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Approaching truly sustainable solar cells by the use of water and cellulose derivatives

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1	Approaching truly sustainable solar cells by the use of water and cellulose derivatives		
2			
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16			
17 18 19 20 21 22 23 24 25 26 27 28 29	Abstract: Aqueous dye-sensitized solar cells (DSSCs) are emerging as the first truly safe, cheap and eco-friendly photovoltaic technology, at the same time overcoming the well-known instability upon moisture/water contamination typical of many solar cells. While many aqueous DSSCs recently proposed still contain little amounts of organic solvents or oil-derived polymeric matrices, here we propose the first 100% hydrogel electrolyte, consisting of carboxymethylcellulose as green jellifying agent, water and iodide/triiodide redox mediator. Electrochemical and photoelectrochemical properties of the resulting electrolytes and solar cells are thoroughly investigated, with a special focus on the long-term stability of the aqueous devices under different operating and aging conditions. The obtained promising efficiencies and stabilities, combined with a metal-free sensitizer, lead here to sustainable, stable, transparent and building-integrable solar cells, without suffering any safety and/or toxicity issues.		
30 31	Keywords : Dye-sensitized solar cells; Aqueous electrolyte; Gel electrolyte; Cellulose; Sustainability.		
32			
33	Introdu	iction	
34 35	About 25 years ago O'Regan and Grätzel developed the dye-sensitized solar cell (DSSC), which was the first photovoltaic device based on (photo)electrochemical reactions. DSSC represented		

36 a sparkling revolution for the photovoltaic scientific community in terms of novel approaches

- 37 for solar energy conversion [¹,²]. An enormous research and development work has been
- 38 carried out in the last decades [³,⁴], and DSSCs are now the best choice for high efficiency solar

cells at low cost, having aesthetic features suitable for architectural integration, and providing
 full performance even under low irradiance conditions and/or indoor environments [⁵,⁶].

However, two main aspects have been not yet properly addressed so far, which is indeed 41 42 rather surprising if one considers that the DSSC was originally conceived to be an eco-friendly 43 bioinspired technology mimicking artificial photosynthesis [^{7,8,9}]. First, the standard electrolyte 44 system is based on volatile organic solvents, such as acetonitrile, methoxypropionitrile, being 45 also flammable and toxic, which certainly affects safety requirements upon module/panel 46 installation in real operating environment [¹⁰]. Secondly, either solid or quasi-solid polymer 47 electrolytes introduced so far for targeted stability enhancements are typically oil-derived [^{11, 12, 13, 14}], which in fact definitely clashes the original green and sustainable address for the 48

49 DSSC technology.

50 Recently green chemistry has contributed significantly to the step forwards of DSSC systems 51 into sustainability by developing devices operating in truly aqueous electrolyte environment 52 [10,¹⁵]. The use of water as the only solvent for the redox mediator would assure the realization of safe, cheap and eco-friendly solar cells [16,17], at the same time overcoming the 53 54 well-known instability of DSSC upon moisture/water contamination [18,19]. Aqueous DSSCs have recently scored an interesting 5.97% efficiency [²⁰]. Presently, research efforts are 55 56 focused on developing new water-soluble redox mediators [21, 22], improving the photoanode 57 wettability [²³], and synthesizing water-resistant organic sensitizers [²⁴, ²⁵]. Another 58 fundamental step towards stability enhancement is represented by the replacement of the 59 common liquid electrolytes by quasi-solid systems. To this purpose, different polymeric 60 matrices (pure gelatin [26], acrylic acid/gelatin [27] and xanthan gum [28]) are now investigated 61 by several research groups to jellify the aqueous redox mediator. Unfortunately, the acrylic 62 acid/gelatin-based electrolyte demonstrated rather limited stability of the initial power 63 conversion efficiency (PCE) (20% loss in the first 2 h of cell operation) [27], while the xanthan 64 gum-solidified solution cannot be classified as truly aqueous system as it contained 50% of 65 methoxypropionitrile [28]. To our knowledge, Spiccia's group proposed the best jellified 66 system so far, where gelatin (with a 1 wt% addition of PEG-300) was used to prepare a quasi-67 solid electrolyte providing 4.1% PCE, stable (80%) after 35 days under dark at room 68 temperature [26]. However, even if gelatin represents a suitable and sustainable matrix for 69 aqueous DSSCs, being a polypeptide made by hydrolytic degradation of collagen, this 70 electrolyte also contained polyethylene glycol (oil-derived), 1-methylimidazole (oil-derived, 71 harmful and corrosive) and a heavy metal (cobalt) based redox couple. 72 Considering such a scenario, here we present the first example of truly oil-derived, toxic, 73 corrosive and/or heavy metal-based components free biopolymer electrolyte for aqueous 74 DSSCs. In particular, we make use of the sodium salt of carboxymethylcellulose (CMC) as a

75 green jellifying agent for a 100% aqueous electrolyte based on the iodide/triiodide redox

76 mediator. The effect of the CMC-based gel composition on the electrochemical and

photoelectrochemical properties of the resulting electrolytes and solar cells is thoroughly

investigated, as well as the long-term stability of the aqueous devices under different

operating and aging conditions. We demonstrate how this newly proposed electrolyte system,

80 in conjunction with a metal-free sensitizer, easily leads to sustainable, stable, transparent and

81 building-integrable solar cells, without suffering any safety and/or toxicity issues.

