}\right)_{3} \mathrm{Si}_{3} \mathrm{O}_{12}(\mathrm{OH})$, and different Fe contents were selected. The Fe -free $\left(\mathrm{X}_{\mathrm{ps}}=0\right)$ sample is from the Merelani Hills in the Arusha Region, United Republic of Tanzania (Barot and Boehm 1992) and was kindly provided by A. Zanetti. The Fe-rich sample ( $\mathrm{X}_{\mathrm{ps}}=0.12$ ) is from Carinthia, Austria (sample number 8952 from the Museo di Mineralogia, Dipartimento di Scienze della Terra, Pavia, Italy).

## Electron microprobe analysis

Chemical analyses were performed on different crystals from the same samples that X-ray data collections and structure refinements at ambient conditions showed had very similar structural parameters. Analysis-Analyses of the Fe-free sample $\left(X_{\mathrm{ps}}=0\right)$ sample are is from Cámara et al.
(2011). A Cameca-Camebax electron microprobe with a fine-focused beam ( $1 \mu \mathrm{~m}$ diameter) operating in the wavelength-dispersive (WDS) mode was used. Operating conditions were 20 kV accelerating voltage and 20 nA beam current for for $\mathrm{Al}, \mathrm{Si}, \mathrm{K}, \mathrm{Ca}$ and Fe and 20 kV and 40 nA for $\mathrm{Na}, \mathrm{Mg}, \mathrm{Ti}, \mathrm{V}, \mathrm{Mn}, \mathrm{Sr}, \mathrm{La}$ and Ce , with a beam diameter of $20 \mu \mathrm{~m}$. The following standards were used (spectral line, analyzing crystal): Amelia albite ( $\mathrm{NaK} \alpha, \mathrm{TAP}$ ), wollastonite ( $\mathrm{CaK} \alpha, \mathrm{PET}$ ), diopside ( $\mathrm{Si} K \alpha, \mathrm{TAP}$ ), MgO ( $\mathrm{Mg} K \alpha, \mathrm{TAP}$ ), orthoclase ( $\mathrm{K} K \alpha, \mathrm{PET}$ ) $\mathrm{MnTiO}_{3}$ ( $\mathrm{Mn} K \alpha$ and $\mathrm{Ti} K \alpha$, PET), vanadinite (VK $\alpha$, LIF), $\mathrm{Fe}_{2} \mathrm{O}_{3}(\mathrm{Fe} K \alpha, \mathrm{LiF})$, corundum (AlK $\alpha, \mathrm{TAP}$ ), celestine (SrL $\alpha$, PET), and REE glass 3 (LaLa and CeLa, LIF). WDS scans showed the absence of fluorine. Data were corrected using the PAP method of Pouchou and Pichoir, (1984). The empirical formulae of our zoisite samples are $\left(\mathrm{Ca}_{1.99} \mathrm{Sr}_{0.01}\right)\left(\mathrm{Al}_{2.99} \mathrm{~V}_{0.01}\right)\left[\mathrm{Si}_{2} \mathrm{O}_{7}\right]\left[\mathrm{SiO}_{4}\right] \mathrm{O}(\mathrm{OH})$ and $\left(\mathrm{Ca}_{1.99} \mathrm{Sr}_{0.01}\right)\left(\mathrm{Al}_{2.88} \mathrm{Fe}_{0.12}\right)$ $\left[\mathrm{Si}_{2} \mathrm{O}_{7}\right]\left[\mathrm{SiO}_{4}\right] \mathrm{O}(\mathrm{OH})$, respectively with the OH content being assumed by stoichiometry as the crystals are fluorine-free (Table 1).

## High-pressure experiments

Two crystals, one Fe-free (size $0.150 \times 0.100 \times 0.080 \mathrm{~mm}$ ), and one Fe-rich (size $0.110 \times$ $0.090 \times 0.060 \mathrm{~mm}$ ) were selected for the high-pressure study on the basis of their size, optical sharp extinction, absence of twinning, and X-ray diffraction profiles. The Fe-free and Fe-rich crystals were loaded in two different ETH-type diamond-anvil cells (DACs, Miletich et al. 2000) using a steel gasket (T301) pre-indented to a thickness of $100 \mu \mathrm{~m}$ and with a hole of $250 \mu \mathrm{~m}$ in diameter for both the experiments. A single crystal of quartz was used as an internal diffraction pressure standard (Angel et al. 1997) and a 4:1 mixture of methanol:ethanol was used as pressure medium, which remains hydrostatic throughout the pressure range investigated in this work (Angel et al. 2007). The unit-cell parameters were determined by single-crystal X-ray diffraction using a Huber four-circle diffractometer (non-monochromatised Mo-K $\alpha$ radiation) operating at 50 kV and 40 mA , automated
by the SINGLE software (Angel and Finger 2011). The unit cell parameters were measured at 18 different pressures up to $\sim 7.76 \mathrm{GPa}$ for the Fe-free sample and at 13 different pressures, up to $\sim 7.63$ GPa ; for the Fe -rich sample by centering not less than 20 reflections in the $2 \theta$ range between $10^{\circ}$ and $30^{\circ}$ for each high-pressure data point. Typical half-widths of reflections were between $0.05^{\circ}$ and $0.08^{\circ}$ in $\omega$ and no broadening was detected at any pressure. Full details of the instrument and the peak-centering algorithms are provided by Angel et al. (1997). During the centering procedure the effects of crystal offsets and diffractometer aberrations were eliminated from refined peak positions by the eight-position centering method of King and Finger (1979). Unconstrained unit-cell parameters, obtained by vector least-squares (Ralph and Finger 1982), were found to be similar, within one estimated standard deviation, to the symmetry-constrained ones which are reported in Table 2.

Intensity data for structural refinements were collected simultaneously for the Fe-free crystal on an Xcalibur-1 Oxford Diffraction diffractometer equipped with a point detector (kappageometry, graphite-monochromatised $\mathrm{Mo}-\mathrm{K} \alpha$ radiation). Integrated intensity data were then corrected for absorption effects due to the crystal and the DAC using the ABSORB computer program (Burnham 1966; Angel 2004).

All the structure refinements were performed with the SHELX-97 software (Sheldrick, 2008), starting from the atomic coordinates of Cámara et al. (2011). Because of the complexity of the structure and the limited number of observations, all of the refinements were performed with isotropic displacement parameters for all atoms. Details of the structure refinement, bond distances and angles are reported in Tables 3 and 4. Atomic coordinates and displacement parameters are listed in Table 5 (deposited), and CIFs are deposited.

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## RESULTS AND DISCUSSION

## Elastic behavior

The evolution of the unit-cell parameters and unit-cell volume of the two crystals measured in this study as a function of pressure is shown in Figure 2, along with the previously-published data. All of the unit cell parameters undergo a strong non-linear decrease up to 7.77 GPa and 7.63 GPa for the Fe-free and Fe-rich samples respectively (Figure $2 \mathrm{a}, \mathrm{b}, \mathrm{c}, \mathrm{d}$ and Table 2). Over this pressure range the $a, b$ and $c$ lattice parameters decrease by $1.07 \%, 1.90$ and $2.29 \%$ respectively for the Fe-free sample and by $1.03,1.89$ and $2.33 \%$ respectively for the Fe-rich sample. The unit-cell volume decreases non-linearly at an average rate of $6.06 \AA^{3} / \mathrm{GPa}$ for the Fe-free sample and $6.13 \AA^{3} / \mathrm{GPa}$ for the Fe-rich sample. The compression of all of the unit cell parameters is accompanied by significant stiffening of the $b$ and $c$ axes while the $a$ axis shows marginal softening. The lattice parameter variation of both of the samples follows the same compressibility scheme, and for both the compression is completely reversible, with no indication of any change in elastic behavior upon decompression, but for the Fe-free sample there is significant scatter in the $f$ - $F$ plots of the unit-cell parameters in the pressure range 4-4.5 GPa (Figure 3). This appears to be associated with a subtle change in the structural evolution with pressure, as there is no discontinuity in the $P-V$ curve nor is there a change in the slope of the $f$ - $F$ plot of the volume compression (Figure 3d) at this pressure (see discussion).

