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Intrusion-Extrusion of Electrolyte Aqueous Solutions in Pure Silica Chabazite by in Situ High Pressure Synchrotron X-ray Powder Diffraction

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Intrusion-Extrusion of Electrolyte Aqueous Solutions in Pure Silica Chabazite by in situ High Pressure Synchrotron X-ray Powder Diffraction

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3 **Intrusion-Extrusion of Electrolyte Aqueous Solutions in Pure Silica Chabazite by *in situ* High**
4 **Pressure Synchrotron X-ray Powder Diffraction**
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6 Giorgia Confalonieri^a, Andrey Ryzhikov^{b,c*}, Rossella Arletti^{a*}, Habiba Nouali^{b,c},
7
8 Simona Quartieri^d, T. Jean Daou^{b,c}, Joël Patarin^{b,c}
9

10
11 ^aDipartimento di Scienze della Terra, Università di Torino, Via Valperga Caluso 35, Torino, Italy

12
13 ^bUniversité de Haute Alsace (UHA), Axe Matériaux à Porosité Contrôlée (MPC), Institut de Science
14 des Matériaux de Mulhouse (IS2M), UMR CNRS 7361, 3 bis rue Alfred Werner, F-68093 Mulhouse,
15 France.
16

17
18 ^c Université de Strasbourg, Strasbourg, France
19

20 ^dDipartimento di Scienze Matematiche e Informatiche, Scienze Fisiche e Scienze della Terra,
21 Università di Messina, Viale Ferdinando Stagno d'Alcontres 31, 98166 Messina S. Agata, Italy
22
23
24

25 Corresponding Authors

26
27 Rossella Arletti

28
29 *E-mail: rossella.arletti@unito.it

Phone:+39 0116705129

Fax:+390116705128
30

31
32 Andrey Ryzhikov

33
34 E-mail: andrey.ryzhikov@uha.fr

Phone: +33 389336754

Fax: +33 389608799
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Abstract

Structural investigation of the high pressure intrusion/extrusion of different electrolyte aqueous solutions (NaCl, NaBr and CaCl₂) with different concentrations (2M and 3M) in a pure-silica chabazite was carried out. *In situ* synchrotron X-ray powder diffraction experiments were performed in the pressure range of 0.12 - 2.6 GPa and upon pressure release, in order to unravel the interactions among intruded species and host material. The energetic performance of the systems were determined by porosimetric studies. Results show that cation in the salt seems to influence the intrusion-extrusion pressures, whereas the structural evolutions, undergone by the systems upon pressure-induced intrusion, are essentially independent on the nature of the penetrating media. Moreover, the initial electrolyte concentration seems to influence only the value of the intrusion pressure, but neither the amount nor the interaction mode of the intruded species. Both water and salt molecules enter the pores and the penetration of comparable extra-framework volumes occurs at similar pressure values. However, the composition of intruded species is different from that of initial solution and depends on applied pressure that reinforces the hypothesis on ion desolvation under penetration into the pores. After pressure release, pure-silica chabazite intruded by NaCl and NaBr aqueous solutions does not recover the initial cell volume and partially retains the intruded extra-framework species. On the contrary, the zeosil intruded by CaCl₂ recovers the original cell parameters. These differences have been structurally interpreted on the basis of the electrolyte/zeolite interactions. Interestingly, the extrusion behavior results to be mainly determined by the interactions of the anion with silanol defects of chabazite framework, rather than by the coordination bonds of the cation with the framework oxygen atoms.

1. Introduction

One of the promising technologies of mechanical energy absorption, storage and generation is based on forced penetration of a nonwetting liquid in porous solids¹⁻¹⁰, where the energy of applied pressure is converted into the energy of solid-liquid interface. First “porous solid/nonwetting liquid” systems – called also heterogeneous lyophobic systems (HLS) – were developed by V. Eroshenko and co-workers on the base of porous silica^{1, 2, 11, 12}. Since 2001, pure-silica zeolites (zeosils) were proposed for this kind of application¹³. Since then, many zeosil-based heterogeneous lyophobic systems were studied¹⁴⁻²⁴. Due to the presence of highly hydrophobic

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3 micropores with very small openings, these materials demonstrate high energetic performance
4 under water intrusion, with intrusion pressures and stored energies up to 180 MPa and 15 J/g,
5 respectively ¹⁷. Depending on the zeosil structure [pore system type (cage or channels), its
6 dimensionality, size, presence of defects], the “zeosil/water” systems are able to restore, dissipate
7 or absorb the supplied mechanical energy accumulated during the compression step that
8 corresponds to a spring, shock absorber and bumper behavior, respectively.
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13 The use of saline solutions instead of water was found to be an effective way to improve the
14 energetic performance of heterogeneous lyophobic systems due to a considerable increase of
15 intrusion pressure with salt concentration ²⁵⁻³¹. This increase is particularly pronounced in the case
16 of zeosils, where a huge rise of intrusion pressure up to 7.4 times can be achieved for LTA-type
17 zeosil using LiCl saturated solution (20 M) instead of water ³². The highest ever stored energy value
18 observed for zeosil-based heterogeneous lyophobic systems (93 J/g) shows the importance of
19 using electrolyte solutions from an applied point of view ³¹.
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25 Several hypotheses were proposed to explain this increase: (i) rise of liquid-solid interfacial tension
26 ³³, confinement effect of nanopore walls ^{34, 35}, osmotic phenomenon ³⁶ and ion desolvation ²⁷. The
27 last hypothesis, focusing on the distortion and desolvation of the water/ion system inside the
28 micropores, seems to be the most pertinent, since the strong rise of intrusion pressure with
29 electrolyte concentration cannot be explained by a simple increase of interface tension or osmotic
30 phenomena ³⁶. The desolvation phenomenon was confirmed by our previous study on the
31 intrusion of MgCl₂ solution (MgCl₂•21 H₂O) in FER-type zeosil, performed by *in situ* X-ray powder
32 diffraction (XRPD) under high pressure (HP). It was shown that the intruded liquid was neither
33 water nor the initial diluted solution, but close to the saturated one (MgCl₂•10 H₂O) ³⁷.
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41 The structural studies on the evolution of the zeosil systems under water or salt solution intrusion,
42 and the analysis of the intruded species arrangement inside the pores are of great interest since
43 they should allow to better understand, from an atomistic point of view, the mechanisms of
44 intrusion-extrusion of electrolyte solutions in zeosils, which remain still unrevealed. The high
45 potentialities of the structural studies, in particular by XRPD, on porous materials compressed with
46 penetrating media have been clearly demonstrated by several investigations performed by many
47 authors on a series of HP-induced phenomena (*e.g.* pressure-induced over-hydration, penetration,
48 nanoconfinement and self-organization of guest molecules, pressure-induced deformations of the
49 host matrix etc.)³⁸⁻⁴².
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3 Understanding the behavior of confined species in these systems is extremely important for both
4 technological applications (*e.g.* water desalination, ions and molecules separation, heterogeneous
5 catalysis, adsorption, transport through biological membrane etc.) and advances in fundamental
6 science.
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9 In the present work, the HP intrusion-extrusion of electrolyte aqueous solutions (NaCl, NaBr and
10 CaCl₂ at different concentrations) in a pure silica chabazite is studied by *in situ* synchrotron XRPD
11 in order to unravel the nature and the location of the intruded species in zeosil pores. The results
12 are compared with those obtained by intrusion-extrusion tests performed by porosimeter. The
13 pure-silica chabazite (CHA-type zeosil, Si-CHA from now on) was chosen because of its 3D pore
14 system and relatively simple structure ($R\bar{3}m$ space group). The framework can be described as an
15 ABC sequence of double 6-rings (D6R) of tetrahedra linked together through single 4-rings⁴³.
16 These building units define cages with high volume characterized by small 8MR (diameter of 0.38
17 nm) pore openings. It should be noticed that the HLS based on Si-CHA showed earlier promising
18 energetic performance under the intrusion-extrusion of LiCl aqueous solutions³⁰.
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28 **2. Experimental and Materials**