83 Materials and Methods

84 Materials

- 85 Sodium iodide (Nal), potassium iodide (KI), iodine (I₂), sodium carboxymethyl cellulose (CMC,
- 86 $M_w = 250000 \text{ g mol}^{-1}$, chenodeoxycholic acid (CDCA), ethanol, acetone, *t*-butanol (*t*-BuOH)
- and acetonitrile (ACN) were purchased from Sigma-Aldrich. Deionized water (DI-H₂O, 18 M Ω
- 88 cm at 25 °C) was obtained with a Direct-Q 3 UV Water Purification System (Millipore).
- 89 Sensitizing dye 2-[{4-[4-(2,2-diphenylethenyl)phenyl]-1,2,3,3a,4,8b-
- 90 hexahydrocyclopento[b]indole-7-yl}methylidene]-cyanoacetic acid (D131) was purchased from
- 91 Inabata Europe S.A.
- 92 Fluorine-doped tin oxide (FTO) glass plates (sheet resistance 7 Ω sq⁻¹, purchased from
- Solaronix) were cut into 2 cm × 1.5 cm sheets and used as substrates for the fabrication of
 photoanodos and counter electrodes
- 94 photoanodes and counter electrodes.
- 95
- 96 Electrolytes: preparation and characterization
- 97 The experimental work related to the proposed aqueous electrolyte can be summarized in98 three sections.
- 99 In the first one, gel electrolytes containing aqueous Nal 0.50 M and various weight
- percentages (3.5 5 7.5 10 12.5 15 20) of CMC were prepared by weighting the
- 101 polymer in a vial and adding dropwise the proper amount of the iodide stock solution (Nal 0.50
- 102 M). Gel electrolytes were gently stirred overnight (18 h). In the case of the sample containing
- 103 20 wt% CMC, the gel was stirred in a warm bath setting the hot plate at 50 °C to get an
- 104 homogeneous mixture.
- 105 The resulting gels were tested for their ionic conductivity (σ) in lab-scale cells (ECC-Std, EL-CELL
- 106 GmbH) by means of electrochemical impedance spectroscopy (EIS), using a potentiostat
- 107 (PARSTAT 2273, Princeton Applied Research) in the range of frequencies between 10 kHz and 1
 108 Hz. Gels were sandwiched between two stainless steel blocking electrodes separated by a 100
- 109 μm-thick spacer featuring a round shape (14 mm inner diameter). The amplitude of the AC
- signal was 10 mV. The reference electrode was short-circuited with the counter-electrode in all
- 111 cases. The ionic conductivity was calculated according to:
- 112

113
$$\sigma = \frac{l}{A \cdot R_b}$$
 (Eq. 1)

114

115where I stands for the cell thickness, A is the area of the electrode in contact with the gel116electrolyte, and R_b is the bulk-resistance to ion diffusion. R_b was determined directly as the117value on the Z_{re} axes at $Z_{im} = 0$ in the Nyquist plot. For ionic conductivity measurements at118different temperatures, ECC-Std cells were placed in a climatic chamber (MK 53, BINDER119GmbH). After proper cell equilibration at 70 °C, the temperature was decreased by 10 °C120stepwise in the range from 70 to -25 °C. Each measurement was made 1 h and 30 min after

121 the chamber had reached the thermal equilibrium at the set temperature. The ionic

122 conductivity of aqueous Nal 0.50 M was also measured for comparison.

123 In a second phase of the study, and for a more in depth investigation of both the electrolyte 124 and the cathode/electrolyte interfaces, symmetrical (dummy) cells were assembled. Gel 125 electrolytes containing aqueous Nal 0.50 M, I₂ 30 mM and CMC 10 wt% (this value was chosen 126 as intermediate value within the investigated experimental domain) were prepared by 127 weighing the polymer in a vial and adding dropwise the proper amount of the aqueous 128 electrolyte solution. The gel electrolyte was gently stirred overnight (18 h) in a warm bath at 129 50 °C. The FTO-coated glass plates used for dummy cell assembly were platinized by a sputter 130 coater (Q180T ES by Quarum Technologies Ltd). A small amount of gel electrolyte was casted 131 onto one platinized electrode using a spatula. The dummy cell was then assembled clamping 132 the two electrodes separated by a spacer (with an internal round area of 0.785 cm² and a 70 133 μm thickness).

134 In the final part of the work, the cellulosic matrix was also used to jellify solutions with

optimized I^{-}/I_{3}^{-} content for aqueous DSSC application [²⁹]. In detail, both potassium- (KI 5.5 M

and I_2 50 mM) and sodium-based (NaI 4.5 M and I_2 50 mM) mixtures were used to prepare gel

electrolytes. For DSSCs application, these quasi-solid electrolytes were obtained by weighingthe redox mediator aqueous solution in a vial and adding the proper amount of biopolymer

under gentle stirring. CMC was added in 3.5, 4.5, 5.5, 6.5 and 7.5 wt% weight percentages.

140 After gelation, samples were left in a bath at 30-40 °C for 20 min to allow a proper

141 homogenization. Besides DSSCs fabrication, a dummy cell assembled as previously described

was used to measure the diffusion limited current by steady state cyclic voltammetry. The gel
 electrolyte tested contained aqueous KI 5.5 M, I₂ 50 mM and CMC 6.5 wt%. Its liquid

- 144 counterpart was tested under the same conditions for comparison. The diffusion coefficient *D*
- 145 of triiodide was calculated according to:
- 146

147
$$D = \frac{1}{2} \frac{d}{nFc_i} J_{\text{lim}}$$
 (Eq. 2)

148

149 where *d* is the distance between the two electrodes, *n* is the number of electrons involved in 150 the electrode reaction, *F* is the Faraday constant, c_i is the initial concentration of I_3^- and J_{lim} is 151 the diffusion-limited current density.

All electrolyte samples and cells were freshly prepared before each single testing and each ofthe experiments was repeated at least three times.