The $P(V)$ data were fitted, for both Fe-free and the Fe-rich sample, by a third-order (BM3EoS) Birch-Murnaghan equation of state (Birch 1947), using the EOSFIT-5.2 software (Angel, 2000). The EoS coefficients of the two samples are reported together in Table 6. The values of the bulk modulus $\left(K_{\text {TOFe-free }}=122.1(7) \mathrm{GPa}\right.$ and $\left.K_{\text {TOFe-rich }}=119.1(7) \mathrm{GPa}\right)$, and its first pressure derivative $\left(K{ }^{\prime}{ }_{\text {Fe-free }}=6.8(2)\right.$ and $\left.K^{\prime}{ }_{\text {Fe-rich }}=7.3(2)\right)$ are both in excellent agreement with those indicated by the normalised stress $F_{\mathrm{E}}-$ Eulerian strain $f_{\mathrm{E}}$ plot as shown in Figure 3 (Angel, 2000; $F_{\mathrm{E}}$

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$\left.=P / 3 f_{\mathrm{E}}\left(1+2 f_{\mathrm{E}}\right)^{5 / 2}, f_{\mathrm{E}}=\left[\left(V_{0} / V\right)^{2 / 3}-1\right] / 2\right)$. The bulk moduli obtained from isothermal hydrostatic
compression measurements such as those reported here correspond to the Reuss bound on the isothermal bulk modulus. The values of the elastic moduli tensor of end-member zoisite (Mao et al. 2007) yield a value of the Reuss bound of the adiabatic modulus of 121.6(4) GPa which, using their value for the factor $(1+\alpha \gamma \mathrm{T})$ of 1.016, corresponds to an isothermal bulk modulus of 119.7(4) GPa, which is in between our values. The apparent agreement reported by Mao et al. (2007) between their Brillouin data and the previous compression studies of Grevel et al. (2000) and Pawley et al. (1998) was an artifact of Mao et al. (2007) incorrectly using the VRH average modulus instead of the Reuss bound as a basis for comparison with hydrostatic compression experiments. The slightly higher value of $K_{\mathrm{T} 0}=125(2)$ for end-member zoisite obtained by Grevel et al. (2000) is probably a consequence of their fixing the value of $K^{\prime}$ to 4 . Considering the data scatter in the other published studies (Figure 2) agreement with any EoS parameters derived from them is fortuitous, and depends on the assumptions made about constraining other EoS parameters such as $\mathrm{V}_{0}$ and $K^{\prime}$.

The axial compressibilities at room pressure for $a, b$, and $c$ (Table 7) were obtained using a parameterised form of the BM3-EoS, in which the individual axes are cubed, following the procedure implemented in the EoS-FIT5.2 software (Angel 2000). Note that the resulting moduli given by EosFit (Table 7) must be multiplied by a factor of 3 to obtain the linear moduli equal to the inverse of the linear compressibilities defined as $\beta_{l 0}=\frac{-1}{l}\left(\frac{\delta l}{\delta P}\right)_{P=0}$ which is in turn equal to the sum of the three compressional terms in the elastic compliance matrix for the axis. Thus, for example, the compressibility of the $a$-lattice parameter of zoisite is given by $\beta_{1}=s_{11}+s_{12}+s_{13}$. The resulting linear compressibilities for the Fe-free sample in this study (Table 87) agree within the uncertainties with those determined from the individual values of the compliance matrix inverted from the $c_{\mathrm{ij}}$ values given by Mao et al. (2007). The compressibility scheme is $\beta_{\mathrm{a}} \ll \beta_{\mathrm{b}}<\beta_{\mathrm{c}}$ for both samples, $\left(\beta_{a}: \beta_{b}: \beta_{c}=1.00: 2.29: 2.74\right.$ for the Fe-free sample and $\beta_{a} \lll \beta_{b}<\beta_{\epsilon}-\left(\beta_{a}: \beta_{b}: \beta_{c}=\right.$ 1.00:2.34:2.99 for the Fe-rich sample) with $\boldsymbol{a}$ being the most rigid direction and $\boldsymbol{c}$ the softest. Although the compressibility scheme shows the same general pattern for both samples, and the

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compressibilities of the $\boldsymbol{a}$ and $\boldsymbol{b}$ directions do not change between the two, the increased Fe content leads to a significant softening of $\boldsymbol{c}$, and it is this axial softening that results in the reduction of bulk modulus with the addition of Fe to the structure. The same general pattern of axial compressibilities is apparent in the previous experimental studies (Figure 2), although the compressibility of $\boldsymbol{a}$ was significantly over-estimated by Pawley et al. (1998) and no meaningful values of the linear compressibilities can be obtained from those data.

## Structure evolution with pressure

In order to better understand the reasons of the change in the compressibility of zoisites with Fe content, and why the biggest change in compressibility is along the $c$-axis, in the following section we describe briefly the structural evolution of the Fe-free crystal with pressure. The compression of the bond lengths and bond angles at the bridging oxygen atoms alone is not sufficient to explain either the compression mechanism in each direction, nor the strong anisotropy of the structure itself. Therefore, distortion of the polyhedra and their cooperative rotations are the key to explain the main mechanisms that control the average compression of the whole structure.
a) Framework compression

During the pressure increase from room pressure up to 7.77 GPa the $a$-axis undergoes a compression of about $0.17 \AA$. The [100] direction is the least compressible of the whole structure with a bulk modulus of 244(5) GPa for the Fe-free sample and 248(4) for the Fe-rich one. The M3 octahedra share one oxygen atom (O8) with $T 1$ tetrahedra on one side while on the other side they share one edge ( $\mathrm{O} 1-\mathrm{O} 4$ ) with M12 octahedra, which also share an oxygen atom (O3) with Tl tetrahedra (see Figures 1a, b and c). These polyhedral linkages follow the path of the $2_{1}$ screw axes running parallel to $\left[\begin{array}{lll}1 & 0 & 0\end{array}\right]$ (Figure 1b) and therefore they span one half of the unit cell length (a/2) which compresses by $0.085 \AA$ over the pressure range studied. The contribution to the compression from the rotation and compression of M12 and M3 octahedra is equal to the shortening and rotation
of the vector along the $\mathrm{O} 8-\mathrm{O} 5(\mathrm{O} 8$ shared between $T 1$ and $M 3$ and O 5 shared between $T 1-T 3$ and M12 $(0.008 \AA))$ and the one of the vector O5-O3 $(0.028 \AA)$. The rotation of the M3 and M12 octahedra relative to one another shortens the M3-M12 cation - cation distance by $0.033 \AA$, and results in significant compression of the $T 3$ tetrahedra (from $2.236 \AA^{3}$ to $2.142 \AA^{3}$, see Table 4 and Figure 4) and increase of the distortion (the volume distortion parameter of Makovicky and Balić Žunić, (1998) increases from 1.43 to $1.94 \%$ ) of the $T 3$ tetrahedron (see Table 4 and Figure 5). The largest contribution from the framework to the shortening along [100] actually comes from the shortening and rotation of the O3- O8 edge of the $T 1$ tetrahedron that contributes as $0.051 \AA$ which also leads to a significant increase in distortion of the tetrahedra.

The lengths of the O-O edges shared between the consecutive M12 octahedra forming continuous chains along the [010] direction (see Figure 1c) are considerably shorter (by $0.2-0.3 \AA$ ) than the unshared edges, indicating significant cation-cation repulsion between the octahedral cations. One might therefore expect that the [010] direction would be the stiffest direction in the structure. That it is not indicates that compression and/or deformation of the octahedra is important in the compression of the structure. Each octahedron is related to the next in the chain by a mirror plane perpendicular to $b$ (shown in Figure 1c with a solid line), and therefore the compression of $b / 2$ along $1 / 2[010]$ must be accommodated within a single octahedron. Under compression to 7.6 GPa the opening of the O5-M12-O10 and O6-M12-O4 angles (by $1.4^{\circ}$ and $0.25^{\circ}$ respectively), together with the shortening of the bond distances M12-O5 (by $0.033 \AA$ ), the-M12-O10 (by $0.015 \AA$ ), the M12-O4 (by $0.0028 \AA$ ) and the-M12-O6 (by $0.043 \AA$ ) are the main mechanisms of compression and contribute the total of $0.053 \AA$ to the shortening of each octahedron along [010] (see Table 4 and Figure 6). The individual chains of M12 octahedra are linked to one another through parallel chains of alternating M3 octahedra and $T 3$ tetrahedra (Figure 1a), which also have to shorten under compression. With increasing pressure the distance between the O 2 atoms linked to $M 3$ decreases by $0.073 \AA$ through a combination of bond shortening and closing of the O2-M3-O2 angle, while
the O2-O2 distance forming the edge of the $T 3$ tetrahedron shortens by $0.033 \AA$ through bond compression alone.