29 **2.1. Zeolite synthesis and characterization**

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31 The Si-CHA sample was synthesized in fluoride medium according to the procedure described in
32 the earlier works^{30,44}. The N,N,N-trimethyladamantammonium (TMAda⁺) was used as a structure-
33 directing agent. The starting gel with molar composition 1SiO₂:0.5TMAdaOH: 0.5HF: 4.7H₂O was
34 introduced in a Teflon-lined stainless-steel autoclave and heated at 150 °C for 5 days. After
35 hydrothermal synthesis, the product was filtered, washed with distilled water and ethanol, and
36 dried at 70 °C overnight. Then the solid was calcined at 600 °C for 8 hours to completely remove
37 the organic template.
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39 A P_{amb} data collection at ambient pressure (P_{amb}) on the synthesized material was carried out at
40 XPRESS beamline at ELETTRA (Trieste, Italy) using a LaB₆ calibrated wavelength of 0.4957 Å. A
41 structural refinement was performed starting from the coordinates taken from Diaz-Cabanas *et al.*
42 1998⁴⁴. The material resulted to be a pure Si-CHA sample with a formula Si₃₆O₇₂. As a whole of 1.5
43 water molecules were located in the porosity before high pressure experiments (structural data
44 are reported in SI). These data were taken as starting structural model for HP refinement.
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2.2. Porosimetric measurements with electrolyte solutions

The intrusion–extrusion of water and aqueous salt solutions in Si-CHA was performed at room temperature using a Micromeritics mercury porosimeter (Model Autopore IV) in the range of 0.1–350 MPa. Before the measurements, the zeolite powder was compressed in a pellet. The cell containing zeosil pellet and electrolyte aqueous solution consists in a polypropylene cylinder of 2 mL capacity sealed using a mobile piston. This cell is introduced in the glass cell of the porosimeter, which is filled with mercury. The volume variation is determined from the capacity measurement, which depends on the height of mercury in the capillary tube of the glass cell. The experimental intrusion–extrusion curve is obtained after subtraction of the curve corresponding to the compressibility of the medium. Three intrusion-extrusion cycles (with measurement time of 1.5 hour per cycle) were performed. The values of the intrusion (P_{int}) and extrusion (P_{ext}) pressures correspond to that of the half volume total variation. Pressure is expressed in MPa, and volume variation in mL per gram of anhydrous calcined zeolite. The experimental error is of the order of 1% on the pressure and on the volume.

2M and 3M aqueous solutions of NaBr, NaCl and CaCl₂ were used for intrusion-extrusion experiments, in order to study the influence of both cation (Na⁺ and Ca²⁺) and anion (Br⁻ and Cl⁻) nature. The concentrations were chosen to be sufficiently high in order to observe the influence of salt ions on the intrusion, but quite below the saturated ones, in order to avoid possible crystallization of the salts from solutions under HP. From now on, the “Si-CHA – salt aqueous solution” systems will be denoted shortly with the label “Si-CHA/Salt”.

2.3. High pressure synchrotron X-ray powder diffraction experiments

HP *in situ* XRPD experiments were performed by means of modified Merrill-Basser Diamond Anvil Cell⁴⁵. Pressure was calibrated using the ruby fluorescence method⁴⁶ on the non-linear hydrostatic pressure scale (estimated error is 0.05 GPa)⁴⁷. As mentioned above, five different penetrating Pressure Transmitting Media (PTM) were used: NaCl 2M and 3M, NaBr 2M and CaCl₂ 2M and 3M aqueous solutions. The experiments with 3M NaBr solution were not carried out since the salt crystallizes at low pressure. Measurements, in Debye Scherrer geometry, on “Si-CHA/NaCl” and “Si-CHA/CaCl₂” systems were done at SNBL1 (BM01) beamline at ESRF (Grenoble, France) with a fixed wavelength of 0.68202 Å. Diffraction intensities were collected by Pilatus IP detectors (with pixel dimensions of 172x172 μm) positioned at a fixed distance of 239 mm. The exposure time for each image was of 30 and 60 s for “Si-CHA/NaCl” and “Si-CHA/CaCl₂” systems,

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3 respectively. Data, on the “Si-CHA/NaBr” system, was obtained at BL04-MSPD beamline at Alba
4 (Barcelona, Spain) by CCD camera, SX165 (Rayonix). Wavelength was set at 0.5340 Å and sample-
5 detector distance was 160 mm. Each point collection lasted 50 s. Each collection was preceded by
6 about 15 minutes of pressure stabilization. The powder patterns of the five systems were collected
7 in the pressure ranges (P ranges) reported in Table 1 and after pressure release at ambient
8 conditions. Diffraction images were integrated and treated obtaining one-dimensional patterns
9 using Dioptas program ⁴⁸.
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16 Table 1: Data collection pressures for the investigated systems

| System | P range (GPa) | P of salt crystallization (GPa) |
|-----------------------------|------------------------|---------------------------------|
| Si-CHA/NaCl 2M | P _{amb} -2.91 | 2.91 |
| Si-CHA/NaCl 3M | P _{amb} -1.98 | |
| Si-CHA/NaBr 2M | P _{amb} -3.45 | 3.45 |
| Si-CHA/CaCl ₂ 2M | P _{amb} -1.07 | 1.07 |
| Si-CHA/CaCl ₂ 3M | P _{amb} -1.29 | |

31 **2.4. Rietveld structure refinements**

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34 Structure refinements were performed by means of Rietveld method using GSAS package ⁴⁹ with
35 the EXPGUI ⁵⁰ interface. No phase transitions were observed under compression in the
36 investigated systems, therefore profile fitting was carried out using the original $R\bar{3}m$ (hexagonal
37 axes) space group. Cell parameters were determined up to 1.83, 2.16 and 0.82 GPa for “Si-
38 CHA/NaCl”, “Si-CHA/NaBr” and “Si-CHA/CaCl₂” systems, respectively. However, the limited data
39 quality of the patterns collected at HP allowed performing the structural investigations only up to
40 1.21, 0.96 and 0.82 GPa, respectively. Data were fitted refining 2θ -shift, scale and unit cell
41 parameters. A Chebyshev polynomial with 30 coefficients was applied to refine the background,
42 while Thomson pseudo-Voigt function ⁵¹ was used for the peak profile, setting the peak cut-off as
43 0.1% of the peak maximum. The initial atomic framework coordinates were taken from the P_{amb}
44 capillary data collection. In all the refinements the framework coordinates were refined imposing
45 soft-restraints to the Si-O distances (1.60 Å) and gradually decreasing their weight (up to 10) after
46 the initial stages. The Fourier Difference Map of the electronic density was inspected in order to
47 locate the intruded extra-framework species. The position and the chemical nature of the intruded
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atoms were carefully defined taking into account the bond distances and mutually exclusion rules³⁷. The interactions between anion-cation, anion-water, cation-water, water-water and the interactions of all the extra-framework species with the framework were then monitored during the refinements. Isotropic thermal parameters were forced to be equal for all the framework oxygen atoms; identical constraints were applied to the extra-framework atoms belonging to the same chemical species. Selected examples of observed and calculated profiles are shown in the Supporting Information (Figure S4-S13), where some refined structures are reported, as well.