154

155 DSSC devices: fabrication and characterization

156 FTO covered glasses were rinsed with acetone and ethanol in an ultrasonic bath for 10 min.

157 Then, all traces of solvents were removed by a compressed air gun. Front electrodes were

prepared by depositing a single layer of porous TiO₂ on top of conductive substrates using a

screen printer equipped with a 43T mesh frame. After paste (18NR-T, Dyesol) deposition and

 $160 \qquad 20 \text{ min rest to let the paste relax, the TiO_2 layer was dried at 80 °C for 20 min and sintered}$

161 increasing the temperature up to 480 °C in 45 min. The fabricated photoanodes (with a 162 thickness of $\approx 6 \ \mu m$ and a covered area of 0.25 cm²) were finally activated by heating at 450 °C 163 for 20 min and, after cooling, soaked into a D131 dye solution (0.50 mM in t-BuOH:ACN 1:1). 164 CDCA 0.90 mM as coadsorbent in the dye solution was added. Dipping in dye solutions was 165 carried out at 22 °C for 5 h in the dark under shaking in a Buchi Syncore platform equipped 166 with a cooling plate. After dye-loading, the photoanodes were washed in acetone to remove 167 residual dye not specifically adsorbed on the TiO_2 layer. As regards the preparation of the 168 counter electrodes, FTO conductive glasses were platinized by spreading a H_2PtCl_6 solution (5 169 mM in ethanol) on the plate surface and heating. A thermoplastic Surlyn frame (3-4 mm large 170 and with an internal area of 0.8 cm × 0.8 cm) was used as spacer. The amount of gel electrolyte 171 required to fill a cell was determined assembling several trial cells. For each DSSC, a drop of gel 172 electrolyte (5.4-5.6 mg) was casted onto the platinized counter-electrode by a spatula. The 173 complete DSSC was then assembled by hot-pressing at 100 °C for 20 s. All DSSCs assembled in 174 this phase were stored in the dark for six days. After this period, one cell for each electrolyte 175 experimented was stored at 60 °C under dark condition. I-V curves of these cells were 176 periodically recorded right after the extraction from the oven and after 24 h of cooling at RT. 177 The photovoltaic performances were evaluated recording three consecutive I-V curves using a 178 Keithley 2420 Source Measure Unit. Cells were irradiated under simulated sun light at 1 sun 179 (AM 1.5G) after calibration using a silicon diode.

180

181 Results and Discussion

182 Electrochemical characterisation of CMC as jellifier for aqueous electrolytes

Here we attempt to exploit CMC, often cited as "cellulose gum", as a green jellifying agent for
aqueous DSSC electrolytes, which assure quasi-solid-like features. In CMC the hydroxyl groups
of the glucopyranose monomers of cellulose are bound with carboxymethyl groups; the
functional properties of the resulting product depend on the degree of substitution of the
initial cellulose structure (0.9 in the present study).

188 As a first experiment when proposing a novel macromolecular matrix for polymer electrolytes, 189 we assessed whether CMC at $M_{\rm w}$ = 250000 g mol⁻¹ could behave as a proper jellifier for 190 aqueous electrolytes. To this purpose, an aqueous solution containing NaI 0.5 M was jellified 191 by CMC in various amounts up to 20 wt% (Fig. 1A-G). In this phase, I_2 was not yet introduced 192 since our aim was that of investigate the proposed hydrogel step-by-step by adding a 193 component at a time. All samples were clearly homogeneous, and showed enhanced 194 viscoelastic characteristics upon increasing the jellifier content; above 5 wt%, we clearly 195 experienced the proper gelation of the electrolytes that formed a homogeneous sticky gel well 196 consistent and self-standing even when overturning the vessel (Fig. 1C). The homogeneity was 197 retained up to 20 wt%; samples at higher CMC content were excluded from our investigation.

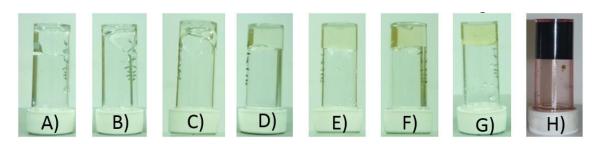


Fig. 1. Gel electrolytes containing aqueous Nal 0.5 M and 3 (A), 5 (B), 7.5 (C), 10 (D), 12.5 (E),
15 (F) and 20 wt% (G) CMC. The image of a gel electrolyte containing aqueous Nal 0.5 M, I₂ 30
mM and CMC 10 wt% is also shown (H).

203

The promising prospects of the obtained materials as electrolytes for DSSCs were determined by ionic conductivity (σ) measurements through EIS analysis. Gels containing CMC in the 2.5-20 wt% range displayed very similar σ values, in the range of 10 mS cm⁻¹ at ambient temperature (**Fig. 2A, Table S1**). This value is lower with respect to that (48.4 mS cm⁻¹) of the liquid counterpart, which suggests, as expected, a slightly restricted ion mobility in the gel electrolytes. Anyway, it is worth noting that our hydrogel electrolyte shows room temperature σ values one order of magnitude higher than pure ionic liquids, such as 1-hexyl-3-

211 methylimidazolium iodide (HMII), which was successfully proposed in efficient DSSCs [³⁰].

212 Gels containing 5, 10 and 20 wt% of CMC were further tested for their ionic conductivity 213 between 70 and -25 °C (Fig. 2B, Table S2), to cover the practical operational conditions of a DSSC. The variation of Log σ as a function of T^{-1} in the range from 70 to -10 °C was barely 214 215 compatible with the linear correlation predicted by the Arrhenius model. On the other hand, a 216 dramatic decrease of σ was observed below –10 °C, likely ascribable to water solidification. As 217 a result, we can confidently assume that our proposed biosourced aqueous electrolyte can be 218 safely employed in DSSCs operating in places where the temperature frequently drops below 219 zero, without affecting the charge transport properties up to -10 °C.