The octahedral chains in zoisite are softer, along their length, than the $\mathrm{AlO}_{6}$ edge-sharing octahedral chains of the $\mathrm{Al}_{2} \mathrm{SiO}_{5}$ polymorphs, in which the direct compression of $\mathrm{AlO}_{6}$ octahedra results in moduli along the chain directions in excess of 200 GPa in sillimanite and andalusite (Burt et al. 2006), while in kyanite (Yang et al. 1997, Comodi et al. 1997) the linear modulus is still 166 GPa, much greater than in zoisite. In all cases the chains are compressed by shortening Al-O bonds (see Table 8 and Figure 8). The compression of the chains of octahedra must be accommodated by the rest of the structure; therefore the difference in compressibility rise rose from the fact that in andalusite and sillimanite the octahedral chains are better supported by the surrounding structure.

Back to zoisite, the [001] direction is the most compressible of the entire structure with a modulus of 89.1 (9) GPa for the Fe-free sample and 83.2(5) GPa for the Fe-rich one. In Figure 1b the horizontal line joining the M12 sites indicates a pair of sites related by the $2_{1}$ screw axis running along [001]. Therefore their z-coordinates differ by $1 / 2$ and the compression of the framework along $\boldsymbol{c}$ must result in the shortening of this M12-M12 vector, which requires shortening of both the polyhedral links M12-T3-M12 and M12-T1-T2-M12. Within the pressure increase from room $P$ to 7.77GPa the biggest contribution to the shortening of $c / 2(0.115 \AA)$ in the first link is from the shortening by $0.07 \AA$ of the O5-O6 edge of the $T 3$ tetrahedron. Most of the compression of this O5O6 edge (see Figure 9) comes from a decrease in the O5-T3-O6 angle of about $3^{\circ}$, and a small contribution comes from shortening of T3-O5 but not T3-O6 (Figure 7e and Table 4). The rest of the required compression $(0.04 \AA)$ comes from shortening of the M12-O5 and M12-O6 bonds (Figure 7a and Table 4). In the parallel polyhedral link the T1-T2 distance shortens by $0.03 \AA$ (by shortening T2-O9, not by changing the T-O-T angle, see Figure 10 and Table 4). The remaining $0.08 \AA$ compression is supplied by the M12 octahedral rotation together with a small bond distance shortening. The M12 is rotating around the O atoms shared with $T 1$ and $T 2$, which thus compresses
the $T 3$ bridge. The different elastic behaviour with composition can be attributed to the contribution of the M3 polyhedra: compression along [001] can be explained by the softening of the M3-O1 bonds, which can become softer in M3 site populated $\mathrm{Fe}^{3+}$ (Table 4, Figure 7b).
b) Extra framework cations contribution

The distribution of the $\mathrm{Ca}-\mathrm{O}$ bonds in space (especially with respect to the orientation of the shorter vs the longer bonds) does not give any explanation for the anisotropic behavior of the structure. Looking at the Ca 1 site, the shortest of the $\mathrm{Ca}-\mathrm{O}$ bonds, the $\mathrm{Ca} 1-\mathrm{O} 7$ bend-(oxygen shared with $T 2$ tetrahedron, see Figure 1b) is the stiffest one and shortens by only $0.0034 \AA$ (see Table 4). This bond is sub-parallel to [100] and is opposite to the $\mathrm{Ca} 1-\mathrm{O} 6$ and $\mathrm{Ca} 1-\mathrm{O} 5$ (see Figure 1b) that are the longest and softest among the $\mathrm{Ca} 1-\mathrm{O}$ bonds, with a compression of $0.094 \AA$ and $0.061 \AA$ respectively (see Table 4). Therefore the combined effect given by those bonds is that the Ca 1 cavity shrinks isotropically. In the Ca 2 site the stiffest bonds are to the O 3 oxygen which is shared with $T 1$ tetrahedron (Figure 1b) although these are not the shortest bonds. These $\mathrm{Ca}-\mathrm{O} 3$ bonds lie sub-parallel to [100] and would therefore stiffen $\boldsymbol{a}$ were it not for the fact that the two Ca2 -O 2 , which are the longest and softest among the $\mathrm{Ca} 2-\mathrm{O}$ bonds are also sub-parallel to [100]. As a consequence the combined effect is that the Ca 2 cavity, as well as the Ca 1 cavity, shrinks almost isotropically.
c) Hydrogen-bond geometry

As described above, the Fe-free sample shows significant scatter in the $f-F$ plots of the unitcell parameters in the pressure range $4-4.5 \mathrm{GPa}$ (Figure 3), which is not accompanied by a discontinuity in the $P-V$ curve nor is there a change in the slope of the $f-F$ plot of the volume compression (Figure 3d). Therefore, this data scatter is unlikely to be associated with the isostructural phase transition that occurs with increasing iron contents (Liebscher et al. 2002) as that transition leads to an increase in volume compared to that of the end-member composition, and so is not expected to occur upon compression. Nevertheless, it is worth noting that there is a sharp
increase of distortion of the $T 1$ polyhedron (Figure $5 b$ ) and a discontinuity in the 07-09-O8 angle (Figure 11) between 4 and 5 GPa . A close inspection of the $\mathrm{O} 10 \ldots \mathrm{O} 4$ distance (with the O 4 anion being the acceptor of the hydrogen bond in zoisite structure) shows that there is a consistent reduction of this distance in the room- $P$ to $4-5 \mathrm{GPa}$ pressure range (up to $0.2 \AA$, Figure 12 a ). The same is observed for the $\mathrm{O} 10 \ldots \mathrm{O} 2$ distance, which remains unchanged up to 4 GPa , and then startbegin-decreasing. If we combine shortening of these distances with the behaviour of the T1 tetrahedron we can infer that something is happening to the hydrogen bonding topology. Quite likely, the H atom is getting going off of the $\mathrm{O} 4 \ldots \mathrm{O} 10$ junction and start begin to forming a bifurcated bond with O 2 . This change in bonding topology must be instantaneous and would explain the softening in $\mathbf{c}$ lattice parameter and the hardening in $\mathbf{b}$ lattice parameter (Figures $3 \mathbf{b}$ and c).

## CONCLUSIONS

We have shown that the addition of Fe to the structure of pure zoisite results in a slight decrease in the bulk modulus which is the result of a softening of the $\boldsymbol{c}$ axis. The pattern of pressureinduced strains and the structural evolution at high pressure is also similar to the high-temperature behavior measured by Cámara et al. (2011) on an Fe-free crystal from the same sample studied in this work. At high temperatures, in the $T$ range from 303 K to 770 K , the $a$ lattice parameter shows negative thermal expansion (NTE) and shortens by $0.018 \AA$, while $b$ and $c$ lattice parameters expand during the temperature increase by $0.076 \AA$ and $0.150 \AA$ respectively. The anisotropy of the elastic response of zoisite to pressure and temperature is therefore elose-related. In both cases the $c$ lattice parameter shows the biggest variation, being the softest direction at high- $P$ (with a bulk modulus of $K_{\mathrm{T} 0 \mathrm{cFe} \text {-free }}=89.1(9) \mathrm{GPa}$ ) and at high- $T$ it is the direction of greatest expansion, with a thermal expansion coefficient of $\alpha^{\circ}{ }_{33(1 \text { bar,298K })}=18.95(2) \cdot 10^{-6} \mathrm{~K}^{-1}$ (Cámara et al. 2011). The $a$ lattice
parameter is the stiffest direction at high- $P$ (with a bulk modulus of $K_{\text {T0cFe-free }}=244(5) \mathrm{GPa}$ ) and at high- $T$ it shows negative thermal expansion $\left(\alpha^{\circ}{ }_{1(1 \text { larar,298K })}=-1.18(3) \cdot 10^{-6} \mathrm{~K}^{-1}\right.$, Cámara et al. 2011) . The $b$ lattice parameter has a bulk modulus close to that of $c$ at high $-P\left(K_{\text {T0bFe-free }}=105(2) \mathrm{GPa}\right)$ and a thermal expansion coefficient also close to $c\left(\alpha^{\circ}{ }_{22(\operatorname{bar}, 298 \mathrm{~K})}=17.31(2) \cdot 10^{-6} \mathrm{~K}^{-1}\right.$ Cámara et al. 2011). The reasons for this anisotropy can be found in the linkage system of the framework. Along the [001] direction, the compression occurs mainly by shortening of the inter-polyhedral distances (see Figure 6), and is therefore certainly bigger with respect to the compression along [100] and [010] that is achieved by compression of the intra-polyhedral distances (see Figure 7) and rotation of the polyhedra with respect one to another.