2.5. Other characterizations

Si-CHA sample was characterized before and after intrusion-extrusion experiments with 2M NaBr solution by thermogravimetric analysis, nitrogen adsorption-desorption and ²⁹Si solid-state NMR spectroscopy. The description and the results of these characterization methods are reported in Supporting Information.

3. Results

3.1. X-ray diffraction results

3.1.1. Cell parameters dependence on pressure

The unit cell parameter values derived from the experiments performed intruding Si-CHA with 2M or 3M solutions of the salts are reported in Table 2 and Table S35, respectively. The inspection of the Tables indicates that the evolution of the unit cell parameters upon intrusion-extrusion is almost independent on the solution concentration. Therefore, in the following, only the structural data obtained using 2M electrolyte solutions will be discussed.

The evolution of Si-CHA cell parameters as a function of pressure for the investigated systems are reported in Figure 1 (as normalized values). For all the systems, the cell parameters were measured also upon pressure release to ambient conditions (labeled from now on as P_{amb} (rev)).

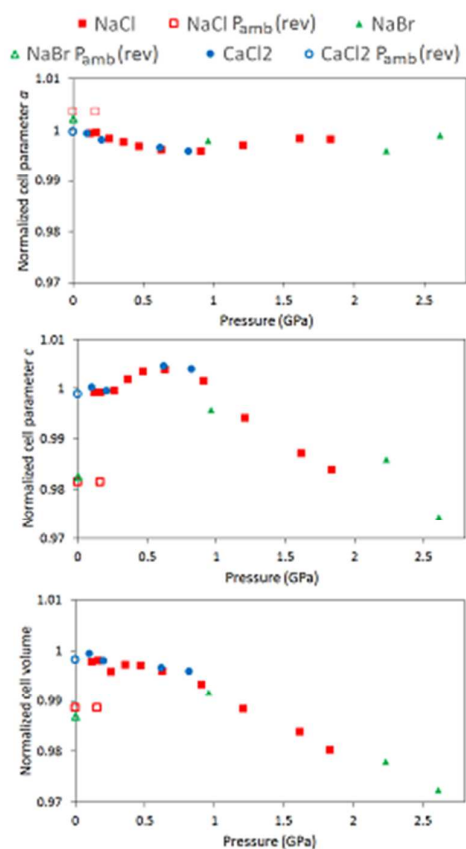


Figure 1: Evolution upon pressure of the normalized cell parameters of the investigated systems during the intrusion-extrusion process (full symbols: values obtained during the intrusion process, open symbols: refer to the cell parameter values measured upon decompression to ambient conditions P_{amb} (rev)). All the data are relative to the 2M electrolyte solutions.

Table 2: Cell parameters as function of the pressure during the intrusion-extrusion process in the investigated systems with 2M initial concentration. The label P_{amb} (rev) refers to the unit cell parameter values measured upon pressure release to ambient conditions.

| System | Pressure (GPa) | a (Å) | c (Å) | V (Å ³) |
|------------------|-----------------|------------|------------|---------------------|
| Si-CHA Capillary | P_{amb} | 13.5453(2) | 14.7636(5) | 2345.87(8) |
| Si-CHA/NaCl | 0.12 | 13.5362(1) | 14.7515(2) | 2340.80(2) |
| Si-CHA/NaCl | 0.16 | 13.5380(1) | 14.7511(2) | 2341.32(3) |
| Si-CHA/NaCl | 0.26 | 13.5233(1) | 14.7507(2) | 2336.19(3) |
| Si-CHA/NaCl | 0.36 | 13.5141(1) | 14.7901(3) | 2339.23(4) |
| Si-CHA/NaCl | 0.47 | 13.5028(1) | 14.8139(3) | 2339.11(4) |
| Si-CHA/NaCl | 0.63 | 13.4928(1) | 14.8187(3) | 2336.39(5) |
| Si-CHA/NaCl | 0.91 | 13.4892(3) | 14.7859(6) | 2329.98(8) |
| Si-CHA/NaCl | 1.21 | 13.5057(2) | 14.6773(2) | 2318.53(8) |
| Si-CHA/NaCl | 1.61 | 13.5227(2) | 14.5732(4) | 2307.86(5) |
| Si-CHA/NaCl | 1.83 | 13.5209(2) | 14.5242(4) | 2299.52(6) |
| Si-CHA/NaCl | P_{amb} (rev) | 13.5942(1) | 14.4900(3) | 2319.02(4) |
| Si-CHA/NaBr | 0.96 | 13.5182(2) | 14.7012(5) | 2326.62(9) |
| Si-CHA/NaBr | 2.23 | 13.4905(3) | 14.5558(8) | 2294.16(1) |
| Si-CHA/NaBr | 2.61 | 13.5320(5) | 14.3857(1) | 2281.3(2) |
| Si-CHA/NaBr | P_{amb} (rev) | 13.5753(3) | 14.5055(6) | 2315.07(9) |

| | | | | |
|--------------------------|------------------------|------------|------------|------------|
| Si-CHA/CaCl ₂ | 0.10 | 13.5396(1) | 14.7668(3) | 2344.39(4) |
| Si-CHA/CaCl ₂ | 0.20 | 13.5355(1) | 14.7565(3) | 2341.34(5) |
| Si-CHA/CaCl ₂ | 0.62 | 13.4918(1) | 14.8303(3) | 2337.86(5) |
| Si-CHA/CaCl ₂ | 0.82 | 13.4915(1) | 14.8211(3) | 2336.31(4) |
| Si-CHA/CaCl ₂ | P _{amb} (rev) | 13.5409(1) | 14.7473(2) | 2341.74(4) |

Regardless from the composition of the penetrating electrolyte solution, the evolution of the cell parameters with pressure shows common trends. In particular, from Figure 1 and Table 2, the following main steps are recognizable:

- 1) From P_{amb} to 0.20 GPa, cell volume displays a very small decrease.
- 2) Upon rising pressure up to ~0.6 GPa, *c* increases (0.45 % with respect to the ambient conditions). Since the *c* increase is associated to a small *a* reduction, the unit cell volume slightly decreases or remains almost constant.
- 3) These trends completely change in the range 0.63 < P < 2.61 GPa: a cell volume reduction is observed due to a strong *c* decrease (coupled with an only slight *a* increase).
- 4) Conversely to the intrusion process, the three systems have different behaviors upon pressure release. While “Si-CHA/CaCl₂” system completely regains the initial unit cell parameters, “Si-CHA/NaCl” and “Si-CHA/NaBr” systems do not recover the original unit cell volume. In both cases, this is due to *c* parameter, which does not regain its original value remaining shorter.