220 Overall, the data collected suggest that supply of I⁻ ions at the photoanode is likely not limited 221 by the slow diffusion through the three-dimensional CMC network in a wide range of 222 temperatures. Indeed, efficient I⁻ ions supply is a fundamental requirement to ensure fast 223 regeneration, since it is well known that a slow regeneration with respect to the rate of back 224 electron transfer from the semiconductor to the oxidized dye molecules would affect the photovoltaic performance [2,^{31,32}]. Evidences of this phenomenon ascribed to the low ionic 225 226 conductivity of some classes of polymer electrolytes were reported by Nogueira et al. [33], who 227 showed that the ionic conductivity of polymeric systems increased up to $\approx 10^{-5}$ S cm⁻¹ with 228 increasing [Nal] values [³⁴]. However, one should consider that this value is lower by two 229 orders of magnitude when compared to the one we obtained here with the water-based 230 Nal/CMC gel electrolytes.

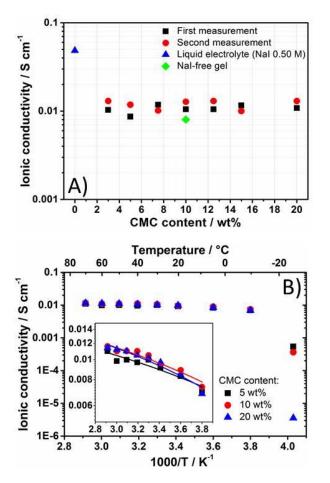


Fig. 2. A) lonic conductivity at ambient temperature of aqueous Nal 0.50 M gelled with different amounts of CMC. The ionic conductivity of pristine aqueous Nal 0.50 M and of a Nalfree gel containing 10 wt% CMC are also shown for comparison; B) lonic conductivity, in the temperature range of -25/+70 °C, of aqueous Nal 0.50 M gels containing different amounts of CMC. The inset shows data points up -10 °C fitted as lines. The corresponding R^2 values are 0.87 (5 wt%), 0.93 (10 wt%) and 0.95 (20 wt%).

239

240 A self-standing gel composed of CMC 10 wt%, aqueous Nal 0.50 M and I_2 30 mM (Fig. 1H) was 241 used to fill a dummy cell and investigated in depth by means of EIS analysis. It is worth noting 242 that this gel electrolyte was not fully homogeneous; indeed, small brownish aggregates (also 243 recalled and shown later on in the followings) randomly appeared in the sample likely 244 ascribable to the formation of polymer-I₂ complexes. The formation of polymer-I₂ complexes 245 is common in many cellulose derivatives and other natural and synthetic polymers; indeed, 246 literature studies showed that binding sites were found to be located in the amorphous 247 regions [³⁵]. Anyway, the slight inhomogeneity did not affect at all the electrochemical 248 performance of the sample, which remained constant upon time. The value of *R*_{Pt} from the impedance measurement was lower than $1 \Omega \text{ cm}^2$, similarly to what observed in the case of 249 250 liquid Nal 1.50 M + I_2 50 mM (**Fig. 3**).

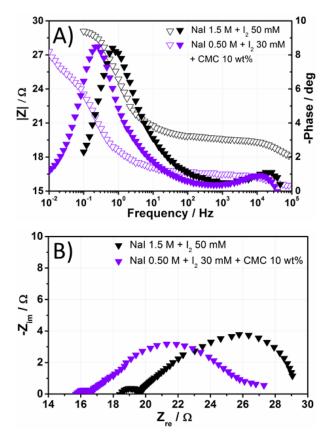


Fig. 3. A) Bode (modulus = open triangles; phase = filled triangles) and B) Nyquist plots of a
dummy cell filled with aqueous Nal 0.50 M, I₂ 30 mM and CMC 10 wt%. The spectra of a
dummy cell filled with aqueous Nal 1.5 M + I₂ 50 mM are shown for comparison.

256

257 *Effect of CMC content on the photovoltaic performance of aqueous DSSCs*

The data reported in the previous paragraph suggest that CMC-containing electrolytes are
 likely to be successfully applied in truly aqueous DSSCs, and the faint inhomogeneity (brownish
 spots) in the gels containing both I₂ and NaI does not affect significantly the electrochemical
 performance.

The most performing aqueous liquid electrolytes (according to our previous work focused in the unravelling of the I^{-}/I_{3}^{-} chemistry in water [29]) were used to prepare gel electrolytes at various CMC amounts. These starting electrolyte solutions contained I_{2} 50 mM and NaI 4.5 M or I_{2} 50 mM and KI 5.5 M. When CMC 10 wt% was introduced in these liquid electrolytes, the gel point was surpassed in both cases. Thus, we added CMC amounts in the range of 3.5 (viscous fluid) to 7.5 wt% (self-standing gel).