The compressibility behavior is also affected by the Fe-contents at the M3 site. As reported by Cámara et al. (2011) the thermo-elastic parameters of the unit-cell volume of zoisite obtained in their study compared with those of a natural epidote previously reported by Gatta et al. (2011) with higher Fe contents [i.e. $\mathrm{Ca}_{1.925} \mathrm{Fe}_{0.745} \mathrm{Al}_{2.265} \mathrm{Ti}_{0.004} \mathrm{Si}_{3.037} \mathrm{O}_{12}(\mathrm{OH})$ ] showed significant differences, being $\alpha_{\mathrm{V}_{\mathrm{o}}}=5.93(5) \cdot 10^{-5} \mathrm{~K}^{-1}$ for zoisite and $5.1(2) \cdot 10^{-5} \mathrm{~K}^{-1}$ for epidote. Therefore, compearing two slightly different structures (epidote and zoisite) it appears-sems that at high- $T$, higher Fe contents reduce the volume thermal expansion coefficient (Cámara et al. 2011); however several high temperature data on zoisite sample with different Fe contents (Camara pers. Comm.), clearly showed that the volume thermal expansion coefficient increases with increasing the Fe contents. This is clearly in agreement with the high- $P$ behavior showed by our data for all the lattice parameters, where the higher the Fe contents the lower the bulk modulus, as showed in Figure 2d and Table 6.

The key to understand this behavior can be found in the evolution of the structure with increasing the-Fe contents. As reported by Liebscher et al. (2002), and confirmed by our data, increasing Fe content causes an increase in the volume of the $M 3$ octahedron while the other polyhedral volumes remain practically constant (Liebscher et al. 2002). The increase in volume of

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the $M 3$ is accommodated by increasing the distance between the cation and the apical oxygens ( O 4 - O8, see Figure 1a) together with the M3-O1 bond lengths (see Figure 1a and b). The increase of the $\mathrm{O} 4-\mathrm{O} 8$ distance causes both a regularization and a rotation of the $M 3$ octahedra together with a rotation (by $0.5^{\circ}$ ) and regularization of the M12 octahedra (Liebscher et al. 2002). The increase of the $\mathrm{O} 4-\mathrm{O} 8$ distance along $a$ together with the rotation of $M 3$ is compensated by rotation of the $T 2$ tetrahedra around the relatively fixed apical O3 atom of the M12 octahedron. The rotation of $T 2$ leads to rotation of the corner-linked $T 1$ tetrahedron in the opposite direction, which is reflected by the increasing 08-09-O7 angle of the $\mathrm{Si}_{2} \mathrm{O}_{7}$ group. Therefore, an increase in the Fe content has essentially the same effect on the tilt of $T 1$ and $T 2$ tetrahedra as increasing pressure that opens the O8-09-O7 (see Figure 11). Therefore, this mechanism of rotation of $T 1$ and $T 2$ tetrahedra that takes place with increasing Fe content softens the [001] direction by making the relative rotation of $T 1$ and $T 2$ tetrahedra even easier, while the increase in the distance $\mathrm{O} 4-\mathrm{O} 8$ together with the rotation of M12 stiffens the [100] direction. In addition, the increase of the M3-O1 bond lengths with increasing Fe content is a further mechanism that softens the [001] direction.

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Figures

Figure 1



Figure 1. Schematic view of the structure along different crystallographic directions: (a) along $b$ lattice tilted to show the direction of the octahedra chain going along $b$ direction.; (b) along $b$
lattice, with over imposed the $2_{1}$ screw axis running along [100] and the black line joining a pair of M12 sites related by a $2_{1}$ screw axis running alon [ $\left.\begin{array}{lll}0 & 1 & 0\end{array}\right]$; (c) inclined along $c$ lattice direction showing the mirror plane (black slid line) relating the M12 sites.

Figure 2


Figure 2. Evolution of the unit cell parameters as a function of pressure: (a) $a / a_{0}$; (b) $b / b_{0}$; (c) $c / c_{0}$ and (d) $V / V_{0}$ unit-cell parameters as a function of pressure for the samples Fe-free (filled squares), Fe-rich (filled circles), data by Comodi and Zanazzi, (1997) (open inverted triangle) Pawley et al., (1998) (open square) and Grevel et al., (2000) (open triangle).

Figure 3


a

c
b

d

Figure 3. $\mathrm{F}_{\mathrm{E}}-f_{\mathrm{E}}$ plot for the (a) $a$ lattice parameter, (b) $b$ lattice parameter, (c) $c$ lattice parameter and (d) unit cell volume.

Figure 4
a

b

c


Figure 4. Evolution of the volumes of the polyhedra with increasing pressure: (a) Cal and Ca 2 sites (open and filled triangles respectively); (b) M12 and M3 sites (opene and filled triangles respectively); (c) $T 1, T 2$ and $T 3$ (squares, circles and triangles respectively).

Figure 5


Figure 5. Variation of the polyhedra distortions with pressure. (a) $C a 1$ and $C a 2$ sites (open triangle and filled triangle respectively); (b) $T 1, T 2, T 3, M 12, M 3$ (open triangle, filled triangle, open inverse triangle, filled inverse triangle and open square respectively).

Figure 6


Figure 6. Inter-octahedra distances O4-O10 (open triangle) and O5-O6 (filled triangle).

Figure7
a b


c
d



e
Figure 7. Evolution of the $M 12-\mathrm{O}$ (a), $M 3-\mathrm{O}$ (b), $T 1$ (c), $T 2$ (d) and $T 3$ (e) bond distances with increasing pressure.

Figure 8


Figure 8. Comparison of the structure of andalusite (a), sillimanite (b) and kyanite (c) showing the difference in the linkage between the octahedra and the different orientation of the chain of octahedra.

Figure 9


Figure 9. Variation of O5-O6 distance with $P$.

Figure 10


Figure 10. Variation of $T 1-T 2$ distance with $P$.

## Figure 11



Figure 11. Evolution of the $\mathrm{O} 7-\mathrm{O} 9-\mathrm{O} 8$ and $\mathrm{O} 1-\mathrm{O} 9-\mathrm{O} 3$ angles with increasing pressure.

Figure 12

b


Figure 12. Evolution of the $\mathrm{O} 4-\mathrm{O} 10$ and $\mathrm{O} 2-\mathrm{O} 10$ bond distances increasing pressure (a) and with the variations of the $T 1$ tetrahedron distortion parameter (b) as defined by Makovicky and
Balić Žunić, (1998)

## Tables

| $\mathrm{Na}_{2} \mathrm{O}$ | 0.01(1) | Na | 0.001 | $\mathrm{Na}_{2} \mathrm{O}$ | 0.06(12) | Na | 0.009 |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| MgO | 0.01(1) | Mg | 0.001 | MgO | 0.04(3) | Mg | 0.005 |
| $\mathrm{Al}_{2} \mathrm{O}_{3}$ | 33.56(15) | AI ${ }^{(\mathrm{V})}$ | 2.986 | $\mathrm{Al}_{2} \mathrm{O}_{3}$ | 31.83(21) | $\mathrm{Al}^{(\mathrm{VI})}$ | 2.876 |
| $\mathrm{SiO}_{2}$ | 39.91(18) | Si | 3.014 | $\mathrm{SiO}_{2}$ | 38.96(15) | Si | 2.987 |
| $\mathrm{K}_{2} \mathrm{O}$ | 0.00(1) |  | 0.000 | $\mathrm{K}_{2} \mathrm{O}$ | 0.01(1) | K | 0.001 |
| CaO | 24.25(16) | Ca | 1.962 | CaO | 24.23(15) | Ca | 1.990 |
| $\mathrm{TiO}_{2}$ | 0.05(1) | Ti | 0.003 | $\mathrm{TiO}_{2}$ | 0.08(1) | Ti | 0.005 |
| $\mathrm{V}_{2} \mathrm{O}_{3}$ | 0.15(1) | V | 0.009 | $\mathrm{V}_{2} \mathrm{O}_{3}$ | 0.01(2) | v | 0.001 |
| $\mathrm{Cr}_{2} \mathrm{O}_{3}$ | 0.00(0) |  | 0.000 | $\mathrm{Cr}_{2} \mathrm{O}_{3}$ | 0.00(0) | Cr | 0.000 |
| MnO | 0.01(1) | Mn | 0.000 | MnO | 0.02(1) | Mn | 0.005 |
| $\mathrm{Fe}_{2} \mathrm{O}_{3}$ | 0.02(2) | $\mathrm{Fe}^{3+}$ | 0.001 | $\mathrm{Fe}_{2} \mathrm{O}_{3}$ | 2.06(22) | $\mathrm{Fe}^{3+}$ | 0.119 |
| SrO | 0.17(5) | Sr | 0.008 | SrO | 0.45(4) | Sr | 0.020 |
| $\mathrm{La}_{2} \mathrm{O} 3$ | 0.02(2) |  | 0.000 | $\mathrm{La}_{2} \mathrm{O} 3$ | 0.02(2) | La | 0.000 |
| $\mathrm{Ce}_{2} \mathrm{O} 3$ | 0.02(3) | Ce | 0.001 | $\mathrm{Ce}_{2} \mathrm{O} 3$ | 0.03(2) | Ce | 0.001 |
| $\mathrm{H}_{2} \mathrm{O}$ | 1.99(1) | H | 1.000 | $\mathrm{H}_{2} \mathrm{O}$ | 1.96(1) | H | 1.000 |
| Total | 100.16 | Tota | 8.985 | Total | 99.75 | Total | 9.015 |