3.1.2. Structural features of the “Si-CHA/NaCl” system

The main finding of our experiments is that the application of pressure induces the penetration of both water molecules and Na⁺ and Cl⁻ ions inside Si-CHA structure. Figure 2 and Table 3 report the number of the intruded molecules per Si-CHA unit cell (36 SiO₂) as a function of pressure. In addition, Figure 2 shows the decrease of the available accessible volume starting from the ambient value of 17.27% of Si-CHA accessible volume⁵². The overall intruded volume was calculated taking into account the number of the absorbed species (water molecules and ions) and their volumes. Specifically, the ionic radii and the kinetics radii were considered for ions and waters molecules volume calculation, respectively⁵³.

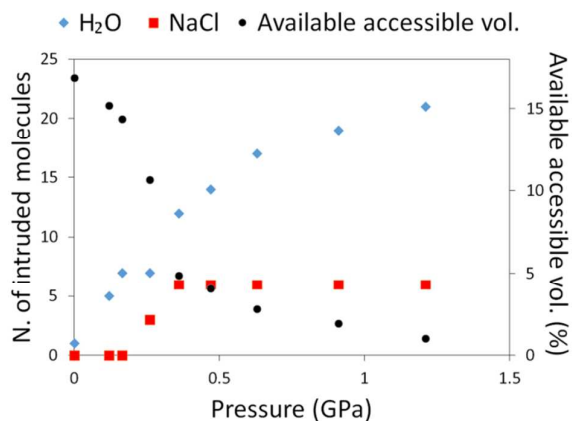


Figure 2: Extra-framework unit cell content (in terms of number of intruded molecules) and available accessible volume as a function of pressure during the intrusion process in the “Si-CHA/NaCl 2M” system.

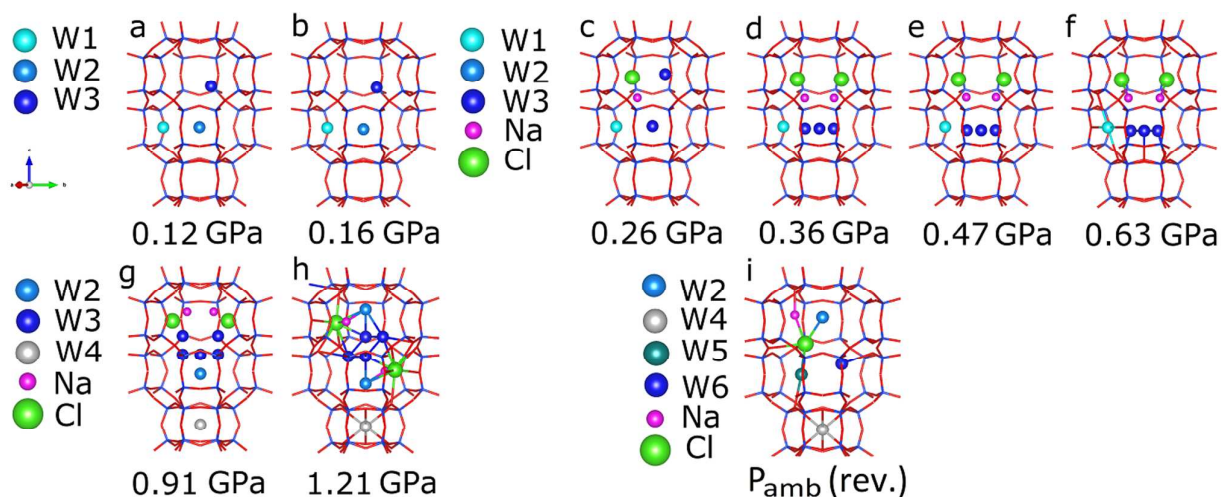


Figure 3: Structural evolution of the extra-framework sites as a function of the pressure during the intrusion-extrusion process of 2M NaCl aqueous solution in Si-CHA. Bond distances are drawn only for the structure of the highest pressure and after pressure release.

Table 3: Number of salt/water molecules hosted in Si-CHA unit cell as function of pressure during the intrusion-extrusion process in the “Si-CHA/NaCl 2M” system.

| Pressure (GPa) | W1 | W2 | W3 | W4 | W5 | W6 | NaCl |
|----------------|-----|-----|------|----|----|----|------|
| 0.12 | 1.3 | 0.6 | 3.1 | | | | |
| 0.16 | 1.9 | 1 | 4.3 | | | | |
| 0.26 | 2.6 | | 5 | | | | 3 |
| 0.36 | 3.4 | | 8.5 | | | | 6 |
| 0.47 | 4.2 | | 10.1 | | | | 6 |
| | | | 10 | | | | |

| | | | | | | |
|-----------------|-----|------|------|------|-----|---|
| 0.63 | 5.1 | 11.7 | | | 6 | |
| 0.91 | | 2.6 | 16.2 | 0.12 | 6 | |
| 1.21 | | 5.7 | 13.9 | 1.9 | 6 | |
| P_{amb} (rev) | | 3 | 3 | 2.9 | 1.8 | 2 |

As shown in Figure 3 (where intruded species inside the CHA cage at selected P values are reported), up to 0.16 GPa, only water molecules enters the porosity (reaching a value of about 7 mol. per u.c., Table 3). At this stage, neither water-water nor water-framework bonding interactions are present.

Na^+ and Cl^- ions -at bonding distance 2.75 Å- penetrate the zeolite cage at 0.26 GPa. Their amount rises reaching the maximum value of about 6 NaCl molecules per u.c. at 0.36 GPa, accompanied by the increase of the water content. At 0.63 GPa, the structural arrangement of the extra-framework species inside CHA cage can be described as formed by two distinct “layers”, not bonded to each other. The first layer is formed by a cluster of three water molecules (placed at about 3 Å from the framework), while the second one contains two distinct NaCl molecules (Figure 3) with only weak interactions with the framework. Above 0.63 GPa, the available accessible volume decreases more slowly: water molecules continue to intrude, while the ions content remains almost constant.

At 1.21 GPa the extra-framework species occupy the whole available accessible volume (Table 3 and Figure 2). Their structural organization – marked by the intensity change of some diffraction peaks – is radically changed (Figure 3): water is appreciably present also inside the double 6-ring (d6R), and an aggregate, formed by H_2O and NaCl, hydrogen-bonded to the framework occupies the CHA cage.

The structure refinement of the pattern collected upon decompression to P_{amb} ($P_{amb}(\text{rev})$) reveals the irreversibility of the NaCl aqueous solution intrusion process: the original cell volume is not recovered (Figure 1) and some extra-framework species remain in the zeolite cavities (Figure 3). In each CHA cage, a chlorine anion coordinates one sodium cation and two water molecules (W2 and W5, at a distance of about 3 and 3.5 Å, respectively). The salt species are connected to the framework oxygen atoms by Na^+ cations and to silanol defects by Cl anion interactions ($\text{Cl-O1}=3.4$ Å, and $\text{Cl-O2}=3.37$ Å). Additional water molecules, bonded to the framework, remain inside the d6R and in the CHA cage. Therefore, at $P_{amb}(\text{rev})$, half of the cell accessible volume is still occupied by the intruded species.