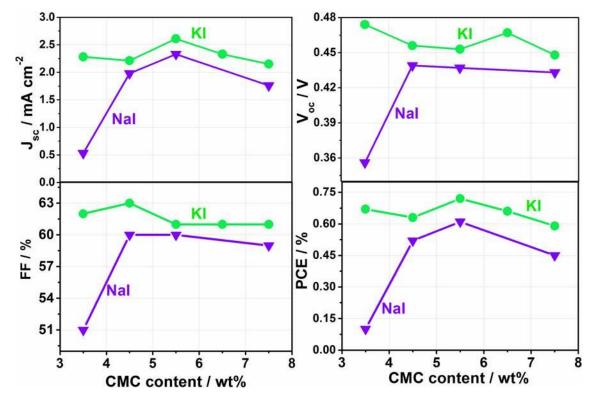
The photovoltaic parameters referred to the best performances of the solar cells co-sensitized with D131 and CDCA (D131 : CDCA = 1 : 1.8) are shown in **Fig. 4** and **Table S3**. Data were

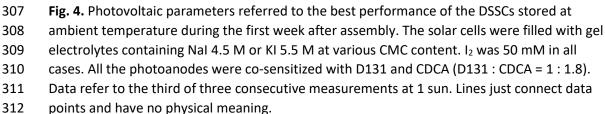
270 selected among those recorded during the first week of dark storage at ambient temperature

- 271 (all results are shown in **Tables S4-S13**). The cells filled with the electrolyte containing CMC 6.5
- 272 wt% and Nal were excluded because of massive air infiltration during sealing by hot-pressing.

273 It was noted that the short circuit photocurrent density (J_{sc}) was the main factor affecting 274 efficiency. With both Nal- and KI-containing electrolytes, the best photovoltaic performances 275 were achieved at CMC 5.5 wt%. The best efficiencies reached by the gels containing KI were 276 similar at different CMC content. This is likely ascribable to the fact that the ionic conductivity 277 of the gels containing iodide and CMC did not change as a function of the cellulose content. 278 When discussing these PCE values, it must also be noticed that no additives were used to 279 selectively improve the photovoltaic parameters; indeed, our purpose is to evaluate the 280 characteristics of a pure aqueous electrolyte upon addition of the natural cellulose-based 281 jellifier. Moreover, almost none of the additives conventionally used in the DSSC field, such as 282 4-tert-butylpyridine (TBP) and N-methylbenzimidazole (NMBI), are soluble in water. In this 283 regard, it is rather surprising that some research groups recently proposed TBP- and NMBI-284 based aqueous DSSCs [³⁶,³⁷]; in our laboratories we were not able to solubilize them in water, 285 even in the presence of surfactants.

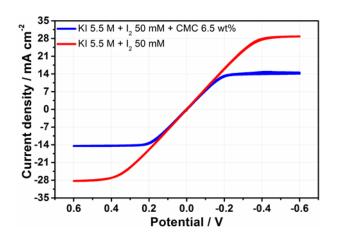
286 By comparing the photovoltaic parameters of this work with our recent outcomes in the field 287 of aqueous liquid electrolytes [29], it emerges that the efficiency values reached by the cell 288 containing CMC and Nal were about 0.1% lower than those achieved with CMC and KI. We also 289 noticed that the J_{sc} value achieved by the cell containing NaI and CMC 5.5 wt% was higher than 290 its liquid counterpart, thus confirming the excellent transport properties of the quasi-solid 291 electrolyte proposed in this work. In all of the other cases, J_{sc} and open-circuit potential (V_{sc}) 292 values were lower in gel electrolytes. In particular, V_{oc} values achieved with CMC and NaI were 293 significantly low. Moreover, such a reduction of V_{oc} values in the gel electrolytes was 294 somewhat unexpected. Actually, CMC hydrogels are supposed to be basic because of the 295 protonation of carboxylic moieties leading to the release of hydroxyl groups. Under this 296 condition, a shift of the flat band potential of the semiconductor towards negative potentials 297 vs. SHE is expected, with a consequent increase of V_{oc} values. Anyway, the flat band potential 298 also depends upon the surface dipole moment due to the anchored dye layer, which can be 299 affected by the interactions with the electrolyte. The increase of dark current in the solar cells 300 filled with gel electrolytes would explain the decrease of V_{oc} when compared to the liquid 301 counterparts. Anyway, under this condition, the fill factor (FF) should be decreased as well, 302 and this was not found to be the case. Sealing problems may have played a role, as well; 303 however, we may almost completely exclude this hypothesis as the long-term stability of these 304 cells is excellent (see the followings).

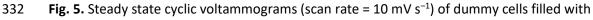




306

314 Fig. 5 (blue curve) shows the steady state cyclic voltammogram of the gel electrolyte 315 containing CMC 6.5 wt% and KI, as representative sample that also allowed the solar cells to 316 retain most of their efficiency over time, both at ambient temperature and at 60 °C. The 317 diffusion limited current density in the gel electrolyte is nearly halved as compared to that of 318 the liquid electrolyte (red curve). Anyway, its value is much higher ($\approx 15 \text{ mA cm}^{-2}$) than the 319 photocurrent generated by the cells. This result suggests that mass transport through the 320 electrolyte is not supposed to limit the photocurrent, unless diffusion is somewhat restricted 321 inside the nanostructured semiconductor layer. The diffusion coefficient of triiodide was 322 calculated assuming $[I_3^-] = 50 \text{ mM}$ is $5.2 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$. It is comparable with the values 323 reported by Hauch and Georg for liquid nitrile-based electrolytes [38] and, in fact, one order of 324 magnitude higher to those obtained with aqueous gels based on amphiphilic copolymers such 325 as Pluronic F77 and P123 mixed with an aqueous solution of 1-butyl-3-methyl imidazolium 326 iodide (BMII) [37]. Such a high value can be also due to the fact that, at high I[–] concentration 327 such as those experimented here, sizeable amounts of polyiodides are supposed to be present 328 in the electrolyte solution. This fact may promote the Grotthuss-type charge transfer 329 mechanism in the gel electrolytes [³⁹].





333 aqueous liquid and gel electrolytes.