Notes: ${ }^{*}$ Calculated assuming 1 a.p.f.u. of H . F was below the detection limit.

Table 2a. Unit-cell parameters at the different pressure values, measured on the Fe-free crystal.

| $P(\mathrm{GPa})$ | $a(A)$ | $b(A ̊)$ | $c(A)$ | $V\left(\hat{A}^{3}\right)$ |
| :---: | :---: | :---: | :---: | :---: |
| 0.0001 | 16.2004 (5) | 5.5529 (4) | 10.0423 (5) | 903.39 (6) |
| 0.697 (5) | 16.1869 (5) | 5.5411 (4) | 10.0169 (6) | 898.45 (7) |
| 1.144 (7) | 16.1760 (4) | 5.5342 (4) | 10.0007 (6) | 895.27 (6) |
| 1.926 (8) | 16.1587 (4) | 5.5215 (3) | 9.9746 (5) | 889.93 (6) |
| 2.945 (7) | 16.1353 (4) | 5.5063 (3) | 9.9432 (5) | 883.41 (6) |
| 3.426 (6) | 16.1239 (4) | 5.5001 (3) | 9.9288 (5) | 880.51 (6) |
| 4.025 (11) | 16.1124 (5) | 5.4917 (4) | 9.9110 (6) | 876.96 (6) |
| 4.685 (12) | 16.0988 (5) | 5.4827 (3) | 9.8921 (6) | 873.12 (6) |
| 5.533 (11) | 16.0796 (5) | 5.4721 (4) | 9.8690 (6) | 868.36 (6) |
| 6.515 (12) | 16.0583 (4) | 5.4602 (3) | 9.8439 (5) | 863.12 (5) |
| 6.511 (8) | 16.0540 (4) | 5.4609 (4) | 9.8437 (6) | 863.00 (6) |
| 7.766 (9) | 16.0271 (4) | 5.4473 (3) | 9.8119 (5) | 856.63 (6) |
| 5.145 (6)* | 16.0845 (4) | 5.4777 (3) | 9.8785 (6) | 870.36 (6) |
| 4.325 (9)* | 16.1038 (5) | 5.4886 (3) | 9.9011 (6) | 875.13 (6) |
| 4.243 (8)* | 16.1063 (5) | 5.4896 (4) | 9.9026 (6) | 875.56 (7) |
| 4.243 (8)* | 16.1054 (5) | 5.4897 (3) | 9.9020 (5) | 875.47 (6) |
| 4.239 (6)* | 16.1067 (6) | 5.4893 (5) | 9.9036 (8) | 875.61 (9) |
| 2.496 (5)* | 16.1468 (4) | 5.5130 (3) | 9.9560 (5) | 886.26 (5) |
| 1.680 (5)* | 16.1623 (5) | 5.5253 (4) | 9.9833 (7) | 891.52 (7) |
| 0.722 (5)* | 16.1831 (5) | 5.5409 (4) | 10.0150 (6) | 898.03 (7) |
| 0.0001 | 16.1992 (4) | 5.5529 (4) | 10.0420 (6) | 903.29 (7) |

Note: *data measured during decompression. Standard deviation are given in parenthesis

Table 2b. Unit-cell parameters at the different pressure values, measured on the Fe-rich crystal.

| $\boldsymbol{P}(\mathbf{G P a})$ | $\boldsymbol{a}(\mathbf{A})$ | $\boldsymbol{b}(\mathbf{A})$ | $\boldsymbol{c}(\mathbf{A})$ | $\boldsymbol{V}\left(\AA^{3}\right)$ |
| :---: | :---: | :---: | :---: | :---: |
| $0.0001(0)$ | $16.2095(6)$ | $5.5654(2)$ | $10.0536(2)$ | $906.95(5)$ |
| $3.377(7)$ | $16.1363(7)$ | $5.5133(2)$ | $9.9353(4)$ | $883.88(6)$ |
| $4.398(9)$ | $16.1132(8)$ | $5.4994(3)$ | $9.9044(3)$ | $877.66(7)$ |
| $5.375(7)$ | $16.0916(5)$ | $5.4868(2)$ | $9.8776(3)$ | $872.10(5)$ |
| $6.709(9)$ | $16.0623(7)$ | $5.4712(2)$ | $9.8428(4)$ | $864.98(6)$ |
| $7.139(9)$ | $16.0535(5)$ | $5.4659(3)$ | $9.8321(5)$ | $862.72(7)$ |
| $7.626(10)$ | $16.0430(7)$ | $5.4603(2)$ | $9.8196(4)$ | $860.19(6)$ |
| $5.875(9)_{-}^{\star}$ | $16.0801(6)$ | $5.4809(2)$ | $9.8644(4)$ | $869.38(5)$ |
| $2.785(5)_{-}^{\star}$ | $16.1503(5)$ | $5.5211(2)$ | $9.9538(3)$ | $887.55(5)$ |
| $1.542(7)_{-}^{\star}$ | $16.1753(8)$ | $5.5402(4)$ | $9.9958(2)$ | $895.77(8)$ |
| $1.124(6)_{-}^{\star}$ | $16.1850(7)$ | $5.5469(2)$ | $10.0103(3)$ | $898.70(6)$ |
| $0.0001(0)_{-}^{\star}$ | $16.2084(11)$ | $5.5658(3)$ | $10.0539(4)$ | $906.99(9)$ |

Note: *data measured during decompression. Standard deviations are given in parenthesisparentheses