3.1.3. Structural features of the “Si-CHA/NaBr” system

Figure 4 and Table 4 report the structural organization of the extra-framework species in the “Si-CHA/NaBr 2M” system at 0.96 GPa and after decompression to ambient conditions. At 0.96 GPa, each CHA cage contains two molecular aggregates, interacting through water-water bonds. In each aggregate a bromine anion, bonded to a framework defect, is coordinated to Na, W7 and W3. W3 is, in turn, bonded to another water molecule (W1) placed in the middle of the so-called 8MR window. At this pressure, the accessible volume of the cage is completely occupied, except for the d6R, which is only partially filled by water.

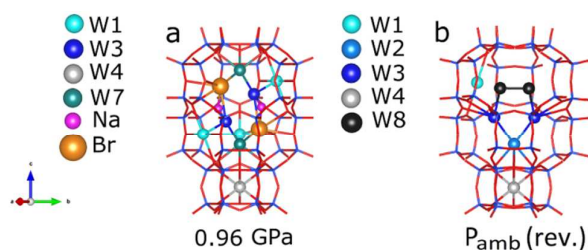


Figure 4: Results of the structure refinements at 0.96 GPa (a) and after pressure release to ambient conditions ($P_{\text{amb}}(\text{rev})$) (b) for Si-CHA intruded with 2M NaBr solution.

Table 4: Number of salt/water molecules present in the unit cell of Si-CHA during the intrusion-extrusion process of 2M NaBr solution.

| Pressure (GPa) | W1 | W2 | W3 | W4 | W7 | W8 | NaBr |
|------------------------------|-----|----|-----|-----|-----|-----|------|
| 0.96 | 9 | | 6.5 | 0.5 | 7.5 | | 5.5 |
| $P_{\text{amb}}(\text{rev})$ | 1.6 | 3 | 4.5 | 3 | | 5.5 | |

As already discussed for the “Si-CHA/NaCl” system (Table 2 and Figure 1), also for the “Si-CHA/NaBr” the intrusion phenomenon is not reversible, since the original cell volume is not recovered upon decompression. Concerning the extra-framework content, at $P_{\text{amb}}(\text{rev})$ while Na^+ and Br^- ions are released, some water molecules remain trapped inside the pores, occupying half of the available volume. A water cluster (made by one W2, two W3 and two W8 molecules, Figure 4b) is placed at the center of the cage, interacting with the framework. Far from the cluster, even W1 and W4 (in the 8MR window and in the d6R, respectively) remain trapped in the pores.

3.1.4. Structural features of the “Si-CHA/ CaCl_2 ” system

Figure 5 shows the variation of extra-framework content and of the residual available accessible volume during the intrusion of CaCl_2 aqueous solution in Si-CHA. Up to 0.20 GPa only water molecules were located in the zeolite pores. Their number in each unit cell increases from 2 to 10 between 0.10 and 0.20 GP (Table 5). Above this pressure the water amount increase is accompanied by the initial intrusion of Ca^{2+} and Cl^- ions (Figure 6). At 0.62 GPa, Ca site is at coordination distance with two Cl^- sites. Each anion is linked to a framework silanol and bonded to the water molecule W1_B , located in the neighboring cage (Figure 6d). As a whole, each cage hosts six water molecules: four W1_B , one W2 and one W3 (Figure 6c). This arrangement is very stable also at higher pressure (0.82 GPa, Figure 6d). During decompression (Figure 6e and Table 5), almost all the extra-framework content is released and at ambient conditions chabazite recovers its original cell volume (Figure 1).

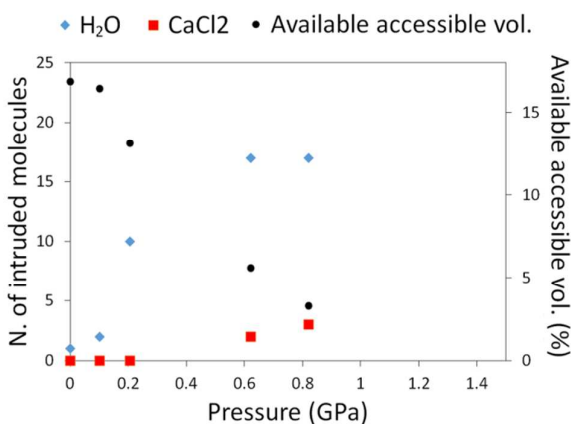


Figure 5: Extra-framework cell content, in terms of molecules number, and the available accessible volume (values related to the secondary axis) are reported as a function of the pressure during the intrusion process in the “Si-CHA/ CaCl_2 2M” system.

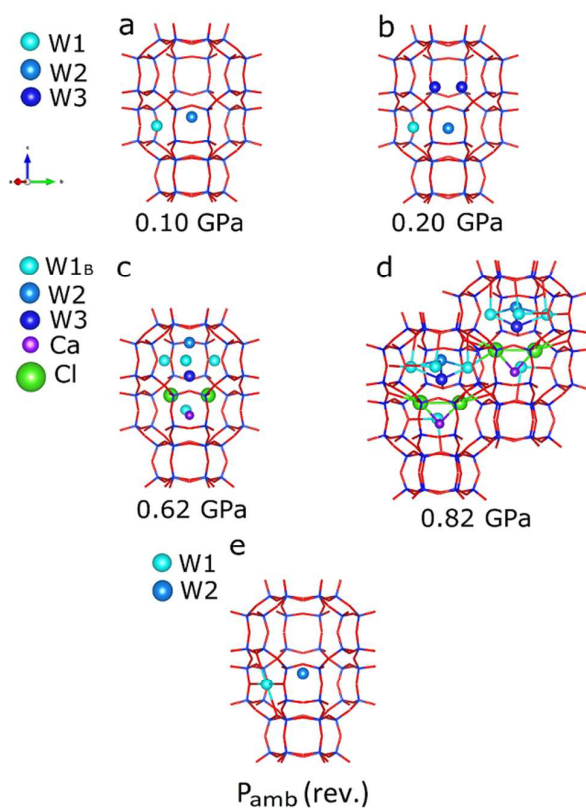


Figure 6: Structural evolution of the extra-framework atoms as a function of the pressure during the intrusion-extrusion process of 2M CaCl_2 aqueous solution in Si-chabazite. For sake of clarity, bond distances are drawn only for the structure at highest pressure and after pressure release. In figure d) interaction between two neighboring cages is displayed.