334

335 Long-term stability of lab-scale DSSCs based on aqueous gel electrolyte

336 Fig. 6 shows the photovoltaic parameters of the solar cells stored at ambient temperature. At 337 low CMC contents of 3.5 and 4.5 wt%, the efficiency values slightly increased over time both in 338 Nal- and KI-based devices. At high CMC contents, the cell performance worsened over time in 339 the case of KI-containing gels. Overall, the most stable cell from the KI pool (CMC 6.5 wt%) 340 retained 85% efficiency after about one month. The conversion efficiency of the cells filled 341 with Nal-based gels at high CMC content fluctuated over time, but the values observed after 342 about one month were similar to those recorded right after cell assembly. The most efficient 343 cell from the Nal pool retained 93% efficiency at the end of the monitoring period. This is a 344 truly impressive result if one considers that other biosourced polymer matrices (e.g., gelatin) 345 reported in the literature showed poor stability, with >20% loss in PCE in about 2 h after cell 346 assembly [27].

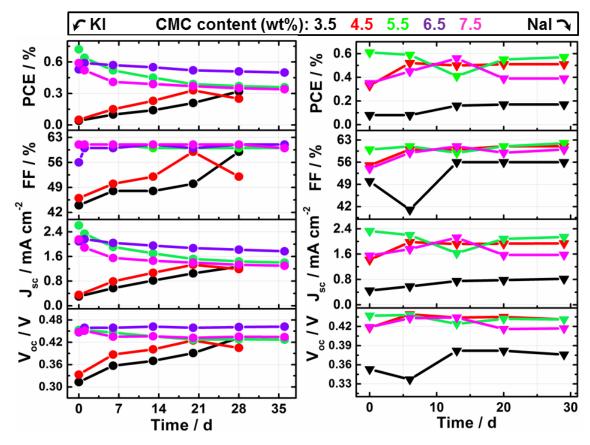
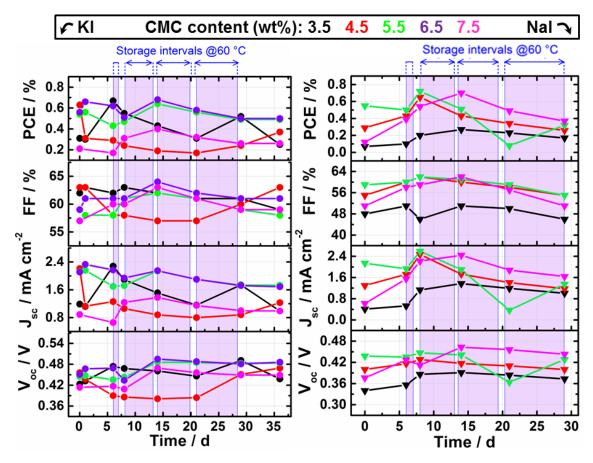


Fig. 6. Photovoltaic parameters versus time of the solar cells co-sensitized with D131 and CDCA
(D131 : CDCA = 1 : 1.8) stored in the dark at ambient temperature. The solar cells contained KI
5.5 M + I₂ 50 mM (left panel) or NaI 4.5 M + I₂ 50 mM (right panel), jellified with different
amounts of CMC. Data points refer to the third of three consecutive measurements at 1 sun.
Lines just connect data points and have no physical meaning.

354

355 The stability of our aqueous solar cells was more deeply investigated by inducing a thermal 356 stress at 60 °C in a 35 days aging test. The photovoltaic performances of the cells stored at 60 357 °C fluctuated over time (Fig. 7, Tables S4-S13). The highest efficiency values after ageing were 358 recorded at 5.5 and 6.5 wt% CMC with KI-containing electrolytes. In most cases, efficiency 359 values peaked during the first week of storage at 60 °C. Exceptions are represented by the cells 360 filled the KI containing electrolytes at 3.5 and 4.5 wt% CMC. This may be due to better pore 361 filling due to both the rearrangement of the biosourced macromolecules and the lowered 362 viscosity of the gel promoted by heat. Actually, the slight inhomogeneity of the gels appeared 363 to be reduced right after cell extraction from the oven. The brownish spots reappeared after 364 cooling at ambient conditions; anyway, their presence surprisingly did not detrimentally affect 365 both the performance and the stability of the cells, as for the data plotted in Fig. 7.



367

Fig. 7. Photovoltaic parameters of the solar cells co-sensitized with D131 and CDCA (D131:
 CDCA = 1 : 1.8) stored in the dark at 60 °C versus storage time. The solar cells contained KI 5.5
 M + I₂ 50 mM (left panels) or NaI 4.5 M + I₂ 50 mM (right panels) jellified with different
 amounts of CMC. The time intervals during which the cells have been stored at 60 °C are
 represented by filled coloured areas. The cells were cooled at ambient temperature for 24 h
 before recording the J-V curves. Data points refer to the third of three consecutive
 measurements at 1 sun. Lines just connect data points and have no physical meaning.

The efficiency values recorded about one month after cell assembly and ageing at 60 °C were
similar to those recorded during the first week in the case of the less performing cells. As
regards the photovoltaic performance of the cells that displayed efficiency values ≥0.5%,
some of them were found to slightly decrease during the same period. Anyway, devices
demonstrated proper operation during the whole test, thus accounting for the stable
performance of the proposed electrolyte.