| $P$ (GPa) | 0.0001 | 1.144(7) | 2.945(7) | 4.025(11) | 5.533(11) | 6.511(12) | 7.766(9) | 5.145(6)* |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Label | P0 | P2 | P4 | P6 | P8 | P9b | P10b | P11d |
| $a(A)$ | 16.2004(5) | 16.1760(4) | 16.1353(4) | 16.1124(5) | $16.0796(5)$ | 16.0539(4) | $16.0271(4)$ | 16.0845(4) |
| $b(A)$ | $5.5529(4)$ | $5.5342(4)$ | 5.5062(3) | $5.4917(4)$ | $5.4721(4)$ | 5.4609(4) | 5.4473(3) | $5.4777(3)$ |
| $\theta(\hat{A})$ | 10.0423(5) | $10.0007(6)$ | 9.9432(5) | $9.9110(6)$ | 9.8690(6) | 9.8437(6) | 9.8119(5) | 9.8785(6) |
| $V\left(\hat{A}^{3}\right)$ | 903.40 (8) | 895.27 (8) | 883.41(7) | 876.96(8) | 868.36(8) | 863.00(8) | 856.63(7) | 870.36(8) |
| space group | Pnma | Pnma | Pnma | Pnma | Pnma | Pnma | Pnma | Pnma |
| $z$ | 4 | 4 | 4 | 4 | 4 | 4 | 4 | 4 |
| $\begin{gathered} \text { Density (calc.) } \\ M g / m^{3} \end{gathered}$ | 3.223 | 3.252 | 3.296 | 3.32 | 3.353 | 3.374 | 3.399 | 3.345 |
| Absorption coeff. $\left(\mathrm{mm}^{-1}\right)$ | 2.03 | 2.05 | 2.08 | 2.11 | 2.11 | 2.13 | 2.14 | 2.12 |
| Resolution ( $\theta$ ) | $\begin{aligned} & 2.51 \text { to } \\ & 29.99^{\circ} \end{aligned}$ | $\begin{aligned} & 2.39 \text { to } \\ & 29.99^{\circ} \end{aligned}$ | $\begin{aligned} & 2.41 \text { to } \\ & 29.96^{\circ} \end{aligned}$ | $\begin{aligned} & 2.42 \text { to } \\ & 29.99^{\circ} \end{aligned}$ | $\begin{aligned} & 2.42 \text { to } \\ & 29.93^{\circ} \end{aligned}$ | $\begin{aligned} & 2.43 \text { to } \\ & 29.99^{\circ} \end{aligned}$ | $\begin{aligned} & 2.43 \text { to } \\ & 29.99^{\circ} \end{aligned}$ | $\begin{aligned} & 2.42 \text { to } \\ & 29.99^{\circ} \end{aligned}$ |
| Completeness (\%) | 66.2 | 72.3 | 73.1 | 72.7 | 72.4 | 71.0 | 71.9 | 70.8 |
| Extinction coeff | 0.015(2) | 0.015(2) | 0.029(2) | 0.035(2) | $0.034(2)$ | 0.038(5) | 0.038(3) | $0.021(2)$ |
| n. of $/ \mathrm{l}$ s $>4$ | 591 | 493 | 539 | 459 | 459 | 552 | 421 | 439 |
| $n$ all | 952 | 1028 | 1024 | 1014 | 1003 | 1000 | 990 | 998 |
| GooF | 1.06 | 0.937 | 0.923 | 0.903 | 0.905 | 1.023 | 0.929 | 0.909 |
| $R_{\text {int }}$ (\%) | 0.059 | 0.069 | 0.064 | 0.086 | 0.099 | 0.012 | 0.101 | 0.086 |
| $R_{4 s}$ (\%) | 6.20 | 6.21 | 5.22 | 5.40 | 5.65 | 7.84 | 6.58 | 5.92 |
| $R_{w}$ (\%) | 9.25 | 9.49 | 10.62 | 10.51 | 11.07 | 14.56 | 11.14 | 10.11 |
| $R_{\text {wall }}$ (\%) | 13.7 | 18.2 | 14.36 | 18.79 | 18.35 | 16.80 | 13.61 | 12.52 |
| Rall (\%) | 10.9 | 11.74 | 12.25 | 12.63 | 13.20 | 17.04 | 21.06 | 20.10 |
| $N \mathrm{P}^{\dagger}$ | 57 | 57 | 57 | 57 | 57 | 57 | 57 | 57 |
| Note: * data measured during decompression; $\dagger$ NP = number of refine parameters. |  |  |  |  |  |  |  |  |



| <M3-O> | 1.96(12) | 1.95(12) | 1.94(12) | 1.94(12) | 1.93(11) | 1.92(11) | 1.93(11) | 1.92(11) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $V_{\text {M } 3(~}\left({ }^{3}\right)$ | 9.77(4) | 9.59(5) | 9.51(4) | 9.44(5) | 9.39(5) | 9.27(5) | 9.30(5) | 9.25(5) |
| $\mathrm{OQE}_{\text {м }}$ | $\begin{array}{r} 1.021 \\ 2 \end{array}$ | 1.0214 | 1.0191 | 1.0193 | 1.0182 | 1.0192 | 1.0183 | 1.0187 |
| $\mathrm{OAV}_{\text {M }}$ | $\begin{array}{r} 48.20 \\ 7 \end{array}$ | 50.718 | 47.591 | 47.527 | 48.284 | 50.723 | 48.94 | 50.78 |
| $\mathrm{M} 3 \mathrm{~V}_{\text {distortion }}$ | $\begin{array}{r} 0.019 \\ 6 \end{array}$ | 0.0213 | 0.0197 | 0.0203 | 0.019 | 0.0209 | 0.0201 | 0.0214 |
| Ca1-O7 | 2.261(6) | 2.261 (7) | 2.251 (6) | 2.255 (7) | 2.262 (8) | 2.236 (8) | 2.230 (8) | 2.257 (8) |
| Ca1-O3 $\times 2$ | 2.408(5) | 2.405(5) | 2.399 (4) | 2.381 (5) | 2.386 (6) | 2.385 (5) | 2.376 (6) | 2.374 (6) |
| Ca1-O1 x2 | 2.516(5) | $2.502(5)$ | 2.495 (5) | 2.489 (5) | 2.479 (6) | 2.472 (5) | 2.471 (6) | 2.461 (8) |
| Ca1-O6 | 2.558(6) | 2.541 (7) | 2.507 (6) | 2.509 (7) | 2.499 (7) | 2.483 (7) | 2.484 (8) | 2.464 (6) |
| Ca1-O5 | 2.579(6) | $2.573(7)$ | 2.557 (6) | 2.551 (7) | 2.534 (7) | 2.542 (7) | 2.545 (7) | 2.518 (8) |
| <Ca1-O> | 2.46(17) | 2.46 (17) | 2.44 (17) | 2.44 (17) | 2.43 (18) | 2.43 (17) | 2.42 (17) | 2.42 (18) |
| $V_{\text {Ca1 }}\left(\AA^{3}\right)$ | 19.03(7) | 18.81(8) | 18.52 (7) | 18.36 (8) | 18.20 (9) | 18.01 (9) | 17.99 (9) | 17.80 (9) |
| $\mathrm{Ca2}-\mathrm{O} 7$ | 2.301(7) | 2.293(8) | 2.278 (7) | 2.275 (8) | 2.254 (8) | 2.256 (8) | 2.265 (9) | 2.222 (9) |
| $\mathrm{Ca} 2-\mathrm{O} 3 \times 2$ | 2.454(4) | 2.461 (5) | 2.453 (4) | 2.454 (5) | 2.440 (5) | 2.452 (6) | 2.436 (6) | 2.421 (6) |
| $\mathrm{Ca} 2-\mathrm{O} 2 \times 2$ | 2.515(4) | $2.507(5)$ | 2.483 (5) | 2.465 (5) | 2.454 (6) | 2.452 (5) | 2.451 (6) | 2.438 (6) |
| $\mathrm{Ca2}-\mathrm{O} 2 \times 2$ | 2.794(4) | 2.770 (5) | 2.753 (4) | 2.726 (5) | $2.727 \quad$ (5) | 2.724 (5) | 2.711 (5) | 2.709 (5) |
| <Ca2-O> | 2.55(10) | 2.54(16) | 2.52 (16) | 2.51 (16) | 2.50 (17) | 2.50 (17) | 2.49 (16) | 2.48 (18) |
| $V_{\mathrm{Ca} 2}\left(\mathrm{~A}^{3}\right)$ | 22.55(8) | 22.33(9) | 21.88 (7) | 21.54 (9) | 21.27 (9) | 21.34 (9) | 21.16 (10) | 20.78 (10) |
| O4..010 | 2.749(10) | 2.725(10) | 2.679(9) | 2.632(11) | 2.616(11) | 2.630(11) | 2.603(13) | 2.580(12) |
| O2..010 | 2.954(7) | 2.954(8) | $2.957(7)$ | 2.959(8) | 2.950 (8) | 2.946(8) | 2.938 (8) | 2.932 (8) |
| O5-M12-O10 | 82.7(2) | 83.1(2) | 83.1(2) | 83.8(2) | 83.8(2) | 83.9(2) | 84.1(2) | 84.2(2) |
| O6-M12-O4 | 86.1(2) | 85.9(2) | 86.3(2) | 86.4(2) | 85.7(2) | 85.7(2) | 86.4(2) | 86.3(2) |

Note: TQE, TAV, OQE, and OAV are quadratic elongation and angle variance for tetrahedra and ecthaedraoctahedran (Robinson en
al. 1971); polyhedral volumes calculated with IVTON (Balić-Z̃unić and Vicković, 1996); *Data measured during decompression. Volume distortions, ( $\mathrm{V}_{\text {distortion) }}$ ) are calculated as defined by Makovicky and Balić Žunić; ( 1998 ).