Table 5: Salt/water molecules cell content as function of the pressure during the intrusion-extrusion process in the “Si-CHA/ CaCl_2 2M” system.

| Pressure (GPa) | W1 | W1 _B | W2 | W3 | CaCl ₂ |
|------------------------|-----|-----------------|-------|---------|-------------------|
| 0.10 | 0.4 | | 1.6 | | |
| 0.20 | 2.4 | | 1.1 | 6.6 | |
| 0.62 | | 11.2 | 3 fix | 3.4 fix | 2.5 |
| 0.82 | | 11.3 | 3 fix | 3.3 fix | 3 |
| P _{amb} (rev) | 0.5 | | 2 | | |

3.2. Results of porosimetric measurements

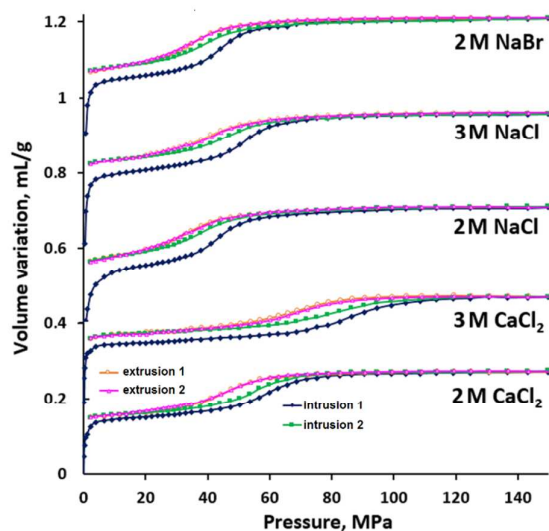


Figure 7 shows the P-V diagrams determined for three intrusion-extrusion porosimetric cycles on Si-CHA/NaBr, Si-CHA/NaCl and Si-CHA/CaCl₂ systems, using both 2M and 3M concentrations of the electrolyte solutions. The curves are shifted along the Y-axis for a better visibility. The 0-2 MPa range on X-axis is not shown (for 2nd and 3^d cycles), since the observed volume variation in this pressure range corresponds to the intrusion of liquid in the interparticular porosity of the pellet (see 1st cycle). Only the first two cycles are presented, since the 2nd and the 3^d ones are superimposable. The measurements were performed up to the pressure of 350 MPa, but only the 0-150 MPa range is shown for better visibility, because no volume variation is observed after 130 MPa. The intrusion-extrusion characteristics are reported in Table 6.

In the first cycle, a partially irreversible intrusion is observed for all the systems that exhibit a combination of bumper (partially irreversible intrusion) and shock-absorber (reversible intrusion with relatively large hysteresis) behavior. In the following cycles, the intrusion is fully reversible and the systems display a spring behavior with a small hysteresis between the intrusion and the extrusion curves. The same behavior was previously observed for the “Si-CHA/water” system³⁰. As expected, the use of electrolyte solutions allows to increase significantly the intrusion pressure, which, in turn, increases with the salt concentration (from 29 MPa for pure water³⁰ up to 90 MPa for 3M CaCl₂ aqueous solution). The decrease of the intrusion pressure from the 1st to 2nd cycle can be ascribed to the formation of hydrophilic silanol defects into the zeosil pores after the first cycle.

Figure 7. Intrusion-extrusion curves of the « Si-CHA/NaBr 2M», « Si-CHA/ NaCl 2M», « Si-CHA/NaCl 3M », « Si-CHA/CaCl₂ 2M» and « Si-CHA/CaCl₂ 3M» systems. The curves are shifted along Y axis for a better visibility.

Table 6. The characteristics of “CHA-type zeosil /NaBr, NaCl or CaCl₂ solution” systems: Intrusion (P_{int}) and Extrusion (P_{ext}) Pressures, Intruded (V_{int}) and Extruded (V_{ext}) Volumes, System behavior.

| System | P_{int} (MPa) | P_{ext} (MPa) | V_{int} (mL/g) | V_{ext} (mL/g) | Behavior |
|------------------------------|-----------------|-----------------|------------------|------------------|-----------|
| Si-CHA/NaBr 2 M | 44*/39** | 35 | 0.165*/0.135** | 0.135 | B+SA*/S** |
| Si-CHA/NaCl 3 M | 54*/47** | 41 | 0.170*/0.135** | 0.135 | B+SA*/S** |
| Si-CHA/NaCl 2 M | 44*/38** | 35 | 0.170*/0.135** | 0.135 | B+SA*/S** |
| Si-CHA/CaCl ₂ 2 M | 60*/55** | 47 | 0.130*/0.115** | 0.115 | B+SA*/S** |
| Si-CHA/CaCl ₂ 3 M | 90*/83** | 72 | 0.130*/0.105** | 0.105 | B+SA*/S** |

* - 1st cycle; ** - 2nd and 3^d cycles. B = Bumper, SA = Shock Absorber, S = Spring.

4. Discussion

4.1. Cell parameters behavior

The pressure dependences of the cell parameters of the three Si-CHA/electrolyte systems display some interesting similarities. To interpret them, the unit cell evolutions can be related to the variations of the intruded extra-framework volume (or, in other words, to the residual available accessible volume). Regardless the composition of the systems, three main sections are evident in Figure 8:

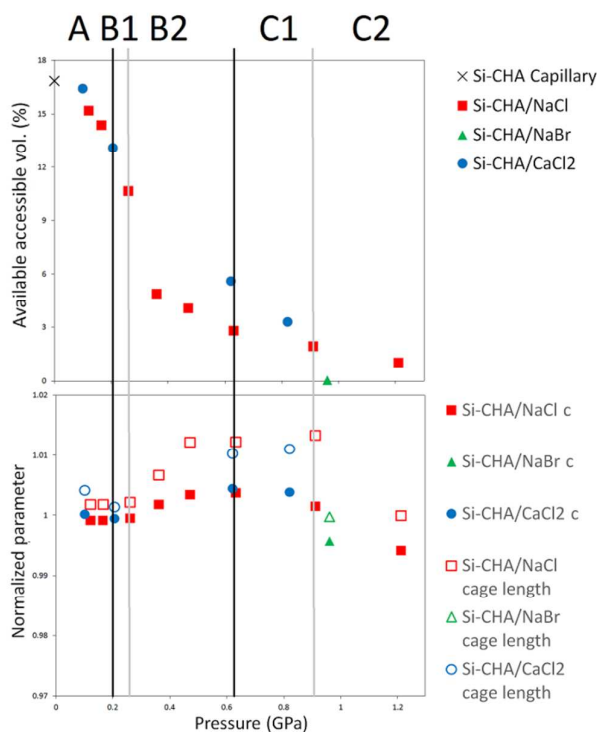


Figure 8: Available accessible cell volume of the investigated systems (top) and evolution of c parameter and of the CHA cage length (bottom) as a function of pressure. Three main sections are evidenced and correlated with the structural evolution, as explained in the body text.

i) A section (0.12-0.20 GPa): cell volume compression is related to a slight c reduction. CHA cage is almost empty, except for some water molecules (two of them already present in the Si-CHA at P_{amb}).

ii) B section (0.20-0.63 GPa): at 0.26 GPa (B1 section), ions start to enter the zeolite in the Si-CHA/NaCl system, (see Figure 2). The penetration of CaCl_2 molecule is observed from 0.62 GPa, even if we can not excluded a penetration in early stages of compression due to the lack of diffraction data in the range 0.20-0.62 GPa. It is reasonable to suppose that a certain pressure is required to induce ion desolvation in the aqueous solution, so allowing their intrusion into chabazite pores³⁷. Ion content increases as a function of pressure and at 0.63 GPa NaCl and CaCl_2 molecules reach the number of 6 and 2.5 per unit cell, respectively, while the water molecules are 17 in both cases (section B2). In this P range, a axis slightly decreases, whereas c undergoes a significant increment. This variation is mainly driven by the deformation of CHA cage, which stretches in order to accommodate the increasing extra-framework intruded species (Figure 8).

iii) C section ($P > 0.63$ GPa): between 0.63 and 0.90 GPa (C1 section), water content increases, while ions content remains almost stable. The available accessible volume is less than 3% and decreases even more as an effect of further water penetration. This trend is more evident above

0.90 GPa. In C2 section, CHA cage is completely filled in both the studied systems (NaCl and NaBr) and just few water molecules can be intruded in the d6R. At this stage, the electrolyte solutions in the diamond anvil cell (DAC) behave essentially as non-penetrating P-transmitting media inducing a relevant decrease of the *c* axis and of the unit cell volume (Figure 8).