Upon observation of the devices subjected to the aging test, dye desorption is likely one of the causes of the slight J_{sc} decrease observed for some of the cells. Indeed, the discoloration of the active area was clearly observed in the cell filled with the electrolyte containing CMC 4.5 wt% and KI (**Fig. 8**, top). In some other cases, the electrolyte partially depleted over time near the active area (**Fig. 8**, bottom). This typically arises from an unsuitable sealing, which may result in gel deswelling (leading to the leakage of the liquid phase). In the present work, we rarely experienced electrolyte traces out of the active area; therefore, we hypothesised that the

- CMC-based gel penetrated the semiconductor pores left unfilled after cell sealing. Indeed, it is
 well known in the literature that DSSC filling with a gel electrolyte is rather tricky, especially
 when the gel consistence renders it impossible to introduce it by vacuum-filling. Therefore,
- electrolyte deposition must be performed manually or by screen-printing, and the proper
- dosage of gel amount in the first case (like in our study) is difficult to be pursued.
- 394



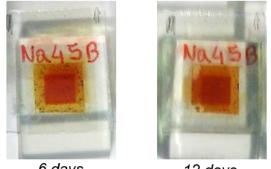




Just assembled

12 days at 60 °C

Nal 4.5 M + I₂ 50 mM + CMC 4.5 wt%



6 days after assembly 12 days at 60 °C

- Fig. 8. Top: cell K45A filled with aqueous KI 5.5 M + I₂ 50 mM + CMC 4.5 wt% after cell
 assembly and after 12 days of storage at 60 °C. Bottom: cell Na45B filled with aqueous NaI 4.5
 M + I₂ 50 mM + CMC 4.5 wt% 6 days after cell assembly and after 12 days of storage at 60 °C.
 Both the cells were cosensitized with D131 and CDCA (D131: CDCA = 1 : 1.8).
- 400

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401 Overall, the most important outcome of this experiment is the truly impressive stability of the 402 CMC-based aqueous electrolytes upon thermal aging. To the best of our knowledge, this is the 403 first report showing the long-term (≈800-875 h) stability of an aqueous DSSC upon thermal 404 aging at 60 °C; the only study on 100% aqueous DSSCs stressed under heat shows an aging test 405 limited at 200 h [40]. Thus, we here demonstrate that CMC can successfully jellify the aqueous 406 solution of the redox mediator, at the same time ensuring its effective trapping within the 407 macromolecular matrix without affecting the diffusion of the shuttle ions between the 408 electrodes in the cell.

410 Conclusions

- 411 CMC hydrogels have demonstrated suitable characteristics as electrolytes in 100% aqueous
- 412 DSSCs. Hydrogels containing both NaI and I₂ allowed for efficient charge transfer at the
- 413 Pt/electrolyte interface. The ionic conductivity of Nal-containing hydrogels and the diffusion
- 414 coefficient of I_3^- ions were slightly lower than in the liquid counterpart, anyway they were
- 415 comparable and/or higher than the recent literature reports on ionic liquids and water-based
- 416 gel electrolytes. The truly promising electrochemical characteristics suggested that the
- 417 photocurrent generation should not be limited by slow diffusion of I^- or I_3^- ions, as for an ionic
- 418 mobility kept constant even down to -10 °C.
- 419 The newly developed biosourced aqueous gel electrolytes were tested in lab-scale solar cells 420 and the best photovoltaic performances (0.72%) were achieved in the presence of CMC 5.5 421 wt%. In this work, no additives were added to the electrolyte and no surface treatment of the 422 photoanodes was performed, in order to keep the whole system as simple, sustainable and 423 cheap as possible; this also means that there is much room for improving PCE values. The most 424 stable cell demonstrated a remarkable 93% efficiency retention after 29 days of dark storage; 425 the CMC-based devices also showed a robust stability when subjected to a further month of 426 thermal aging at 60 °C in oven. We found a few difficulties when manually dosing the exact 427 amount of quasi-solid electrolyte in the cell, therefore we think that future works should be 428 addressed towards in situ gelation of cellulose derivatives, that surely would allow reducing 429 the impact of sealing and ensuring an in depth pore filling.
- 430 Overall, we here demonstrated that photovoltaic cells can be fabricated with (heavy) metal-
- 431 free redox shuttles and dye sensitizers, organic solvent-free electrolytes, bioderived cellulosic
- 432 gellifiers by simple and cost-effective preparation procedures, thus opening the way to the
- 433 next-generation of green solar energy conversion.

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435 **References**

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[¹] B. O'Regan and M. Grätzel, *Nature*, 1991, **353**, 737-740.

- ^{[2}] M. Grätzel, J. Photochem. Photobiol., C, 2003, 4, 145-153.
- [³] A. Hagfeldt, G. Boschloo, L. Sun, L. Kloo and H. Pettersson, *Chem. Rev.*, 2010, **110**, 6595-6663.
- [⁴] M. Grätzel, J. Photochem. Photobiol., C, 2004, 164, 3-14.
- ⁵] F. Bella, *Electrochim. Acta*, 2015, **175**, 151-161.
- [⁶] F. Bella, S. Galliano, C. Gerbaldi and G. Viscardi, *Energies*, 2016, 9, art. no. 384.

[⁷] S. L. Chen, A. C. Xu, J. Tao, H. J. Tao, Y. Z. Shen, L. M. Zhu, J. J. Jiang, T. Wang and L. Pan, *Green Chem.*, 2016, **18**, 2793-2801.

[⁸] M. Ghavre, O. Byrne, L. Altes, P. K. Surolia, M. Spulak, B. Quilty, K. R. Thampi and N. Gathergood, *Green Chem.*, 2014, **16**, 2252-2265.

[⁹] S. Maniam, A. B. Holmes, J. Krstina, G. A. Leeke and G. E. Collis, *Green Chem.*, 2011, **13**, 3329-3332.

[¹⁰] F. Bella, C. Gerbaldi, C. Barolo and M. Grätzel, *Chem. Soc. Rev.*, 2015, **44**, 3431-3473.