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| $\boldsymbol{P}(\mathrm{GPa})$ | 0.0001 | 1.144(7) | 2.945(7) | 4.025(11) | 5.533(11) | 6.511(12) | 7.766(9) | 5.145(6)* |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| T1x | 0.41075(13) | 0.41099(15) | 0.41048(13) | 0.41046(18) | 0.41037(18) | 0.41005(18) | 0.41022(18) | 0.41022(16) |
| y | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 |
| z | 0.2828(3) | 0.2821 (3) | 0.2808(3) | 0.2798(3) | 0.2809(3) | 0.2794(4) | 0.2793(3) | 0.2796(4) |
| $\mathrm{U}_{\text {iso }}$ | 0.0067(5) | 0.0060(6) | 0.0103(5) | 0.0108(6) | 0.0123(7) | 0.0113 (7) | 0.0115(6) | 0.0108(7) |
| T2x | 0.08147 (13) | 0.08098(15) | $0.08073(14)$ | 0.08033(18) | 0.08034(18) | 0.08054(18) | 0.08008(18) | 0.08005(18) |
| y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| z | 0.1061(3) | 0.1059(3) | 0.1057(3) | 0.1059(3) | 0.1051(3) | 0.1054(4) | 0.1059(3) | 0.1058(4) |
| $\mathrm{U}_{\text {iso }}$ | 0.0069(5) | 0.0065(6) | 0.0101(5) | 0.0117(7) | 0.0110(7) | 0.0097(7) | 0.0101(6) | 0.0119(7) |
| T3x | 0.16030(12) | 0.16030(14) | 0.16006(13) | 0.15964(16) | 0.15965(18) | 0.16001(16) | 0.16003(18) | 0.15990(18) |
| y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| z | 0.4355(3) | 0.4349(3) | 0.4354(3) | 0.4349(3) | 0.4353(3) | 0.4350(3) | 0.4350(3) | 0.4356(3) |
| $\mathrm{U}_{\text {iso }}$ | 0.0056(5) | 0.0061(6) | 0.0096(5) | 0.0111(6) | 0.0117(6) | 0.0095(6) | 0.0105(6) | 0.0096(6) |
| M12X | 0.24952(10) | 0.24938(11) | 0.24930(10) | 0.24907(12) | 0.24908(12) | 0.24907(13) | 0.24889(13) | 0.24902(13) |
| Y | 0.9971(3) | 0.9978(4) | 0.9976(3) | 0.9975(4) | 0.9979(4) | 0.9983(4) | 0.9986(4) | 0.9989(6) |
| Z | 0.1895(2) | 0.1895(2) | 0.1888(2) | 0.1882(2) | 0.1885(3) | 0.1875(3) | 0.1878(2) | 0.1873(3) |
| $\mathrm{U}_{\text {iso }}$ | 0.0066(4) | 0.0059(4) | 0.0098(4) | 0.0104(4) | 0.0106(5) | 0.0100(5) | 0.0106(5) | 0.0109(5) |
| M3X | 0.10596(14) | 0.10568(18) | 0.10561(15) | 0.10572(18) | 0.1059(2) | 0.10516(18) | 0.1054(2) | 0.1053(2) |
| Y | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 |
| Z | 0.3005(3) | 0.3000(3) | 0.2994(3) | 0.3002(4) | 0.2996(4) | 0.3000(4) | 0.2989(4) | 0.2993(4) |
| $\mathbf{U}_{\text {iso }}$ | 0.0072(5) | 0.0068(6) | 0.0108(6) | 0.0113(7) | 0.0127(7) | 0.0091(7) | 0.0109(7) | 0.0113(7) |
| CA1X | 0.36666(11) | 0.36624(12) | 0.36599(10) | 0.36572(13) | 0.36573(13) | 0.36545(13) | 0.36554(13) | 0.36544(14) |
| Y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| Z | 0.4375(2) | 0.4370(2) | 0.4365(2) | 0.4360(2) | 0.4363(3) | 0.4363(3) | 0.4359(2) | 0.4359(3) |
| $\mathrm{U}_{\text {iso }}$ | 0.0109(4) | 0.0099(4) | 0.0140(4) | 0.0145(5) | 0.0141(5) | 0.0137(5) | 0.0137(5) | 0.0141(5) |
| CA2X | 0.45185(11) | 0.45266(12) | 0.45335(11) | 0.45397(13) | 0.45376(13) | 0.45426(14) | 0.45466(13) | 0.45481(14) |
| Y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| Z | 0.1150(2) | 0.1139(2) | 0.1127(2) | 0.1118(2) | 0.1117(3) | 0.1113(3) | 0.1109(2) | 0.1103(3) |
| $\mathrm{U}_{\text {iso }}$ | 0.0104(4) | 0.0084(5) | 0.0122(4) | 0.0125(5) | 0.0119(5) | 0.0112(5) | 0.0117(5) | 0.0128(6) |
| 01X | 0.1305(2) | 0.1302(2) | 0.1291(2) | 0.1288(2) | 0.1281(2) | 0.1280(3) | 0.1284(3) | 0.1280(3) |
| Y | 0.9998(7) | 0.9969(8) | 0.9951(7) | 0.9951(9) | 0.9945(8) | 0.9931(9) | 0.9929(8) | 0.9936(9) |
| Z | 0.1465(4) | 0.1465(6) | 0.1475(4) | 0.1471(6) | 0.1486(6) | 0.1478(6) | 0.1469(6) | 0.1472(6) |
| $\mathrm{U}_{\text {iso }}$ | 0.0094(9) | 0.0089(10) | 0.0116(9) | 0.0111(11) | 0.0116(11) | 0.0131(12) | 0.0111(10) | 0.0122(12) |
| 02X | 0.1014(2) | 0.1013(3) | 0.1015(2) | 0.1005(3) | 0.1015(3) | 0.1012(3) | 0.1013(3) | 0.1009(3) |
| Y | 0.0132(7) | 0.0139(9) | 0.0147(8) | 0.0146(9) | 0.0133(9) | 0.0140(9) | 0.0123(9) | 0.0128(11) |