4.2. Structural modifications induced by intrusion/extrusion processes

The intrusion of Si-CHA by different electrolyte aqueous solutions induces noticeable structural modifications in all the investigated systems. Referring again to Figure 8 and to the results of the structure refinements, it can be observed that:

i) Regardless the penetrating media, in A section water molecule configuration changes coherently with pressure: W1 and W2, already existing in Si-CHA at ambient conditions, increase their contents and at 0.12 GPa a new position, labeled W3, is occupied. No significant framework deformation is observed in this P range, neither in CHA cage nor in the d6R.

ii) After ion intrusion, noticeable differences in the nature, amount (Figures 2 and 5) and arrangement (Figures 3, 4 and 6) of the extra-framework contents are visible, depending on the intruded electrolyte solution. This is especially evident observing the structures of the three systems at the highest investigated pressures. Let us compare first Si-CHA/NaCl at 1.21 GPa and Si-CHA/NaBr at 0.96 GPa. In both cases, two symmetry-related aggregates (made by one salt and three water molecules), interacting each other, occupy CHA cage. It is worth noticing that the framework/extra-framework interactions in these systems are driven by the silanol defects. In fact, the two anions (Cl^- and Br^-) are placed at bonding distance to the framework silanols (3.13-3.27 Å and 3.24 Å, respectively). Chlorine site is near the 8MR-window, whereas bromine near the 4MR window. Br^- is more weakly bonded to the framework than Cl^- (Table 7), coherently with the lower charge density of bromine. In both systems, cations are in a inner position with respect to the anions in the cage and coordinates one water molecule. The main differences between the arrangements of the two extra-framework aggregates concern the water bonding systems. In “Si-CHA/NaCl” three water molecules are linked to each chlorine ion (bond lengths ~3.0 and ~3.3 Å), making this cluster rather compact and solidly bonded to the framework through six anion-silanol and one water-framework interactions. Differently, in Si-CHA/NaBr, bromine is bonded to the framework by three anion-silanol interactions and it is connected to a cluster of three water molecules (W1, W3 and W7), in turn bonded to the framework.

In Si-CHA/CaCl₂ at 0.82 GPa, the extra-framework content is organized in a further different configuration (Figure 6): each cage hosts a CaCl₂ molecule and a water cluster. The bonding of the two species possibly occurs only through adjacent cages. Also in this system, the interaction between the framework and the salt molecules occurs through the two anions, by one short and two long distances (Table 7).

Table 7: Anion-framework bond distances of the three systems at the highest investigated pressure.

| Si-CHA/NaCl 1.210 GPa | | Si-CHA/NaBr 0.960 GPa | | Si-CHA/CaCl ₂ 0.820 GPa | |
|-----------------------|----------|-----------------------|----------|------------------------------------|----------|
| O1-Cl | 3.136(9) | O1-Br | 3.456(7) | O1-Cl | 3.265(6) |
| O1-Cl | 3.248(9) | O4-Br | 3.242(4) | O2-Cl | 3.322(7) |
| O2-Cl | 3.371(6) | O4-Br | 3.242(4) | O2-Cl | 3.322(7) |
| O2-Cl | 3.371(6) | | | | |
| O4-Cl | 3.277(6) | | | | |
| O4-Cl | 3.277(6) | | | | |

iii) Despite the structural differences above described, some general features emerge from the intrusion/extrusion processes of the three investigated systems. As previously discussed, regardless the nature of the electrolyte aqueous solution, the framework/extra-framework interactions are mainly driven by the bonding of the anions to the silanol defects present in CHA framework. On the contrary, both sodium and calcium coordinate the respective anion without any additional interactions with the framework. In addition, water molecules can be intruded in the d6R only above a relatively high pressure (~0.9 GPa), after the almost complete filling of CHA cage.

iv) As previously discussed, all the systems show different behaviors upon pressure release. While the intrusion process of CaCl₂ is reversible Si-CHA/NaCl and Si-CHA/NaBr do not recover their initial cell volume. Specifically, in Si-CHA/NaCl system, both NaCl and water molecules partially remain trapped inside zeolite pores; in Si-CHA/NaBr system, only NaBr is released while water molecules remain inside chabazite pores. The response to decompression of the three systems can be interpreted on the basis of the strength of the bonds among the framework and the extra-framework species: the most strongly bonded species remain trapped inside the pores during decompression.

4.3. Porosimetry data and comparison with the XRPD data

The results of HP XRPD have been compared with ones of porosimetric experiments. However, it should be taken into account that the pressure values measured during the two experiments

cannot be directly compared because of the different kinetics conditions under which pressure is applied. Specifically, in the experiments with diamond anvil cell, the applied pressure (in the GPa range) increases rapidly, whereas an intrusion-extrusion cycle with the porosimeter takes 1.5 hour (pressures in the MPa range).

P-V curves, obtained by porosimetric experiments and shown in section 3.2 (Figure 7), look very similar for the “Si-CHA/NaBr 2M” and “Si-CHA/ NaCl 2M” systems, suggesting no influence of anion nature on the intrusion process. It is known that in the case of highly concentrated solutions the influence of the anion nature can be very strong⁵⁴, but in relatively diluted ones the intrusion process seems to be more ruled by water molecules.

Concerning the cation, the influence of its nature on the intrusion pressure is significant: a higher pressure is measured in the case of CaCl₂ solutions with respect to NaCl one (see table 6). This can be related to the stronger interactions that Ca²⁺ ion establishes with the water molecules of its solvation shell.

Another peculiarity of the porosimetric results obtained using CaCl₂ with respect to NaCl and NaBr solutions, is the lower intruded volume accompanied by a higher reversibility in the first intrusion cycle. This last finding is in complete agreement with the *in situ* HP synchrotron XRPD results (see above). More in detail, the intruded volume for sodium salt solutions is ~0.165-0.170 mL/g in the first cycle, the extruded one is of ~0.135 mL/g, corresponding to an irreversibly intruded volume of ~0.030-0.035 mL/g; for CaCl₂ solutions the intruded volume is 0.130 mL/g with an irreversibly intruded one of 0.015-0.025 mL/g.