[¹¹] R. Shanti, F. Bella, Y. S. Salim, S. Y. Chee, S. Ramesh and K. Ramesh, *Mater. Des.*, 2016, **108**, 560-569.

[¹²] M. Gerosa, A. Sacco, A. Scalia, F. Bella, A. Chiodoni, M. Quaglio, E. Tresso, S. Bianco, *IEEE J. Photovoltaics*, 2016, **6**, art. no. 7384432.

[¹³] J. Y. Lim, J. K. Kim, J. M. Lee, D. Y. Ryu and J. H. Kim, *J. Mater. Chem. A*, 2016, **4**, 7848-7858.

[¹⁴] N. H. Ming, S. Ramesh and K. Ramesh, *Sci. Rep.*, 2016, **6**, art. no. 27630.

[¹⁵] Q. Dai and J. Rabani, *Chem. Commun.*, 2001, 2142-2143.

[¹⁶] H. Choi, B. S. Jeong, K. Do, M. J. Ju, K. Song and J. Ko, *New J. Chem.*, 2013, **37**, 329-336.

[¹⁷] C. Law, O. Moudam, S. Villarroya-Lidon and B. C. O'Regan, *J. Mater. Chem.*, 2012, **22**, 23387-23394.

[¹⁸] Y. Liu, A. Hagfeldt, X. R. Xiao and S. E. Lindquist, *Sol. Energy Mater. Sol. Cells*, 1998, **55**, 267-281.

[¹⁹] T. A. G. Risbridger, F. A. Castro and P. J. Cameron, *J. Phys. Chem. C*, 2012, **116**, 22253-22260.

[²⁰] R. Y. Y. Lin, F. L. Wu, C. T. Li, P. Y. Chen, K. C. Ho and J. T. Lin, *ChemSusChem*, 2015, **8**, 2503-2513.

[²¹] H. Ellis, R. Jiang, S. Ye, A. Hagfeldt and G. Boschloo, *Phys. Chem. Chem. Phys.*, 2016, **18**, 8419-8427.

[²²] W. Yang, M. Söderberg, A. I. K. Eriksson and G. Boschloo, *RSC Adv.*, 2015, 5, 26706-26709.

[²³] C. Dong, W. Xiang, F. Huang, D. Fu, W. Huang, U. Bach, Y. B. Cheng, X. Li and L. Spiccia, *Angew. Chem. Int. Ed.*, 2014, **53**, 6933-6937.

[²⁴] R. Cisneros, M. Beley, F. Lapicque and P. C. Gros, *Eur. J. Inorg. Chem.*, 2016, **2016**, 33-39.

[²⁵] R. Y. Y. Lin, T. M. Chuang, F. L. Wu, P. Y. Chen, T. C. Chu, J. S. Ni, M. S. Fan, Y. H. Lo, K. C. Ho and J. T. Lin, *ChemSusChem*, 2015, 8, 105-113.

[²⁶] W. Xiang, D. Chen, R. A. Caruso, Y. B. Cheng, U. Bach and L. Spiccia, *ChemSusChem*, 2015, **8**, 3704-3711.

[²⁷] S. Zhang, G. Y. Dong, B. Lin, J. Qu, N. Y. Yuan, J. N. Ding and Z. Gu, *Sol. Energy*, 2016, **127**, 19-27.

[²⁸] S. J. Park, K. Yoo, J. Y. Kim, J. Y. Kim, D. W. Lee, B. Kim, H. Kim, J. H. Kim, J. Cho and M. J. Ko, *ACS Nano*, 2013, **7**, 4050-4056.

[²⁹] F. Bella, S. Galliano, M. Falco, G. Viscardi, C. Barolo, M. Grätzel and C. Gerbaldi, *Chem. Sci.*, 2016, **7**, 4880-4890.

[³⁰] N. Papageorgiou, Y. Athanassov, M. Armand, P. Bonho, H. Pettersson, A. Azam and M. Grätzel, *J. Electrochem. Soc.*, 1996, **143**, 3099-3108.

[³¹] B. E. Hardin, H. J. Snaith and M. D. McGehee, *Nat. Photonics*, 2012, **6**, 162-169.

[³²] Z. Ning, Y. Fu and H. Tian, *Energy Environ. Sci.*, 2010, **3**, 1170-1181.

[³³] A. Nogueira, M. De Paoli, I. Montanari, R. Monkhouse, J. Nelson and J. Durrant, J. *Phys. Chem. B*, 2001, **105**, 7517-7524.

[³⁴] A. Nogueira, M. Spinace, W. Gazotti, E. Girotto and M. De Paoli, *Solid State Ionics*, 2001, **140**, 327-335.

[³⁵] S. Moulay J. Polym. Eng., 2013, **33**, 389-443.

[³⁶] W. Xiang, F. Huang, Y. B. Cheng, U. Bach and L. Spiccia, *Energy Environ. Sci.*, 2013, **6**, 121-127.

[³⁷] S. S. Soni, K. B. Fadadu, R. L. Vekariya, J. Debgupta, K. D. Patel, A. Gibaud and V. K. Aswal, *J. Colloid Interface Sci.*, 2014, **425**, 110-117.

[³⁸] A. Hauch and A. Georg, *Electrochim. Acta*, 2001, **46**, 3457-3466.

^[39] N. Agmon, Chem. Phys. Lett., 1995, **244**, 456-462.

[⁴⁰] H. J. Son, C. Prasittichai, J. E. Mondloch, L. Luo, J. Wu, D. W. Kim, O. K. Farha and J. T. Hupp, *J. Am. Chem. Soc.*, 2013, **135**, 11529-11532.