| z | 0.4302(6) | 0.4297(6) | $0.4311(4)$ | 0.4314(6) | 0.4329(6) | 0.4320(6) | 0.4313(6) | 0.4321(6) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{U}_{\text {iso }}$ | 0.0087(8) | 0.0097(9) | 0.0131(9) | 0.0125(10) | 0.0173(11) | 0.0148(11) | 0.0132(10) | $0.0128(11)$ |
| 03X | $0.3591(2)$ | 0.3588(3) | 0.3591 (2) | 0.3591 (3) | 0.3580(3) | 0.3588(3) | 0.3579(3) | $0.3584(3)$ |
| Y | $0.9917(7)$ | 0.9913(8) | 0.9912(7) | 0.9927(9) | 0.9958(8) | 0.9948(9) | 0.9945(9) | 0.9940(9) |
| z | 0.2452(4) | 0.2443(6) | $0.2428(4)$ | 0.2430(6) | $0.2421(6)$ | 0.2409(6) | 0.2406(6) | $0.2411(6)$ |
| $\mathrm{U}_{\text {iso }}$ | 0.0079(8) | 0.0068(9) | 0.0115(8) | 0.0110(9) | 0.0108(10) | $0.0137(11)$ | 0.0123(9) | $0.0113(10)$ |
| 04x | 0.2190(3) | 0.2183(4) | 0.2191 (3) | $0.2187(4)$ | 0.2195(4) | 0.2198(4) | $0.2189(4)$ | 0.2194(6) |
| Y | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 |
| z | $0.3014(7)$ | $0.3011(7)$ | 0.3029(6) | 0.3040(8) | $0.3032(8)$ | 0.3036(8) | 0.3020(8) | 0.3033(9) |
| $\mathrm{U}_{\text {iso }}$ | 0.0077(13) | 0.0083(15) | 0.0102(13) | 0.0131(16) | 0.0098(16) | 0.0093(16) | 0.0084(15) | 0.0155(18) |
| 05x | 0.2277(3) | $0.2274(4)$ | 0.2275 (3) | 0.2269(4) | $0.2273(4)$ | $0.2277(4)$ | $0.2273(4)$ | $0.2267(4)$ |
| Y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| z | $0.3122(7)$ | $0.3115(7)$ | 0.3115(6) | 0.3123(7) | $0.3121(8)$ | $0.3119(8)$ | 0.3109(7) | 0.3110(8) |
| $\mathrm{U}_{\text {iso }}$ | 0.0086(13) | 0.0057(13) | 0.0097(12) | 0.0110(15) | 0.0123(16) | 0.0100(15) | 0.0084(15) | 0.0072(15) |
| 06X | 0.2718(3) | $0.2714(4)$ | 0.2701 (3) | 0.2702(4) | 0.2704(4) | $0.2708(4)$ | $0.2693(4)$ | $0.2702(4)$ |
| Y | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 |
| z | 0.0600(7) | 0.0596(7) | $0.0584(7)$ | 0.0596(7) | 0.0602(8) | 0.0579(8) | 0.0588(8) | 0.0575(9) |
| $\mathrm{U}_{\text {iso }}$ | 0.0083(12) | 0.0064(12) | 0.0103(11) | 0.0098(14) | 0.0144(16) | $0.0077(14)$ | 0.0105(14) | 0.0121(15) |
| 07x | 0.9910(3) | 0.9909(4) | 0.9903(4) | $0.9907(4)$ | $0.9903(4)$ | 0.9905(6) | $0.9892(6)$ | 0.9893(6) |
| Y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| z | $0.1647(7)$ | 0.1653(7) | $0.1662(6)$ | $0.1665(8)$ | $0.1682(9)$ | 0.1683(8) | $0.1676(8)$ | 0.1666(9) |
| $\mathrm{U}_{\text {iso }}$ | 0.0103(13) | 0.0091(15) | 0.0141(13) | 0.0144(16) | 0.0153(17) | 0.0178(18) | 0.0171(18) | 0.0185(19) |
| 08x | 0.9969(3) | $0.9967(4)$ | $0.9962(4)$ | $0.9962(4)$ | 0.9953(4) | 0.9955(4) | 0.9961(6) | 0.9953(4) |
| Y | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 |
| z | 0.2938(6) | 0.2965(7) | 0.2988 (6) | 0.2999(8) | 0.3005(8) | 0.3013(8) | 0.3018(8) | 0.3024(9) |
| $\mathrm{U}_{\text {iso }}$ | 0.0095(13) | 0.0094(14) | 0.0135(12) | 0.0130(14) | 0.0144(17) | $0.0117(16)$ | 0.0126(15) | $0.0142(17)$ |
| 09x | 0.4202(4) | 0.4200(6) | 0.4215(4) | 0.4217(4) | 0.4204(6) | 0.4208(6) | 0.4201(6) | 0.4207(6) |
| Y | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 | 0.75 |
| z | 0.4429(8) | 0.4446(9) | 0.4429(7) | 0.4422(9) | 0.4463(9) | 0.4424(11) | 0.4418(11) | 0.4431(11) |
| $\mathrm{U}_{\text {iso }}$ | 0.0189(15) | 0.0205(18) | 0.0237(16) | 0.025(2) | 0.025(2) | 0.028(2) | 0.027(2) | 0.028(2) |
| 010x | $0.2677(4)$ | $0.2668(4)$ | $0.2644(4)$ | 0.2633(4) | $0.2621(4)$ | $0.2613(4)$ | 0.2620(4) | 0.2621(4) |
| Y | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 | 0.25 |
| z | 0.0743(7) | 0.0725(7) | $0.0710(7)$ | 0.0679(8) | 0.0679(8) | 0.0666(8) | $0.0667(8)$ | 0.0660(9) |
| $\mathrm{U}_{\text {iso }}$ | 0.0098(13) | 0.0101(14) | 0.0142(13) | 0.0141(15) | 0.0116(15) | $0.0101(15)$ | $0.0141(15)$ | 0.0125(16) |

Note: *Data measured during decompression.

Table 6. Fitted unit cell parameters, bulk moduli and $1^{\text {st }}$ derivative of the bulk modulus at room-pressure obtained using a BM3 EoS.

|  | Fe-free |  |  |  | Fe-rich |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | $a(A ̊)$ | $b$ ( ${ }_{\text {A }}$ ) | $c(A ̊)$ | $V\left(\AA^{3}\right)$ | $a($ Å) | $b$ ( ${ }_{\text {A }}$ ) | $c(A)$ | $V\left(\AA^{3}\right)$ |
| Valo | 16.2008(8) | 5.5530(3) | 10.0421(5) | 903.39(5) | 16.2098(6) | 5.5655(2) | 10.0536(2) | 906.95(5) |
| $K_{\text {Lo }}$ (GPa) | 244(5) | 105.6(1.6) | 89.1(9) | 122.1 (7) | 248(4) | 106(1) | 83.2(5) | 119.1(7) |
| $K$ | -0.7(1.2) | 8.4(5) | 6.6(3) | 6.8(2) | -0.7(0.9) | 8.2 (3) | 7.5(2) | 7.3(2) |


| $K_{\text {To }}(\mathrm{GPa})$ <br> Linear modulus (GPa) | Fe-free |  |  |  | Fe-rich |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | $\begin{gathered} a \\ 244(5) \end{gathered}$ | $\begin{gathered} b \\ 105.6(1.6) \end{gathered}$ | $\begin{gathered} c \\ 89.1(9) \end{gathered}$ | $\begin{gathered} a \\ 248(4) \end{gathered}$ | $\begin{gathered} b \\ 106(1) \end{gathered}$ | $\begin{gathered} c \\ 83.2(5) \end{gathered}$ | (a) |
|  | 732(15) | 317(5) | 267(3) | 744(12) | 318(3) | 249.60(15) | (b) |
| Isothermal linear compressibilities $\left(\mathrm{GPa}^{-1}\right)$ | 0.00137(3) | 0.00314(6) | 0.00375(4) | 0.00134(2) | 0.00314(3) | 0.004006(2) | (c) |
| adiabatic linear compressibilities $\left(\mathrm{GPa}^{-1}\right)$ | 0.00132 | 0.00309 | 0.00381 |  |  |  | (d) |

Note: a) This work, K value obtained from EosFit 5.2 using the method described in the text; b) This work, linear modulus, calculated as described in the
text; c) This work, Isothermal linear compressibilities, calculated as described in the text; d) Mao et al., (2007), adiabatic linear compressibilities calculated
from the sij as described in the text.

Table 8. Average rate of change of the distances Al-O (to the shared octahedral edge), Al-Al and O-O (length of the shared edge)
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|  |  | Andalusite |  |  | Sillimanite |  |  | Zoisite |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | Label | Bond distances (room P) | $\begin{gathered} \text { rate } \\ (\AA \mathrm{A} / \mathrm{GPa}) \end{gathered}$ | Label | Bond distances (room P) | $\begin{gathered} \text { rate } \\ (\mathrm{A} / \mathrm{GPa}) \end{gathered}$ | Label | Bond distances (room P) | $\begin{gathered} \text { rate } \\ (\mathrm{A} / \mathrm{GPa}) \end{gathered}$ |
| Shared edge | Al1-OA | 1.824(3) | -0.0016 | Al1-OA | 1.916(3) | -0.0074 | M12-04 | 1.841(5) | -0.0005 |
|  | Al1-OB | 1.893(3) | -0.0013 | Al1-OB | 1.863(3) | -0.0039 | M12-010 | 1.843(5) | -0.0019 |
|  |  |  |  |  |  |  | M12-05 | 1.901(5) | -0.0041 |
|  |  |  |  |  |  |  | M12-O6 | 1.925(5) | -0.0055 |
|  | OA-OA | 2.468(7) | -0.0009 | OA-OB | 2.892(8) | -0.0032 | 04-06 | 2.5707(9) | -0.0035 |
|  | OA-OA | 2.468(7) | -0.0009 | OA-OB | 2.433(8) | -0.0139 | 010-05 | 2.4754(9) | 0.0007 |
| Apical oxygen | Al1-OD | 2.076(3) | -0.0077 | Al1-OD | 1.954(7) | -0.0097 | M12-01 | 1.977(4) | 0.0004 |
|  |  |  |  |  |  |  | M12-O3 | 1.862(4) | -0.0037 |
| AI - AI | Al1-Al1 | 2.686(3) | -0.0034 | Al1-Al1 | 2.884(0) | -0.0034 | M12-M12 | 2.744(4) | -0.0046 |
|  | Al1'-Al1' | 2.686(3) | 0.0070 |  |  |  |  |  |  |

Note: Data for sillimanite and andalusite taken from Burt et al., (2004)
The rates of change of the distances are calculated with a linear fit from the length data over the whole pressure range and are recorded | as Å/GPa. Atoms labeled as in Figure 1 and Figure 4.

## Field Code Changed

## Formatted: Subscript