Despite the differences in the measured pressure values between the two types of experiments (porosimeter vs. diamond anvil cell), whatever the system, the intruded volumes determined by porosimetric measurements are in quite good agreement with the amount of extra-framework species determined by *in situ* HP synchrotron XRPD. The comparison of experimental and calculated values of the intruded volume given in Table 8 clearly indicates a close match between the corresponding values. The calculated intruded volumes have been derived from the number of extra-framework species supposing that these are the water molecules which determine mainly the solution volume.

Other correlations could be found between the results of the two types of experiments. At HP, the XRPD data show that the system Si-CHA/CaCl₂ contains 3 CaCl₂ and 17.5 H₂O molecules per unit cell, whereas 6 NaCl • 21.5 H₂O or 5.5 NaBr • 23.3 H₂O are present in Si-CHA/NaCl and Si-CHA/NaBr,

respectively. The lower amount of intruded species present in Si-CHA/CaCl₂ is in agreement with the lower value of the intruded volume measured by porosimetry (0.130 mL/g against 0.165-0.170 mL/g for calcium and sodium salt solutions, respectively).

Table 8. Experimental and calculated values of intruded volume for NaBr, NaCl and CaCl₂ solutions.

| System | H ₂ O molecules per unit cell ^a | Calculated intruded volume ^b (mL/g) | Intruded volume in 1 st cycle ^c (mL/g) |
|------------------------------|---|--|--|
| Si-CHA/NaBr 2 M | 23.3 | 0.19 | 0.165 |
| Si-CHA/NaCl 2 M | 21.5 | 0.18 | 0.170 |
| Si-CHA/CaCl ₂ 2 M | 17.5 | 0.145 | 0.130 |

^a – from HP in situ XRPD measurements at 1.21, 0.96 and 0.82 GPa for NaCl, NaBr and CaCl₂ solutions, respectively;

^b - Intruded volume = (number of H₂O molecules per unit cell * H₂O molecular weight)/(unit cell molecular weight* H₂O density), supposing that the water molecules determine mainly the solution volume. Density of intruded solutions was estimated as 1g/mL as for bulk H₂O. This approximation of density is due to the unknown density of such NaCl and NaBr solutions.

^c – from common intrusion-extrusion tests in porosimeter.

5. Discussion on composition of intruded species

This work has provided original structural results on both composition and site position of the extra-framework species penetrated in Si-CHA during intrusion/extrusion experiments with NaCl, NaBr and CaCl₂ aqueous solutions. Now, it is interesting to comparatively discuss the behavior of the different electrolyte solutions, related to the results of the only other intrusion/extrusion XRPD study reported in literature, concerning the penetration of MgCl₂ aqueous solution in a Si-ferrierite³⁷.

The initial composition of the electrolyte solutions used in the present study is approximately 1 CaCl₂ (NaCl, or NaBr) • 17 or 26 H₂O for 2 and 3 M solutions, respectively. At HP, the unit cell of the system Si-CHA/CaCl₂ contains 3 CaCl₂ species and 17.5 water molecules (see Fig. 5 and Table 5). This content corresponds to the composition CaCl₂ • 5.8 H₂O, close to that of a saturated CaCl₂ aqueous solution (CaCl₂ • 6.6 H₂O). A similar result was obtained in our previous work on the intrusion of MgCl₂ solution in Si-FER, where the concentration of the intruded liquid was considerably higher than that of the initial solution (MgCl₂•21 H₂O) and close to that of a saturated MgCl₂ aqueous solution (MgCl₂•10 H₂O)³⁷.

Particularly remarkable results on the composition of intruded species have been obtained. From the experiments performed with 2M NaBr solution, the resulting intruded fraction under HP has

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3 the composition $\text{NaBr} \cdot 4.2 \text{H}_2\text{O}$ (simplified from $5.5 \text{NaBr} \cdot 23.33 \text{H}_2\text{O}$), that is more concentrated
4 than the corresponding NaBr saturated aqueous solution ($\text{NaBr} \cdot 6 \text{H}_2\text{O}$).
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6
7 In the case of NaCl, the effect of the intrusion on the concentration of the intruded fraction is even
8 more pronounced: in the P range 0.26-1.21 GPa the composition of the intruded solution varies
9 from $\text{NaCl} \cdot 2.5\text{H}_2\text{O}$ to $\text{NaCl} \cdot 3.6\text{H}_2\text{O}$ (to be compared to $\text{NaCl} \cdot 9\text{H}_2\text{O}$ for the saturated one). Hence,
10 the concentration of the intruded solution is not only much higher than the initial one, but it can
11 even largely overpass the concentration of the corresponding saturated solution. This can be
12 interpreted with the high affinity and the stable bonds of the guest ions with the zeosil framework.
13 These results reinforce the hypothesis that the increase of the intrusion pressure when electrolyte
14 solutions are used instead of pure water is determined by the desolvation of the ions. They can
15 also open new perspectives in research fields such as water desalination, ions separation etc.
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23 It should be also noticed that the composition of the intruded phase varies with the applied
24 pressure. At low pressure the water is preferentially intruded into chabazite porosity, whereas at
25 higher pressure also the ions penetrate into the pores and the intruded "solution" becomes more
26 concentrated.
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32 **6. Conclusions**

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34 In the present work a pure silica chabazite has been intruded at HP by 3 different electrolyte
35 aqueous solutions with different concentrations (NaCl, NaBr and CaCl_2 with 2M and 3M initial
36 concentration). Regardless the nature of the penetrating medium, the results of this study show
37 that:
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- 41 - The use of electrolyte solutions instead of pure water allows increasing significantly the
42 intrusion pressure, which increases with the salt concentration. The nature of the cation in
43 the salt seems to influence the intrusion-extrusion pressures with higher values obtained
44 for the salts with Ca^{2+} as cation compared to the one with Na^+ cation.
45
- 46 - The composition of intruded species is different from that of initial solution and depends
47 on applied pressure. Both water and salts penetrate chabazite pores under HP, with salt
48 entering at a second stage with respect water. The resulting intruded fractions is
49 considerably more concentrated than initial electrolyte solutions and even more than
50 saturated solutions of corresponding salts.
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- The different composition between intruded fraction and initial solution reinforces the hypothesis on ion desolvation under penetration into the pores as a main reason of intrusion pressure increase.
- The initial concentration (2M or 3M) of the electrolyte solution has no significant influence on the chabazite structural evolution and on the host-guest interactions of the intruded species.
- From porosimetry experiment, in the first cycle, a partially irreversible intrusion is observed for all the systems that exhibit a combination of bumper and shock-absorber behavior. In the following cycles, the intrusion is fully reversible and the systems display a spring behavior with a small hysteresis between the intrusion and the extrusion curves.

Despite these similarities, the three systems show important differences upon pressure release, in terms of nature and amount of the retained intruded volume. These differences are linked to the presence of silanol defects, which play a crucial role in promoting the interactions with the intruded anions and in influencing the reversibility of the process.

Supporting Information Description

In the Supporting Information experimental description of thermogravimetric analysis, nitrogen adsorption-desorption and ^{29}Si solid-state NMR spectroscopy is shown. In addition, selected structures are reported in details.

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